

# Thermal evolution of nanoquasicrystalline $\text{Al}_{93}\text{Fe}_3\text{Cr}_2\text{Ti}_2$ alloy

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## Abstract

In this work, thermal evolution of bulk nanoquasicrystalline  $\text{Al}_{93}\text{Fe}_3\text{Cr}_2\text{Ti}_2$  material, obtained by consolidation of nanoquasicrystalline powder particles, is studied. The powder particles were obtained by gas atomization and the consolidation of the powders was achieved via warm extrusion [1,2]. The evolution of the microstructure during processing as well as with further thermal treatments was investigated. The consolidated material contains spherical particles of an icosahedral i-phase of around 200 nm in diameter. The quasicrystalline phase is detected even after 1 h at 500 °C, and afterwards, it evolves through a metastable  $\text{Al}_6\text{Fe}$  phase to the equilibrium phases.

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## 1. Introduction

Since the first synthesis of a quasicrystalline (QC) phase in a rapidly solidified AlMn alloy in 1984 and its subsequent interpretation, great efforts have been devoted to the synthesis, structure, physical and chemical properties of QC alloys. The quasicrystalline alloys are characterised by high hardness and brittleness owing to the difficulty of dislocations movement in the quasiperiodic lattice without long-range-periodicity. However, if nanoscale QC particles are dispersed homogeneously in a ductile matrix, the mixed phase alloy is expected to exhibit high strength combined with good ductility. Kimura et al. [3] observed that an icosahedral phase is formed in rapidly solidified Al–Fe–Cr–Ti alloys, which is promising for aerospace and automotive applications.

The aim of the present work is to study the thermal evolution of a nanoquasicrystalline Al alloy with nominal composition of  $\text{Al}_{93}\text{Fe}_3\text{Cr}_2\text{Ti}_2$  (at.%).

## 2. Experimental procedure

The  $\text{Al}_{93}\text{Fe}_3\text{Cr}_2\text{Ti}_2$  at.% powder was produced by helium atomization of pure Al and Ti, and master Al–Fe and Al–Cr alloys. The powder was sieved into four powder particle size fractions, <25, 25–50, 50–100, and 100–200  $\mu\text{m}$ . The chemical composition of the powder was confirmed by means of wet chemical analysis, as shown in Table 1.

Cylindrical bars were prepared by extrusion of the sieved powder particles with a horizontal extrusion press with a container diameter of 42 mm and maximum available pressure of about 1300 MPa. The extrusion temperature was 400 °C, the extrusion ratio was 10:1, and the ram speed was 0.3 mm/s. Prior to the extrusion, the encapsulated powder batches were submitted to heating at 400 °C for 15 min. The resulting bars were examined by means of X-ray diffraction using Cu K $\alpha$  radiation (XRD), and transmission electron microscopy (TEM). To study the thermal evolution, the bars were heat treated in air for 1 h at 400, 500 and 550 °C.

## 3. Results and discussion

Fig. 1 collects the XRD patterns, in the most representative range of  $2\theta$  between 35° and 50°, of the as-extruded and the heat treated bar processed with the 50–100  $\mu\text{m}$  powder particle size fraction. In the as-extruded state, signals of  $\alpha$ -Al and  $\text{Al}_3\text{Ti}$  are seen, together with peaks in  $2\theta$ : 22.9, 41.3, 43.5, 61.6 and 73.7, which can be indexed as an icosahedral phase (i-phase) using the indexation scheme for icosahedral quasiperiodic crystals [4]. The presence of the icosahedral i-phase in the  $\text{Al}_{93}\text{Fe}_3\text{Cr}_2\text{Ti}_2$  as-extruded bar was confirmed by TEM. Selected area diffraction patterns revealed the two-, three- and five-fold reflection spots as can be seen in Fig. 2. Quasicrystals in this alloy have the clear and characteristic spherical shape shown in Fig. 2, with a diameter of around 200 nm.

The XRD patterns of the bar treated for 1 h at 400 °C are very similar to those of the as-extruded bar, with  $\alpha$ -Al, QC and  $\text{Al}_3\text{Ti}$  signals. After annealing at 500 °C, the XRD patterns show  $\alpha$ -Al, a clear decrease of the amount of quasicrystalline phase, the presence of the  $\text{Al}_6\text{Fe}$  phase,  $\text{Al}_3\text{Ti}$  and small signals

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