

HaP/SBA-3 Nanostructured Composite to Remove Fluoride Effectively from Contaminated Water

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Abstract

Highly ordered pore mesoporous silica composites, like SBA-3 and hydroxyapatite (HaP) nanocrystals, characterized by X-ray diffraction (XRD), Fourier Transform Infrared Spectroscopy (FTIR), Scanning Electron Microscopy (SEM), Transmission Electron Microscopy (TEM) and textural properties, were successfully applied to remove fluoride from contaminated water. The proposed procedure to prepare HaP/SBA-3 was successful, which acts as supports to anchor the HaP crystals, in nanometer-scale (<2nm), with higher fluoride retention from contaminated water. The free OH- groups of HaP nanocrystals, within the host, facilitated the high-performance fluoride trapping. The fluoride retention activity was much higher than that of pure HaP and the composites HaP/SBA-15 and HaP/MCM-41.

Keywords: F- retention; Contaminated water; HaP/SBA-3; Nanocomposites

Introduction

Nanostructured materials, based on silica (MCM, SBA [1,2]) and ordered mesoporous carbon (OMC), such as CMK-1, CMK-3 [3] and Multi-Walled Carbon Nanotubes (MWCN) [4], because of their large surface area, pore volume and possibility to vary the pore size, offer a great opportunity for their application in various fields, from industrial processes to biomedical engineering, air and water pollution decontamination and energy reservoirs. Recently, we have developed a technique of preparation of monocrystalline hydroxyapatite (\mbox{HaP}) , and by the same procedure, but in the presence of the respective hosts, forming in situ composites such as HaP/MCM-41 and HaP/SBA- 15 composites [5]. Numerous approaches for fluoride removal from contaminated water, such as precipitation-coagulation [6], membrane technology [7,8], ion exchange [9] and adsorption [10,11]. Among the above methods, fluoride adsorption on adsorbents is attracting increasing attention due to its advantages of convenient design and operation, and low cost [12]. Different types of adsorbents, such as metal adsorbents [13], carbon adsorbents [14], carbon materials [15], natural materials [16], and biosorbents [17], have been used to remove fluoride from water. MOFs have also been used as new adsorbents for fluoride adsorption due to their surface functional groups and ordered atomic arrangement [18]. Calcium phosphate apatites are compounds of the formula Ca5 (PO4)3X, where X can be an F ion (fluorapatite, FaP), OH (hydroxyapatite, HaP), or a Cl- ion (chlorapatite). The ion is substituted by another ion of the same sign but of different charge. Neutrality is maintained by substitutions of ions with different charges or vacancies [19]. Studied the crystallization of HaP (hydroxyapatite) on polymers containing - C-N groups, from supersaturated solutions of HaP has been reported [20]. The growth of hydroxyapatite on silica gelsin the presence of organic additives was studied by Rivera-Muñoz et al. [21]. Laghzizil et al. [22] have correlated the results of HaP fluorination with the physicochemical properties of HaP. Recently, morphological modifications of hydroxyapatite using fatty acids as an organic modifier have been reported [23]. The goal of this work is the successful

development of a HaP/SBA-3 nanostructured composite material, by exploiting the small pore size of SBA-3 (3nm), with very large surface are, to embed nanonized HaP inside the pores, with size less than 2 nanometers, and apply the Hap/SBA-3 nanocomposite in the decontamination of fluorides from the polluted water.

Experimental

Materials synthesis

The procedure designed for the synthesis of SBA-3 was as follows [24,25]: the surfactant (cetyltrimethylammonium bromide), was added to water and HCl; afterwards, 3g of TEOS was also added, stirring to obtain a mixture with a molar composition of: TEOS:H2O:HCl:CTAB=1:150:9:0.15. After 60min, the white precipitate was filtered, washed and dried, and then the surfactant was extracted using ethanol at reflux for 4h, and subsequently calcined at 550 °C in air for 2h. Nanonized HaP was obtained using solutions of varying concentration 1-0.6M CaCl2 in 2H2O (a) and 1.8-2.30 M K2HPO4 (b) and then Benzalkonium Chloride 10% Solution (Alkylbenzyldimethylammonium chloride, C6H5CH2N(CH3)2RCl, R=C18H37), in double distilled water was added to prevent the growth of HaP crystals (ex-situ). Theresulting solution (pH=9) was stirred vigorously for 2h at 65 °C, obtaining nanometric HaP (4-5nm). For synthesizing HaP-Host, the same procedure was performed, with the modification that was added to the mixture of solutions (a) and (b), SBA-3 (50-70 % of HaP wt/ wt with respect to thehost), after the first hour of reaction which was used to prepare HaP ex situ. The suspensions of HaP/SBA-3 were stirred vigorously for 4h, at 60 °C, filtered, washed with triple distilled carbon dioxide-free water, and then dried at 100 °C for 4h. Subsequently, the composite was activated by heating at 500 °C in N2 flow for 5h, and then calcined at 550 °C at a heating rate of 5 °C/min from 100 °C for 4h. Hap/MCM-41 and HaP-SBA-15 [4], were also used in this study to compare the effectiveness of fluoride retention.

Fluoride retention essay

Experimental conditions: 2g of HaP in 100mL of NaF solution

with an initial concentration of 9 x10 $^{-2}$ M were used. The weight of the composite used was normalized to HaP. Fluoride ion solutions were prepared using a Teflon device with magnetic stirring, specially designed to bubble N2 to avoid CO2 contamination at 25 $^{\circ}$ C. The pH of the solutions was measured with a Mettler pH meter; the instrument was calibrated with buffers of pH=4 and 7.5. The concentration of F $^{-}$ ions was determined using an F $^{-}$ specific electrode, with a dynamic range between 1-300 ppm. In addition, trace amounts of F $^{-}$ were followed by FTIR. Fluoride retention capacity was evaluated in successive cycles at different times.

Characterization

Samples were characterized by means of powder X-ray diffraction in a X'Pert Pro PANalytical diffractometer equipped with a CuKa radiation source (k=0.15418nm). N2 adsorption/desorption isotherms were measured on ASAP 2020 equipment after degassing samples at 673K, determining textural properties such as surface area and pore volume; pore size distribution was estimated using Barrett, Joyner and Halenda (BJH) algorithm. TEM was recorded in a JEOL 2100F microscope operated with an accelerating voltage of 200kV (point resolution of 0.19nm). FTIR studies were performed in a JASCO 5300 Fourier transform infrared spectrometer (FTIR).

Results and Discussion

XRD, BET, SEM and HRTEM

The surface area of the nanonized HaP measured by BET (N2) method was $96\text{m}^2/\text{g}$. The surface area of SBA-3 was $1064\text{m}^2/\text{g}$, and $750\text{m}^2/\text{g}$ for Hap/SBA-3 respectively. The pore diameter of the host was: 3.1nm for SBA-3 and a_0 of 3.56 (lattice parameter (a_0 = 2d [100]/p3).). The compositeHaP/SBA-3 isotherms show a residual pore volume of 0.75-0.73mL per gram. We can distinguish the presence of three signals in the symmetry of the hexagonal network by XRD, higher signal at hkl [100] 3.31nm, and lower [110] and [200] typical of the SBA-3 structure, indicating a highly ordered pore system with a high porosity (Figure 1).

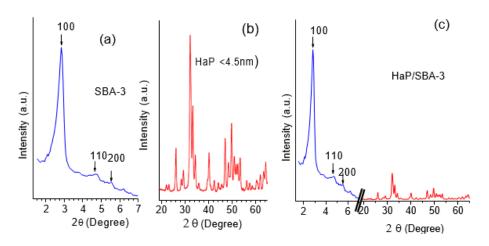


Figure 1: (a) SBA-3, (b) HaP, (c) HaP/SBA-3.

The XRD pattern of the nanonized HaP prepared ex-situ and the HaP/SBA-3 composite are illustrated in Figure 1b & 1c. Composite pattern diffraction peaks confirm a high crystallinity or long-range order structure in the nanostructured host SBA-3.

SEM studies (Figure 2a) show the size and shape of the HaP/SBA-3 composite, which indicate a goodmorphology of the crystals,

(a)

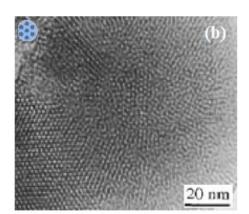


Figure 2: (a) SEM, (b) TEM of HaP/SBA-3.

host.

FTIR studies

The FTIR analysis of HaP/SBA-3 (Figure 3) reveals bands assigned to SBA-3 and HaP in the composite.

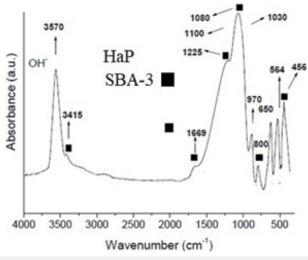


Figure 3: FTIR of HaP/SBA-3 nanocomposite.

The peak at 3570cm^{-1} is associated with the OH stretching mode of HaP and at 3415 with Si-OH stretching. The signals at 1080, 1227 and 1669cm^{-1} (asymmetric T-O, internal and external stretching respectively) are due to SBA-3. PO4 appeared as a very strong band at $\sim 1030 \text{cm}^{-1}$. P-O stretching is found at $\sim 970 \text{cm}^{-1}$. The clearly defined bands at 650 and 564cm^{-1} are ascribed to the O-P-O asymmetric deformation components. The band at 800cm^{-1} is caused by SI-O symmetric stretching of SBA-3. The typical band followed for F⁻ retention is that of OH⁻ at 3570cm^{-1} from HaP in the composite.

Flouride retention

To study the fluoride retention capacity of HaP/SBA-3 nanocomposite and its comparison with HaP, HaP/MCM-41 and HaP/SBA-15 two strategies were employed. a: All fresh materials were evaluated in the function of contact time with a solution containing fluorides and the decrease of fluorides from 20%, 40%, 60%, 80%, and 100%. Thus, Table 1 shows a high F⁻ retention capacity at short contact times with respect to the rest of the materials used, even with the mesoporous silica nanocomposites. The prepared nanosized HaP did not retain 100% F⁻ even after several hours in contact with the solution.

with an absence of other phases caused by HaP. The shape of the

HaP/SBA-3 crystals is practically spherulitic with a mean crystal size

in the range of 1.5 to 2µm. The HRTEM image illustrated in Figure

2b, detail the long-range hexagonal arrangement of HaP/SBA-3. Higher level reflections remain clearly detectable. Consequently,

the nanostructured HaP crystals is within the nanostructure of the

Table 1: Fluoride retention (%) at different contact time of various reservoirs.

Contact Time, h				
F- Retention %	HaP/SBA-3	HaP/MCM-41	HaP/SBA-15	НаР*
20	0.5	8	5	7
40	1	10	9	11
60	2	13	10	16
80	2.7	17	13	22
100	4	20	16	

^{*}Prepared ex-situ

Reusability

HaP/SBA-3 was taken and its reusability was studied, in different cycles with a duration of 2h and the percentage of retained F^- and the amount free OH^- bands of HaP, in % Absorbance Units, in the composite are reported. In Figure 4 it can be observed that after 5 cycles of 4h duration thenanocomposite still retains 100% of F^- from fresh solutions. Inter-cycle, the nanocomposite was filtered, washed only with bi-distilled water and dried at 100 °C, without

any reactivation. The results suggest that in the fresh HaP/SBA-3 composite, the OH $^{\cdot}$ of the HaP has remained intact because of high dispersion with the smaller size of the nanocrystals (less than 2nm) of HaP and the higher proportion of free OH $^{\cdot}$ per gram of the active material. The results also indicated that HaP/SBA-3 nanocomposite developed is not only active in terms of the time in contact with the solution to achieve 100% F^{\cdot} retention, but also that it remains active after several cycles, making ita highly attractive material for F^{\cdot} retention of contaminated water.

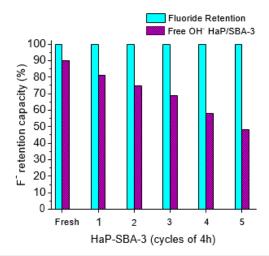


Figure 4: Capacity of F- retention and OH- free (%) of Hap/SBA-3 from contaminated water against the cycles of 4h.

Conclusion

SBA-3 nanostructured materials have good structural and textural properties, useful to acts as hosts for incorporating nanocrystals of hydroxyapatite, forming active HaP/SBA-3 composite. By HRTEM. FTIR, XRD, we found that the HaP nanocrystals are within the hosts, and not on the external surface, indicating good incorporation of nanocrystals in the host with sizes lower than 2 nm. The superior efficiency of HaP/SBA-3 relative to HaP/MCM-41 and HaP/SBA-15 could also be because SBA-3 has a higher abundance of Si-OH groups than the other reservoirs. Thus Ca2+ ofHaP became anchored onto the large quantity of siloxane species composed of adjacent silanol groups contained in SBA-3, enhancing the chance of more growth sites for the smaller HaP nanocrystals, because of the small pore size of SBA-3, than the other mesoporous silica supports studied (4 and 6 nm of MCM-41 and SBA-15 respectively).

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