

Article

Facile synthesis of potassium poly(heptazine imide) (PHIK) / Ti-based Metal-Organic Framework (MIL-125-NH) composites for photocatalytic applications

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125-NH₂) composites for photocatalytic applications

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KEYWORDS: Carbon nitride, Metal-Organic Framework, poly(heptazine imide), MIL-125-NH₂, Composite, Photocatalysis, photochromic effect

ABSTRACT. Photocatalytically active composites comprising potassium poly(heptazine imide) (PHIK) and a Ti-based Metal-Organic Framework (MOF, MIL-125-NH₂) are prepared *in-situ* by simply dispersing both materials in water. The driving forces of composite formation are the electrostatic interactions between the solids and the diffusion of potassium ions from PHIK to MIL-125-NH₂. This mechanism implies that other composites of poly(heptazine imide) salts and

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3 different MOFs bearing positive surface charge can potentially be obtained in a similar fashion.
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5 The suggested strategy thus opens a new avenue for the facile synthesis of such materials. The
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7 composites are shown to have a superior photocatalytic activity in Rhodamine B degradation
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9 under visible light irradiation. The reaction rate is doubled compared to that of pure MOF
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11 compound and is seven times higher than the activity of the pristine PHIK. The results of the
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13 Electron Paramagnetic Resonance (EPR) investigations and the analysis of the electronic
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15 structures of the solids suggest the electron transfer from MIL-125-NH₂ to PHIK in the
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17 composite. The possible pathways for the dye degradation and the rationalization of the
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19 increased activity of the composites are elaborated.
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26 INTRODUCTION

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29 The synthesis of composite materials is an area of much interest in photocatalysis, since the
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31 presence of different functional materials facilitates charge separation inducing charge carrier
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33 transfer between the components that enhances photocatalytic and photoelectrochemical
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35 efficiencies of the integral system.^{1,2}
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40 Many materials have been tested in the synthesis of composites. In particular, there is a recent
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42 interest in graphitic carbon nitrides (g-CN) / Metal-Organic Frameworks (MOF)
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44 heterostructures.³⁻⁵ Carbon nitrides are polymeric metal-free semiconductors extensively studied
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46 for photocatalytic applications.⁶⁻⁸ One of the main advantages of carbon nitrides is that they
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48 absorb a high energy fraction of the visible light, however, they typically have low surface areas
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50 and are characterized by high charge carrier recombination rates.⁹ g-CN is usually obtained by
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52 heating cyanamide⁴/dicyanamide^{3,5} under inert atmosphere at high temperatures (550°C).
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55 However, it has been reported that related polymers with better photocatalytic activities are
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3 obtained by heating substituted 1,2,4-triazole derivatives in LiCl/KCl eutectics. Such a procedure
4 leads to the formation of a carbon nitride polymer salt composed of layers of poly(heptazine
5 imide) with K^+ ions within the interstices.⁹ MOFs are hybrid coordination polymers whose
6 optical and electronic properties can be tuned, e.g. by changing an organic or inorganic precursor
7 to obtain isorecticular materials.^{10,11} ZIF-9^{4,5} and MIL-125³ have been employed to produce
8 carbon nitride containing composites in order to benefit from the high CO_2 adsorption capability
9 of ZIF-9, and the additional catalytically active titanium centers of MIL-125, respectively. MIL-
10 125-NH₂ is isorecticular to MIL-125, but the replacement of terephthalic acid by
11 aminoterephthalic acid leads to the decrease of the material bandgap and higher retention of
12 CO_2 .^{11,12}

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28 In general, the synthesis of composites is not straightforward and depends on the properties of
29 the individual components. Possible strategies for the synthesis of metal oxides deposited on
30 other semiconductors are pulsed laser deposition,¹³ chemical vapor deposition,¹⁴ etc. Using these
31 approaches it is possible to deposit nanometric particles over films or micrometric structures.
32 Solvothermal synthesis under controlled conditions¹⁵ is also suitable to obtain composites.
33 However, PHIK and MIL-125-NH₂ are both obtained as micrometric particles.^{9,11} That makes it
34 more difficult to ensure the crystallization of one material over the other. Many proposed routes
35 in such cases involve a sol-gel synthesis in which a powder is dispersed in a sol of the reagents
36 needed to synthesize the second material. This is especially feasible for metal oxides and TiO₂ in
37 particular.¹⁶⁻¹⁹ Other possibility is to obtain the desired composite by the impregnation
38 technique,²⁰ as reported by Zhou *et al.* to prepare a g-CN/ ZIF-9 composite.⁵

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55 In this work, a simple and straightforward way to prepare PHIK/MIL-125-NH₂ composites
56 with enhanced photocatalytic activity is reported. The suggested strategy involves only the
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3 dispersion of both materials in water. The analysis of the surface ζ -potentials of the pure solids
4 shows that the components can attract each other by means of electrostatic forces. The
5 comparative analysis of FTIR and photoluminescence spectra, together with the morphological
6 changes observed by SEM proves the formation of a composite. The possible influence of the
7 migration of K^+ ions from PHIK to MIL-125-NH₂ within the composite is discussed. The
8 superior photocatalytic activity of the composite compared to pure semiconductors is
9 exemplified in Rhodamine B (RhB) degradation. EPR studies and Mott-Schottky analysis are
10 used to reveal the electronic structure of the composite. Based on these findings, possible
11 pathways for Rhodamine B degradation are discussed. We suggest that the charge transfer from
12 MIL-125-NH₂ to PHIK is responsible for the increase of Rhodamine B photodegradation rates in
13 the presence of composites as compared to pure materials.
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30 MATERIALS AND METHODS

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33 **Chemicals.** LiCl (99%, Roth), KCl (99%, Sigma-Aldrich), 3-amino-1,2,4-triazole-5-thiol
34 (98%, Alfa Aesar), N,N-dimethylformamide (DMF, 99.8%, Sigma-Aldrich), acetonitrile (99.9
35 %, Merck), methanol (99.9%, Merck), titanium (IV) isopropoxide (97 %, Aldrich), pivalic acid
36 (99%, Roth), 2-aminoterephthalic acid (97%, Aldrich), sodium sulfate (99%, Acros Organics)
37 and Rhodamine B (95 %, Sigma Aldrich) were used as received.
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46 **Synthesis of the materials.** PHIK was prepared by a salt melt synthesis described previously.⁹
47 In a typical procedure, 3 g of 3-amino-1,2,4-triazole-5-thiol were grinded with 6.75 g of LiCl and
48 8.25 g of KCl in a mortar. The mixture was transferred to a crucible and heated under nitrogen
49 flow (15 L·min⁻¹) to 550 °C at a rate of 2.3 °C·min⁻¹, then held at 550 °C for 4h. After cooling
50 down, the product was washed with the deionized (DI) water for 12 h, then filtered and washed
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3 three more times with DI water. The humid solid was dried in a vacuum oven overnight (50 °C,
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5 10 mbar).
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9 MIL-125-NH₂ was prepared according to the method reported by Hendon *et al.*¹¹ The first step
10 involves the solvothermal synthesis of a Ti-based oxocluster Ti₈O₈(OOCC(CH₃)₃)₁₆. For that
11 purpose, a solution of 3.2 ml of titanium (IV) isopropoxide and 11.8 g of pivalic acid in 40 ml of
12 acetonitrile was heated at 120 °C during 72 h in a PARR reactor. The product was washed three
13 times with acetonitrile. Finally, 1 g of the oxocluster and 1.3 g of 2-aminoterephthalic acid were
14 dissolved in a mixture of 58 ml of DMF and 18 ml of MeOH. The solution was heated at 120 °C
15 during 72 h in a Teflon-lined PARR reactor. The solid was centrifuged and washed three times
16 with DMF and two times with MeOH, and then dried under vacuum.
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29 Composite materials were prepared *in-situ* by mixing fixed amounts of the pure materials and
30 dispersing the mixture in water. Composites designated as COM25, COM50, COM75 contained
31 25 wt.%, 50 wt.% and 75 wt.% of PHIK, respectively. For the analysis of the composite, the
32 dispersion was dried under vigorous stirring at room temperature. For comparison purposes, a
33 mixture of PHIK and MIL-125-NH₂ of the same composition as COM50 was prepared. The
34 mixture (MIX50) was not dispersed in water to avoid the formation of the composite.
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44 **Characterization.** Powder X-Ray diffraction (PXRD) patterns were measured on a Bruker D8
45 Advance diffractometer equipped with a scintillation counter detector with Cu-K α radiation ($\lambda =$
46 0.15418 nm). FTIR spectra were acquired on a Varian1000 FTIR spectrometer equipped with an
47 attenuated total reflection (ATR) unit. Nitrogen adsorption/desorption measurements were
48 performed after degassing the samples at 150 °C for 20 h using a Quantachrome Quadrasorb SI-
49 MP porosimeter at 77.4 K. Scanning electron microscopy (SEM) images were obtained on a
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3 LEO 1550-Gemini microscope. Optical absorbance spectra of powders were measured on a
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5 Shimadzu UV 2600 spectrophotometer equipped with an integrating sphere. The absorption
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7 spectra of RhB solutions were recorded on a T70 UV/VIS spectrophotometer (PG
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9 instruments Ltd.). Zeta-potentials were measured in aqueous dispersions of the solids using a
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11 Malvern Zetasizer instrument. Mott-Schottky analysis of MIL-125-NH₂ was performed using a
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13 film of the solid. The film was prepared by spin-coating (using a Laurell WS400-E spin
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15 processor) of an ethanolic dispersion (8 g·L⁻¹) on a fluorine doped thin oxide (FTO) glass. The
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17 film was immersed in a 0.5 M Na₂SO₄ aqueous solution. The impedance spectroscopy analysis
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19 was performed with a Voltalab 40 instrument at a frequency of 1 kHz, using a Pt-cable and a
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21 saturated calomel electrode as working and reference electrodes, respectively. N₂
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23 adsorption/desorption isotherms were measured at 77.4 K on a Quadrasorb SI-MP-20
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25 porosimeter (Quantachrome Instruments) after degassing the samples at 150 °C for 20 hours.
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27 The specific surface areas were calculated by applying the Brunauer-Emmett-Teller (BET)
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29 model to adsorption isotherms at 0.05 < p/p₀ < 0.3 using the QuadraWin 5.05 software package.
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31 The pore size distribution was calculated by the nonlocal density functional theory (NLDFT)
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33 method.
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41 Three sets of EPR spectra were obtained at room temperature using a Bruker ELEXSYS
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43 E500T spectrometer operating at the X-band. The first set consists of EPR spectra of solid PHIK
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45 and COM50 samples, which were smoothed using a fast Fourier transform filter for the sake of
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47 clarity. The samples for the second set were prepared by placing different amounts of MIL-125-
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49 NH₂ and COM50 (4 and 8 mg, respectively, to ensure the amount of MOF is the same) in an
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51 EPR tube, and adding 10 µl of a TEOA solution (20% in acetonitrile) purged with N₂. The tubes
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53 were irradiated with a blue LED emitting at 470 nm (3W, LUXEON III – Philips) for 2 h and
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3 then analyzed. The third set was prepared as the second one but employing PHIK and COM50
4 (1.5 and 3.0 mg, respectively, to provide similar quantities of PHIK in both samples) with 10 μ l
5 of the purged TEOA solution. The samples were irradiated with LEDs emitting at 470 and 530
6 nm (3W, LUXEON III – Philips) for 30 minutes.
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14 **Photocatalytic Rhodamine B degradation.** 5 mg of the photocatalyst (pure PHIK, MIL-125-
15 NH_2 or a composite) were dispersed in 5 ml of a 100 $\text{mg}\cdot\text{L}^{-1}$ aqueous solution of RhB. The
16 resulting dispersion was stirred for 2 hours in the dark to allow the adsorption of the dye (and the
17 formation of the composite when a mixture of MIL-125- NH_2 and PHIK was employed). Then
18 the dispersion was irradiated with a blue LED lamp emitting at 465 nm (30 W, OSA Opto
19 Lights). Aliquots of 200 μ l of the irradiated dispersion were taken at different irradiation times,
20 then centrifuged and finally diluted with 2.3 mL of DI water. The solutions were then analyzed
21 by UV-Visible light spectrometry to follow the decay of RhB concentration as a function of
22 irradiation time.
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36 **Determination of the photon flux through the reactor and apparent quantum efficiencies.**
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38 A screw-capped tube was charged with the solution of potassium ferrioxalate (5 mL, 150
39 $\text{mmol}\cdot\text{L}^{-1}$) and irradiated with a blue LED (40 W, 465 nm) for 15 s. Solution aliquot (1 mL) was
40 transferred into a beaker, phenanthroline solution (4 mL, 1 $\text{mg}\cdot\text{mL}^{-1}$) and water (20 mL) were
41 added. The procedure was repeated 3 times and the average absorbance (A_{av}) at 510 nm was
42 calculated. A solution for absorption spectrum baseline correction was prepared by mixing
43 potassium ferrioxalate (1 mL, 150 $\text{mmol}\cdot\text{L}^{-1}$), phenanthroline solution (4 mL, 1 $\text{mg}\cdot\text{mL}^{-1}$) and
44 water (20 mL). The flux of photons (Q_f) through the reactor was calculated using the equation:
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$$Q_f = \frac{A_{av} \cdot V_t \cdot V_r \cdot 60 \cdot 1000}{\epsilon \cdot l \cdot V_a \cdot 10^3 \cdot QE \cdot \tau} = \frac{1.263 \cdot 25 \cdot 5 \cdot 60 \cdot 1000}{11100 \cdot 1 \cdot 1 \cdot 10^3 \cdot 0.85 \cdot 15} = 0.06693 \quad (1)$$

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where Q_f – flux of photons through the reactor, $\text{mmol} \cdot \text{min}^{-1}$; A_{av} – Fe^{2+} /phenantroline complex absorption; ε – extinction coefficient, $\text{L} \cdot \text{mol}^{-1} \cdot \text{cm}^{-1}$; l – cuvette length, cm; V_a – aliquot volume, mL; V_t – total volume, mL; V_r – reaction mixture volume, mL; QE – quantum efficiency of the potassium ferrioxalate photoreduction; τ – irradiation time, s.

Apparent quantum efficiency of the material in a given process can be written as:²¹

$$\varphi_x = \frac{\mp(d[x]dt)}{d[h\nu]_{inc}/dt} \quad (2)$$

where $d[x]/dt$ is the rate of change of the concentration of the reactant (or product) and $d[h\nu]_{inc}/dt$ is the total optical power impinging on the sample.

Apparent quantum efficiencies of the photocatalysts were calculated assuming that degradation of Rhodamine B chromophore (absorbing at 554 nm) is a one electron/hole process (which is, of course, not true, and a number of the electrons/holes participating in dye degradation is much higher in reality). Equation (2) was then rewritten as:

$$\varphi_x = \frac{\mp(d[x]dt)}{\frac{d[h\nu]_{inc}}{dt}} = \frac{n_{RhB}}{n_{photon}} = \frac{n_0 - n}{Q_f \cdot 10^{-3} \cdot \tau} = \frac{C_0 \cdot V / M_{RhB} - C \cdot V / M_{RhB}}{Q_f \cdot 10^{-3} \cdot \tau} = \frac{V \cdot (C_0 - C_0 \cdot e^{-k \cdot \tau})}{M_r \cdot Q_f \cdot 10^{-3} \cdot \tau} = \frac{V \cdot C_0 \cdot (1 - e^{-k \cdot \tau})}{M_r \cdot Q_f \cdot 10^{-3} \cdot \tau} \quad (3)$$

where n_{RhB} – amount of Rhodamine B degraded within time τ , mol; n_{RhB} – amount of photons that passed through the reactor within time τ , mol; n_0 – initial amount of Rhodamine B, mol; n – amount of Rhodamine B after τ minutes, mol; C_0 – initial concentration of Rhodamine B, $\text{g} \cdot \text{L}^{-1}$; n – concentration of Rhodamine B after τ minutes, g/L ; M_{RhB} – molar mass of Rhodamine B, $\text{g} \cdot \text{mol}^{-1}$; k – Rhodamine B degradation rate constant, min^{-1} ; V – reactor volume, L; τ – reaction time, min.

RESULTS AND DISCUSSION

The initial selection of the suitable counterparts to form a composite was done based on the surface charges of the possible candidates. In this view, poly(heptazine imide) bearing negative surface charge and MIL-125-NH₂ having in part a positively charged surface (see below) seemed to be a suitable pair of semiconductors. The poly(heptazine imide)/MIL-125-NH₂ composites were prepared *in-situ* by simply dispersing both materials in water, and then isolated and characterized. Figure 1A shows the PXRD patterns of PHIK, MIL-125-NH₂ and their composite COM50 which was found to be the most active for Rhodamine B degradation (see below). The characteristic diffraction peaks of MIL-125-NH₂¹¹ and the two most intense signals of PHIK⁹ at 8.1° 2θ and 27.9° 2θ are observed in the pattern of the composite that clearly indicates the presence of both crystalline solids.

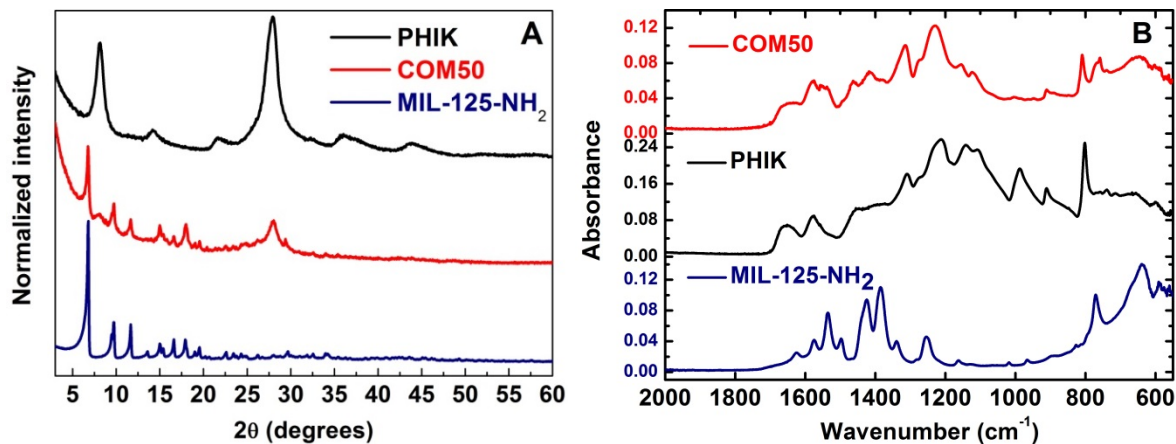


Figure 1. PXRD patterns (A) and FTIR spectra (B) of PHIK (black), COM50 (red) and MIL-125-NH₂ (blue).

Another evidence for composite formation is provided by the comparison of the FTIR spectrum of COM50 with those of pure materials, PHIK and MIL-125-NH₂, as shown in Figure

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3 1B. The spectrum of PHIK exhibits a number of typical vibrations,²² namely the broad band at
4 1100–1500 cm^{-1} , composed of the overlapping bands at 1108, 1142, 1212, 1274, 1310, 1392 and
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6 1449 cm^{-1} , which is due to the stretching vibrations of C–N and C=N bonds in the polymer, the
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8 band at 802 cm^{-1} , related to the deformation vibrations of tri-*s*-triazine ring, the peaks at 1650
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10 and 1575 cm^{-1} , assigned to N-H bending in H_2NC and HNC_2 groups, respectively, while the
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12 absorption bands at 1065 and 987 cm^{-1} are due to the asymmetric and symmetric vibrations of
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14 NC_2 bonds, respectively, in metal– NC_2 groups. In the corresponding spectrum of MIL-125- NH_2 ,
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16 absorption peaks at 1497 and 1575 cm^{-1} are due to stretching of C=C bonds in the aromatic ring,
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18 strong bands at 1340, 1384, 1424 and 1536 cm^{-1} are due to carboxylate groups, the peak at 1625
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20 cm^{-1} is related to NH bend in NH_2 -groups, the one at 1257 cm^{-1} is assigned to CN stretching
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22 vibrations, while the strong absorption bands at 770 and 641 cm^{-1} are due to O–Ti–O
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24 bonds.^{11,23,24} The FTIR spectrum of the composite material contains the characteristic features of
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26 both components; however, the majority of the absorption peaks are slightly shifted if compared
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28 to their position in the pure materials. The peak of heptazine ring deformation vibrations appears
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30 at 808 cm^{-1} in the composite instead of 802 cm^{-1} in PHIK; the O–Ti–O band in the composite is
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32 at 760 cm^{-1} compared to 771 cm^{-1} in the original MOF; the vibrations of carboxylate group in the
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34 composite are at 1416 and 1379 cm^{-1} instead of 1425 and 1384 cm^{-1} in MIL-125- NH_2 . Some
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36 peaks only appear to be shifted due to the overlapping of certain bands of pure materials, e.g. the
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38 bands at 1315 and 1229 cm^{-1} in the composite are formed from the contributions of PHIK (1310
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40 cm^{-1} and 1212 cm^{-1}) and MIL-125- NH_2 (1340 cm^{-1} and 1257 cm^{-1}). A few peaks remain
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42 unchanged upon incorporation of PHIK and MOF into the composite, e.g. those at 1650, 1575
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44 and 1497 cm^{-1} that are due to N-H bending in PHIK and C=C stretching in MIL-125- NH_2 . The
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3 observed shifts of the characteristic IR absorption bands suggest that the two compounds interact
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5 within the composite.
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9 The most remarkable change, however, is the almost complete disappearance of the PHIK
10 vibrations at 1065 and 987 cm^{-1} that are due to the vibrations of NC_2 bonds of $\text{C}_2\text{N}^\ominus \text{K}^+$ groups,
11 upon formation of the composite. This points to the redistribution of the K^+ ions within the
12 composite and their migration from the poly(heptazine imide) to MOF driven by the higher
13 acidity of the 2-aminoterephthalic acid compared to the imide. Upon migration of K^+ to MOF,
14 PHIK becomes negatively charged, what further strengthens the electrostatic interactions
15 between PHIK and MIL-125- NH_2 (see below) and contributes to the integrity of the composite as
16 illustrated further.
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28 To ensure that the observed changes in the FTIR spectra are related to the composite formation
29 and do not originate from some chemical changes in the pure materials upon their contact with
30 water, both PHIK and MIL-125- NH_2 were dispersed in water, then dried and the FTIR spectra of
31 the resulting products were compared with those of the original solids (Figure S1 A, B). As
32 expected, no chemical changes upon contact with water could be detected.
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40 To reveal the nature of the electronic interactions between the materials within the composite,
41 EPR spectroscopy studies were carried out (Figure 2). Pure solid PHIK shows a single
42 absorption signal with a g -value of 2.005, which reveals the presence of localized electrons in its
43 structure even in the dark conditions.^{25,26} In the composite, the corresponding signal is slightly
44 shifted to $g = 2.004$ and greatly enhanced indicating that MIL-125- NH_2 induces the production
45 of additional unpaired electrons in PHIK. This fact can be rationalized by the migration of K^+
46 ions from PHIK to MIL-125- NH_2 , in agreement with the FTIR studies.
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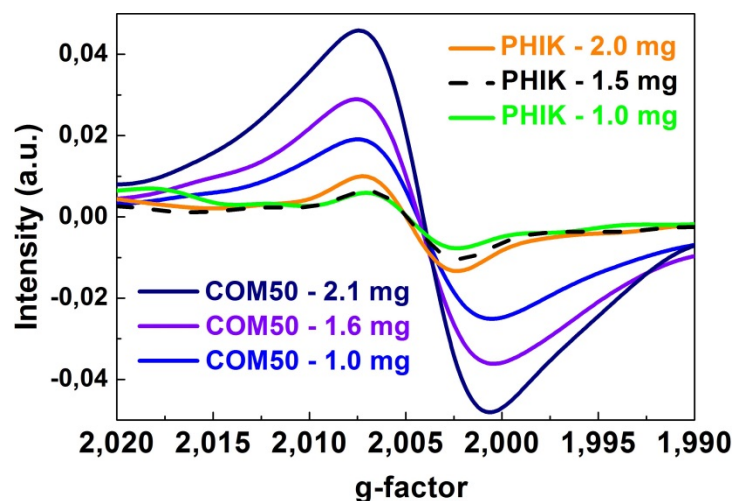


Figure 2. Smoothed EPR spectra of solid potassium poly(heptazine imide), PHIK and composite COM50 measured for different sample weights.

The morphological changes upon the formation of the composite were accessed by scanning electron microscopy (SEM) studies (Figure 3). The pristine PHIK is constituted by micrometric agglomerates covered with nanometric needles (Figure 3A). On the other hand, MIL-125-NH₂ has a disc-like morphology, about 200 nm in diameter and around 50 nm high (Figure 3B). The mixture of PHIK and MIL-125-NH₂ before water treatment (Figure S2) shows the presence of both solids with their typical morphologies. However, the mixture after water treatment exhibits a heterogeneous morphology, represented by nanoparticles of irregular size and shape, and micrometers long layered ‘belts’ (Figure 3 C, D). The irregular morphology of the sample could be a result of more or less favorable interactions of the materials within the composite, e.g. originating from the different amounts of K⁺ ions transferred from PHIK to MIL-125-NH₂ and leading to different spatial arrangements. Clearly, the results of SEM studies further support the formation of the composite upon water treatment of PHIK/MIL-125-NH₂ mixtures.

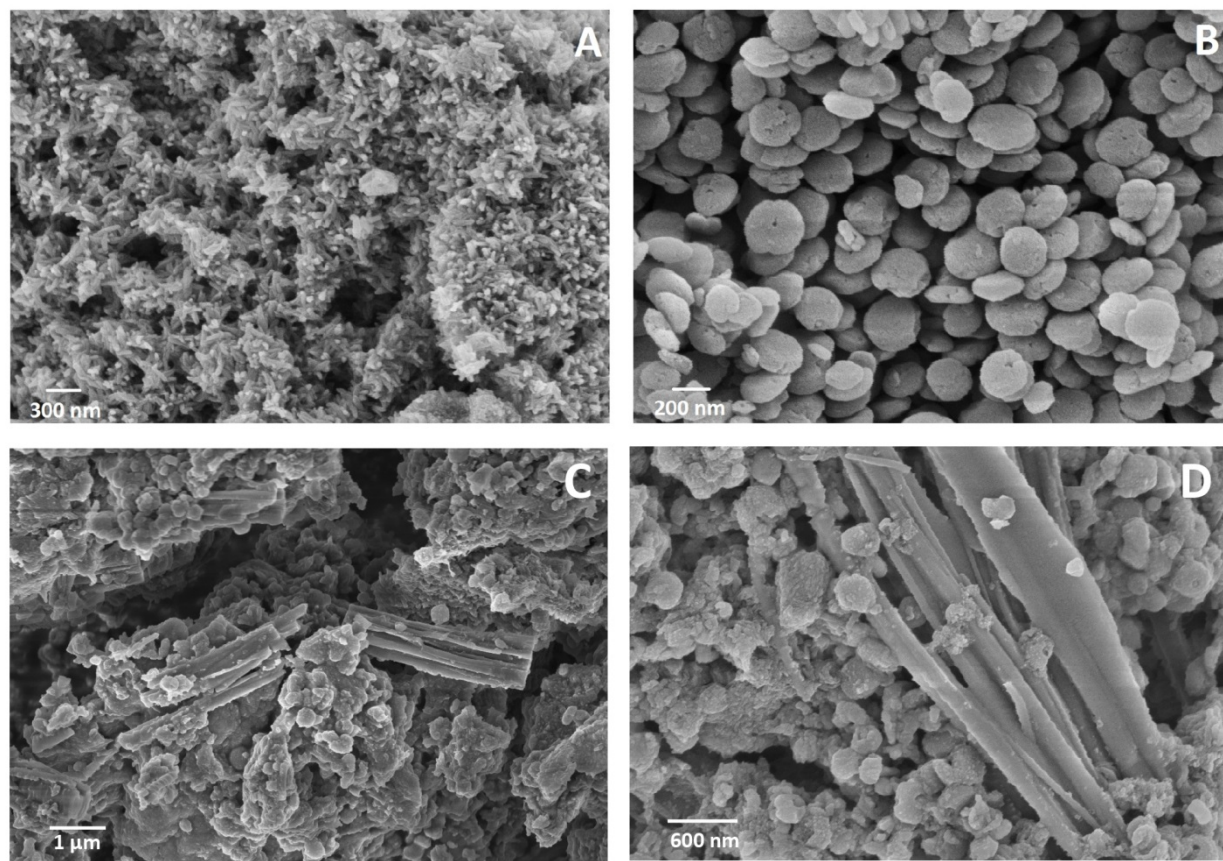


Figure 3. SEM images of PHIK (A), MIL-125-NH₂ (B) and COM50 (C, D).

The N₂ sorption isotherms and the pore size distributions in the starting materials and their composite COM50 are compared in the Figure S3 A and B, respectively. The N₂ isotherm of MIL-125-NH₂ is of type I, reflecting the presence of micropores in the material. A somewhat increased N₂ uptake at p/p_0 close to 1 is an indication of the presence of some macropores. PHIK is a non-porous solid, while the composite exhibits the presence of only few micropores indicating that the major part of micropores is lost upon complex formation. This is probably due to the migration of K⁺ ions from PHIK to MIL-125-NH₂ that partially block the micropores. The BET surface areas of MIL-125-NH₂, COM50 and PHIK are 1160, 185 and 30 m²·g⁻¹, respectively.

For further analysis of the interactions between MIL-125-NH₂ and PHIK, ζ -potential measurements were carried out (Figure 4A). PHIK has a negatively charged surface with -40 mV zeta potential value at pH 6, while MIL-125-NH₂ sample contains both positively and negatively charged particles, with an average zeta potential (-5.41 mV) close to zero. When dispersed in water, positively charged MIL-125-NH₂ particles and PHIK can be attracted together by electrostatic forces and form aggregates.

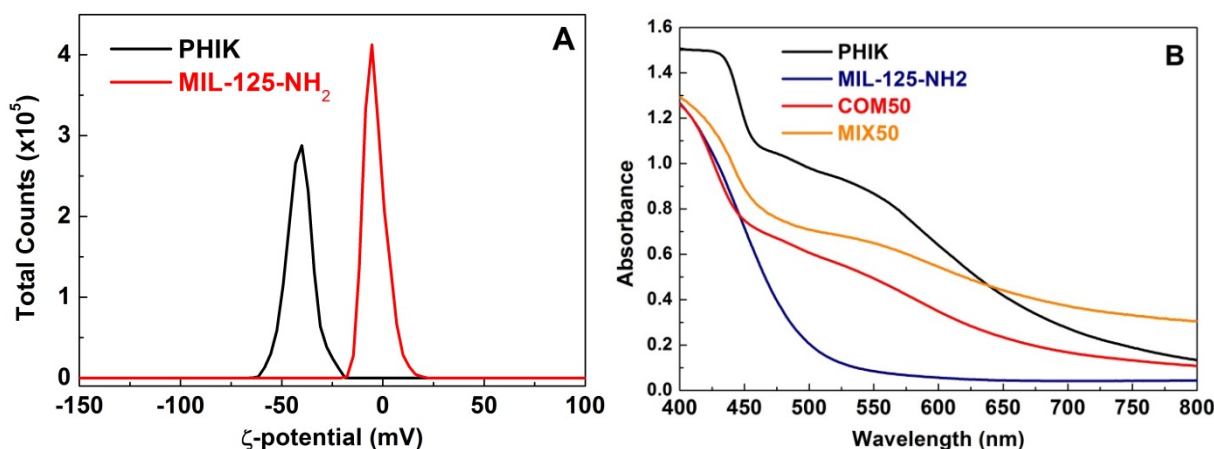
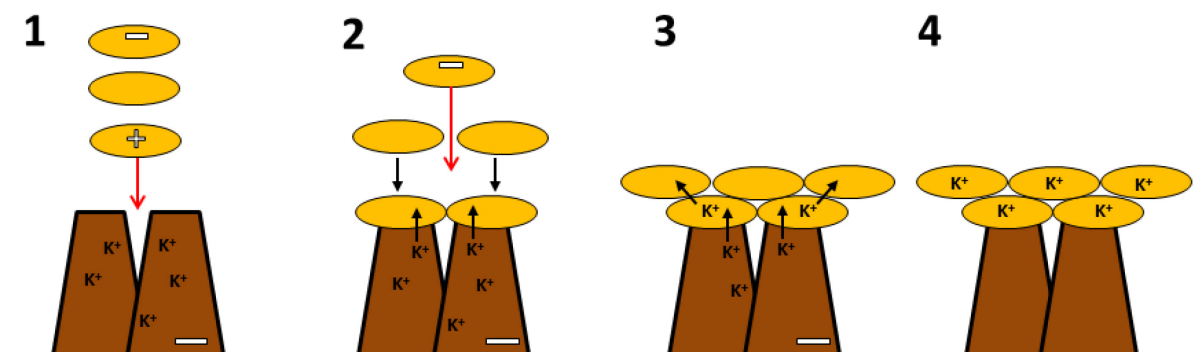


Figure 4. Zeta-potential distributions (A) of PHIK (black) and MIL-125-NH₂ dispersed in water at pH 6 (red). UV/Visible absorption spectra (B) of PHIK (black), COM50 (red), MIX50 (orange) and MIL-125-NH₂ (blue).

Summarizing, FTIR and EPR measurements unequivocally show that K⁺ ions migrate from PHIK to MIL-125-NH₂, while SEM studies illustrate the changes of morphology of PHIK/MOF mixture after being dispersed in water. Based on the fact that ζ -potential measurements show that a fraction of MIL-125-NH₂ particles have a slight positive charge, we assume that those positively charged MIL-125-NH₂ species can interact electrostatically with the negatively charged PHIK (Scheme 1, step 1). Upon the contact of the two materials, the migration of K⁺ ions from PHIK to MIL-125-NH₂ (Scheme 1, step 2) makes PHIK more negatively charged, in

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3 agreement with EPR data (see above), while MIL-125-NH₂ becomes more positively charged
4 further enhancing the electrostatic interactions between the solids. Probable chemical bonds at
5 the interface of the two materials are depicted in Scheme S1. Negatively charged and neutral
6 MIL-125-NH₂ particles can approach and contact PHIK or MOF particles due to their high
7 kinetic energy provided by agitation and electrostatic attraction to positively charged MIL-
8 125-NH₂ particles (Scheme 1, step 2). Upon the contact of the solids, a migration of additional
9 K⁺ ions from PHIK to MIL-125-NH₂ becomes possible, further strengthening the electrostatic
10 interactions in the composite (Scheme 1, step 3). The quantitative diffusion of K⁺ ions from
11 PHIK to MOF would provide the strongest interaction between PHI and MOF in the complex
12 (Scheme 1, step 4).



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43 **Scheme 1.** Proposed mechanism of the composite formation. Step 1 shows the approach of MIL-
44 125-NH₂ (yellow disks) to PHIK (brown tetrahedron) by electrostatic forces (red arrow). Step 2
45 shows the diffusion (black arrow) of K⁺ ions to MIL-125-NH₂. Other MIL-125-NH₂ particles,
46 neutral or negatively charged, can approach the surface of PHIK due to their high kinetic energy
47 provided by agitation and electrostatic attraction to positively charged MIL-125-NH₂ particles.
48 Step 3 shows the diffusion of additional K⁺ ions to MIL-125-NH₂. Step 4 shows the state of the
49 final composite with K⁺ ions inside the micropores of MIL-125-NH₂.

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Optical absorption properties of PHIK, MIL-125-NH₂, their mechanical mixture MIX50 and the composite COM50 are depicted in Figure 4B. The absorption thresholds of MIL-125-NH₂ and PHIK are approximately 525 nm and 750 nm, respectively. The visible light absorption of the MOF originates from the O to Ti charge transfer in TiO₅(OH) inorganic clusters influenced by 2-aminoterephthalate ligands.¹² The two absorption bands in PHIK spectra at $\lambda < 450$ nm and $\lambda \sim 525$ nm are due to $\pi \rightarrow \pi^*$ electron transitions in the conjugated aromatic ring system and $n \rightarrow \pi^*$ transitions involving lone pairs of the edge nitrogen atoms of the heptazine ring, respectively.^{9,27} The bandgaps of the pure materials were estimated from the Tauc plots, depicted in Figure S4. The corresponding values for PHIK and MIL-125-NH₂ are 1.9 eV and 2.6 eV, respectively, in agreement with previous reports.^{9,11} In the spectrum of COM50, the absorption threshold of the first absorption band is blue-shifted compared to the mixture and also to pure MIL-125-NH₂ that points to the slight modification of the electronic properties of materials upon composite formation. The absorbances of PHIK, COM50 and MIL-125-NH₂ in the wavelength range of 465±30 nm that corresponds to the blue LED emission spectrum used in photocatalytic experiments (see below) are related as 1:0.628:0.500.

The photocatalytic activity of the pure materials was compared with the performance of the MOF/PHIK composites in the decoloration of Rhodamine B under blue light irradiation ($\lambda = 465$ nm). This first step of Rhodamine B degradation can be due to de-ethylation of the dye molecule and/or direct oxidation of the chromophore structure.^{3,28-30} The detailed studies of the mechanism of Rhodamine B photodegradation are beyond the scope of the present contribution; and the interested reader is encouraged to refer to the dedicated literature.³¹⁻³³ Figure 5A shows the decrease of RhB relative concentrations (C/C_0) with the irradiation time in the presence of PHIK, MIL-125-NH₂, and their composites, COM25, COM50 and COM75 as photocatalysts, while

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3 Figure 5B depicts the corresponding values of the calculated rate constants. Both MOF and
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Figure 5B depicts the corresponding values of the calculated rate constants. Both MOF and
PHIK are found to degrade RhB with the rate constants $k = 0.010$ and $k = 0.003 \text{ min}^{-1}$,
respectively, however, COM50 shows twice higher activity ($k = 0.020 \text{ min}^{-1}$) compared to MIL-
125-NH₂ and is seven times more active than PHIK alone.

Apparent quantum efficiencies of the photocatalysts could be determined knowing the value of
a photon flux through the reactor (see Equation (1), Experimental Section), which was measured
using potassium ferrioxalate as a standard chemical actinometer.^{34,35} The apparent quantum
efficiencies (φ_x) calculated according to Equation (3) (see Experimental Section for the details)²¹
are $9.1 \cdot 10^{-5}$, $3.9 \cdot 10^{-5}$ and $11.8 \cdot 10^{-5}$ for MIL-125-NH₂, PHIK and COM50, respectively. These
values indicate that COM50 is 30 % more efficient than the MOF and 3 times more efficient than
PHIK. The low values of φ_x are explained by the two factors. Firstly, the calculations were done
assuming that the degradation of Rhodamine B is a one electron/hole process. This is, of course,
not true, and much greater number of the photogenerated charge carriers participate in the
complex mechanism of the dye degradation. Secondly, apparent quantum efficiency does not
take into account the fraction of light absorbed by the photocatalyst, especially considering the
presence of a dye in the reaction mixture, and therefore is in fact a lower limit on the true
quantum yield.²¹

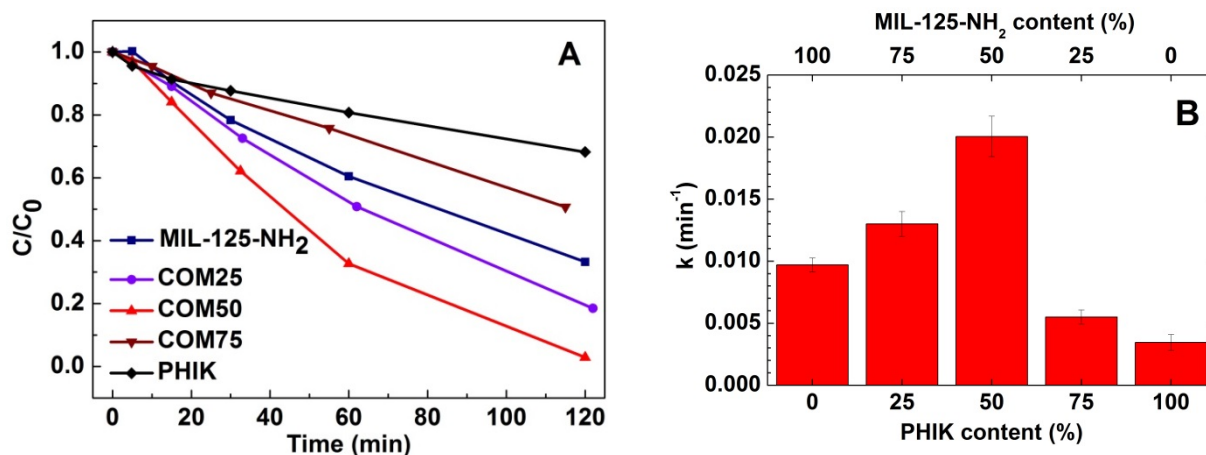


Figure 5. Degradation of RhB by MIL-125-NH₂ (blue), PHIK (black), COM25 (violet), COM50 (red) and COM75 (brown) (A). Calculated rate constants of RhB degradation using PHIK/MIL-125-NH₂ photocatalyst as a function of its composition (B).

To rationalize the synergistic effect of PHIK and MIL-125-NH₂ within the composite in RhB degradation reaction, the electronic structures of the materials should be considered. The electrochemical impedance spectroscopy and Mott-Schottky analysis were used to determine the band structure of MIL-125-NH₂ (Figure S5). The flat-band potential of MIL-125-NH₂ is found to be -0.63 V (vs. Normal Hydrogen Electrode, NHE). Since it behaves as an *n*-type semiconductor, given the positive slope of the Mott-Schottky plot, we conclude that its conduction band (CB) is located at about -0.63 V (vs. NHE). Thereby, the valence band (VB) of MIL-125-NH₂ is located at 1.97 V (vs. NHE). The comparison of the CB positions of the MOF ($E_{CB} = -0.63$ V) and PHIK ($E_{CB} = -0.21$ V (vs. RHE at pH = 7) = -0.62 V (vs. NHE)⁹) suggests that the photogenerated electrons can be transferred from MIL-125-NH₂ to PHIK. Besides, the similarity of the CB potentials of the two materials may facilitate this charge transfer. In general, PHIK/MIL-125-NH₂ composite has a type-I heterostructure.^{1,2} In such composites, the conduction and valence band of the material with the smaller bandgap (PHIK in this case) are located between the CB

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3 and VB levels of the solid with the larger bandgap (MIL-125-NH₂). The consequence of such
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5 positioning of bands is that both holes and electrons are thermodynamically allowed to migrate
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7 from MIL-125-NH₂ to PFIK.
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10 To get deeper insight into the electron transfer processes in pure materials and in COM50, EPR
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12 measurements of the irradiated anaerobic suspensions of the solids in the presence of
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14 triethanolamine (TEOA), an effective hole scavenger, were carried out. Such a procedure
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16 typically leads to the accumulation of electrons in the semiconductors provided that no oxygen or
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18 other electron acceptor is present in the solution. The electrons accumulated under blue light
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20 irradiation in the metallic oxo-clusters of MIL-125-NH₂ sample, produce a photochromic effect
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22 in which yellow MIL-125-NH₂ turns green.¹² This photochromic effect is caused by the
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24 formation of Ti(III) species that can be detected by EPR spectroscopy at room temperature as a
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26 broad signal at $g = 1.944$, as shown in Figure 6A. However, no signal at $g = 1.944$ is observed in
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28 COM50, suggesting that Ti (III) states are not present in the composite. This result confirms the
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30 existence of a charge transfer of photogenerated electrons from MIL-125-NH₂ to PFIK. On the
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32 other hand, upon irradiation of PFIK in the presence of the hole scavenger, numerous localized
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34 photoexcited electrons are accumulated under blue and green light excitation in its conduction
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36 band as suggested by the extraordinary intense and sharp signal at $g = 2.008$ (Figure 6B). The
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38 accumulation of the electrons is also accompanied by a significant photochromic effect leading
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40 to the darkening of the solid as illustrated in Figure S6. The darkening of PFIK is reversible, and
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42 the original state of the material is easily recovered upon its exposure to air, that leads to
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44 scavenging of the accumulated electrons by O₂ molecules. Remarkably, in the case of COM50,
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46 the intensity of the signal at $g = 2.008$ is significantly smaller compared to pure PFIK under both
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48 blue and green light illumination. The fact that COM50 does not accumulate electrons under the
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3 green light irradiation (which is only absorbed by PHIK solid in the MIL-125-NH₂/PHIK
4 composite) and under the blue light irradiation (which is absorbed by both components) probably
5 implies that TEOA is unable to remove holes from PHIK (and MIL-125-NH₂) efficiently, when
6 they are incorporated in the composite. As we discussed above, PHIK and MOF interact strongly
7 within the composite (electrostatically, by H-bonding), thus PHIK – TEOA interactions may be
8 significantly weakened. Moreover, sterically hindered access of TEOA to PHIK solid partially
9 surrounded by MIL-125-NH₂, may be further responsible for weakening of TEOA – PHIK
10 interactions.
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22 Depending on the relative mobilities of the photogenerated electrons and holes within MIL-
23 125-NH₂, two possible explanations are suggested to explain the enhanced photocatalytic activity
24 of the COM50 and the relatively small EPR signal at $g = 2.008$ in COM50 under blue light
25 irradiation, when MIL-125-NH₂ does transfer its photogenerated electrons to PHIK, as discussed
26 above. Some recent kinetics studies on other semiconductors, such as TiO₂ and ZnS, have shown
27 that the times of electron transfer at the surface of the photoexcited semiconductor at pH 7 are
28 typically smaller than the times of the hole loss.^{36–38} This can be considered as an indirect
29 evidence of the higher migration rates of the electrons compared to holes. However, since no
30 kinetic data are available for PHIK and MIL-125-NH₂ so far, below we describe both
31 possibilities.
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46 If the relative mobilities of the charge carriers have comparable rates, both electrons and
47 holes photogenerated in MIL-125-NH₂ are transferred to PHIK. The low EPR signal in COM50
48 may be then due to the fact that TEOA cannot remove holes from PHIK efficiently, e.g. due to
49 steric hindrance of weakened interactions. Thus, the recombination of charge carriers in PHIK
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If the mobility of the electrons in the MOF is higher than that of the holes, electrons will be transferred to PHIK, while holes will stay in MIL-125-NH₂, thus leading to a charge separation within the composite. The electrons transferred from the CB of MIL-125-NH₂ to the CB of PHIK may cause the overpopulation of the latter and induce recombination of the electrons with the holes present in the VB of PHIK, while the remaining holes in the VB of MOF are efficiently scavenged by TEOA. In this way, MIL-125-NH₂ partially suppresses the accumulation of charges in the composite. That would explain the fact why the intensities of the EPR signals of the COM50 upon 470 and 530 nm irradiation are almost the same, while the EPR signal of PHIK at 470-nm irradiation is significantly more intense compared to the one under 530 nm irradiation.

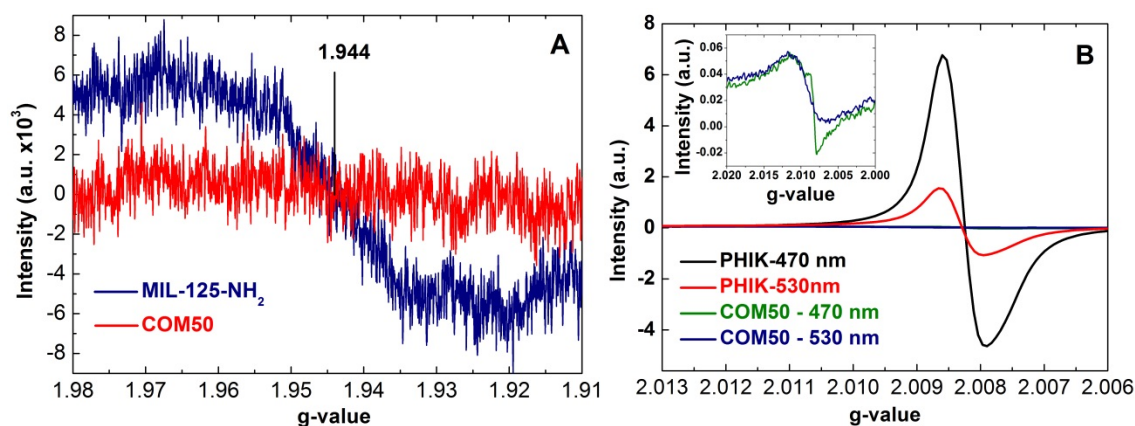


Figure 6. EPR spectra of MIL-125-NH₂ (blue) and COM50 (red) after irradiation in the anaerobic solution of TEOA (A). EPR spectra of the irradiated samples of PHIK (black for 470 nm and red for 530nm) and COM50 (green for 470 nm and blue for 530 nm) in an anaerobic solution of TEOA (B). EPR spectra in (A) and (B) are shown for different g-values to distinguish between the signals of the electrons accumulated at Ti(III) centers (A) and in PHIK(B).

Knowing the electronic structure of the composite, we can try to rationalize the increase of RhB degradation rates in the presence of semiconductor composites in comparison to the reaction rates observed using pure materials as catalysts. RhB can be oxidized by a sensitization

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3 mechanism³ or directly by photogenerated holes. The formation of HO[•] radicals is
4 thermodynamically forbidden, due to the fact that the valence band potentials of both materials
5 are less positive than required for hydroxyl radical formation ($E(\text{H}^+, \text{HO}^\bullet/\text{H}_2\text{O})$ vs. NHE) = 2.75
6 – 0.06 pH = 2.33 V at pH=7).³⁹ However, considering that the reduction potential of RhB⁺/RhB
7 couple (1.36V vs. NHE)⁴⁰ is less positive than the valence band potential of MIL-125-NH₂ (1.97
8 V vs. NHE), the oxidation of RhB by the holes photogenerated in MIL-125-NH₂ is also probable.
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10 The VB potential of PHIK (1.28V vs. NHE) is close to the redox potential of RhB, but is less
11 positive, thus oxidation of RhB in its ground state by the holes photogenerated in PHIK is
12 thermodynamically unfavorable. This may be responsible for a relatively low degradation rate of
13 RhB in the presence of PHIK as a photocatalyst. The activity of pure PHIK in RhB degradation
14 would be then explained either by degradation of excited dye molecules (RhB*) with O₂^{•-}
15 radicals or by direct oxidation of RhB* by the holes in PHIK (see below), or the combination of
16 both.
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34 As already mentioned above, depending on the relative mobilities of the charge carriers in the
35 MOF, we suggest two possible mechanisms to explain better photocatalytic performance of
36 COM50 compared to the pure materials. For the case, when both electrons and holes can be
37 simultaneously transferred from MIL-125-NH₂ to PHIK, a dye sensitization mechanism is
38 suggested as the main pathway for RhB degradation, since holes in MIL-125-NH₂ would not be
39 available to oxidize RhB. Indeed, the excitation energy of the dye is $E_{00} = 2.18\text{eV}$, and under the
40 experimental conditions used the absorbance of RhB is 0.7 at 465 nm (Figure S7). Thus, dye
41 oxidation *via* electron injection from the excited dye to PHIK and MIL-125-NH₂ is feasible, as
42 the reduction potential of RhB⁺/RhB* (-0.83 V vs. NHE)⁴¹ (Figure 7) is more negative than the
43 CB potentials of both materials. Namely, the excitation of the dye gives rise to RhB* that injects
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an electron in CB of MIL-125-NH₂, and the generated RhB⁺ cations undergo self-oxidation. The increase of the dye degradation rates in the presence of the composite is ascribed to the electron transfer from MIL-125-NH₂ to PHIK, which inhibits electron back-transfer (RhB⁺ could, in principle, retake the electrons from the CB of MIL-125-NH₂ and get reduced to RhB ground state).

In the case, when only electrons are transferred to PHIK, irradiation of the composite leads to a charge separation, and the excess of holes in the VB of MIL-125-NH₂ degrades additional RhB molecules. Thus, the increase of the reaction rate is observed.

Finally, dissolved O₂ removes electrons from PHIK, giving O₂^{•-} radicals, which can also oxidize the dye (see Figure 7). According to the obtained kinetic data, the optimal charge transfer from MIL-125-NH₂ to PHIK is reached at 1:1 components ratio.

The summary of the possible degradation pathways explaining, why the composite (COM50) degrades RhB faster than the pure materials, are depicted in Figure 7.

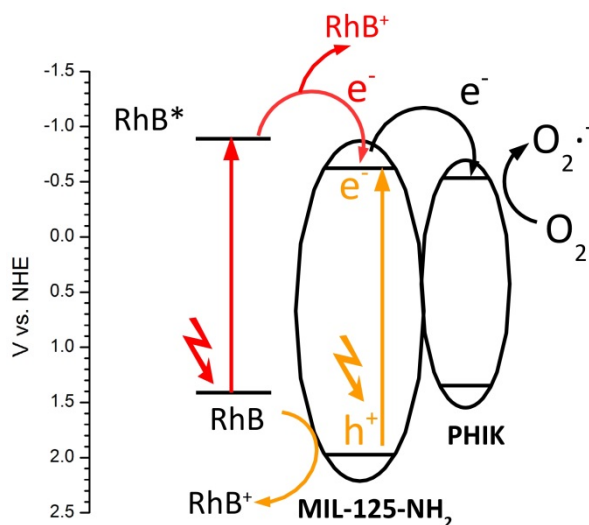


Figure 7. Possible explanations of the enhanced RhB degradation activity by the composite. The scheme depicts the energy levels of conduction and valence bands of MIL-125-NH₂ and PHIK. The pathway involving the oxidation of RhB by holes in MIL-125-NH₂ is depicted with orange

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3 arrows and letters. The mechanism involving the injection of electrons from RhB* towards MIL-
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5 125-NH₂ is shown with red arrows and letters. The shared steps of electron transfer from MIL-
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7 125-NH₂ to PHIK and scavenging of electrons by oxygen is depicted with black arrows and
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9 letters. The actual excitation of PHIK by visible light is not shown for clarity, since it is not
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11 considered as being responsible for the enhanced composite activity.
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15 CONCLUSIONS

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17 We have discovered a facile route to prepare composite photocatalysts composed of PHIK and
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19 MIL-125-NH₂. Potentially, this strategy can be applied to the synthesis of others PHIK/MOF
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21 composites, provided that the solids attract each other (at least partially), for example by given
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23 possibility of other MOFs to accommodate K⁺ ions in their micropores. The obtained type I
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25 heterostructure exhibits a synergistic effect for RhB degradation through an efficient electron
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27 transfer from the MOF to PHIK. Two explanations are suggested depending on the relative
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29 mobilities of the photogenerated charge carriers in MIL-125-NH₂. If the mobilities of the
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31 electrons and holes are similar, both types of the charge carriers are transferred from the MOF to
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33 PHIK. The enhancement of the activity is then due to the electron injection from the RhB* to
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35 MOF and the suppression of the electron transfer back to RhB* by their forward transfer to
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37 PHIK. If the mobility of the electrons in the MOF is higher than that of the holes, electrons
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39 (only) are transferred to PHIK where they can reduce O₂ molecules producing additional O₂^{•-}
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41 radicals able to oxidize RhB. At the same time, the excess of holes in the MIL-125-NH₂ is
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43 oxidizing additional RhB molecules. Overall, the increased reaction rates are due to an efficient
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45 charge separation within the composite.
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53 ASSOCIATED CONTENT

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3 **Supporting Information.** FTIR spectra of MIL-125-NH₂ before and after water treatment, SEM
4 image of the mixture of MIL-125-NH₂ and PHIK before water treatment, illustration of possible
5 chemical interactions at MOF/PHI interface, N₂ sorption isotherms and pore size distributions in
6 solids, Tauc plots for MIL-125-NH₂ and PHIK, Mott-Schottky plot for MIL-125-NH₂, images
7 depicting the photochromic effect in PHIK, absorption spectrum of a Rhodamine B solution. The
8 Supporting Information file is available free of charge at <http://pubs.acs.org>.
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39 Notes

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