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Normalization method for *J*-R curve determination using SENT specimens



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ABSTRACT

According to BS 8571:2014, standard *J*-R curves from SENT geometry can be determined from single or multiple specimens. The recommended methodologies for single specimen include unloading compliance or DCPD, although any validated technique can be used. The normalization method, which is not directly recommended in BSI standards, is an alternative. *J*-R curves through normalization technique based on *P*-CMOD records were experimentally determined using SENT specimens (0.40 < $a_0/W < 0.55$) of two structural steels, later compared with the ones measured by unloading compliance. Results indicated that the normalization method is a valid alternative for *J*-R curve determination using SENT specimens.

1. Introduction

Structural integrity assessment of cracked components and structures needs the fracture mechanics properties, which must be experimentally measured. In general, ductile materials require the application of elastic-plastic methodologies, such as CTOD (δ) or *J*-Integral (*J*), to characterize the fracture toughness. Normally, the crack driving force is measured as a function of stable crack extension, resulting in δ -R or *J*-R curves. The crack-tip constraint level due to a/W, specimen geometry and/or loading type strongly affects the R-curves. High constraint specimens limit plasticity, lead to expected conservative values of fracture toughness, and tend to present lower R-curves, whereas low-constraint specimens produce higher R-curves for the same material [1–3]. The widely known ASTM E1820-17a [4] and BS 7448-4:1997 [5] standards require high constraint three-point bending specimens, SE(B), or compact tension specimens, C(T) or DC(T), containing deep, through-the-thickness cracks with $a/W \ge 0.5$, resulting in lower R-curves. When the structural integrity of cracked pipes containing a circumferential flaw are analyzed, these curves and the associated fracture toughness values could be overly conservative.

In 2003, Nyhus et al. [6] showed that a pipe containing a circumferential flaw subjected to bending efforts presents the crack-tip in low constraint condition and proved that Single Edge Notched Tension – SENT – specimens represent this situation more realistically than SE(B) and C(T) specimens. Other investigations were performed on pin-loaded and clamped SENT specimens, and it was verified that the crack-tip constraint level was similar to an axial surface crack in a pipe with identical crack length and a circumferential surface crack with identical length in a pipeline under combined loading, respectively [7,8]. In this context, SENT specimens are capable of reproducing crack-tip constraint conditions similar to axial and circumferential cracks in a pipeline. Therefore, this geometry could be preferred for the determination of crack growth resistance curves of pipes in laboratory tests, as it seems to be more representative of the material behavior in real operational conditions.

Det Norske Veritas (DNV) [9] and Canada Centre for Mineral and Energy Technology (CanMet) [10-12] developed testing

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(1)

Nomenclature		т	constraint factor
		n	Ramberg-Osgood strain hardening exponent
а	crack length	Р	nominal force
a_0	initial crack length	$P_{\rm N}$	normalized force
a_{bi}	blunting corrected crack size at <i>i</i> th data point	R-curve	crack growth resistance curve
a_f	final crack length	SE(B)	Single edge-notched bending specimen
$\dot{b_0}$	initial remaining ligament	SENT	Single edge-notched tension specimen
В	specimen thickness	$U_{\rm p}$	plastic work calculated from the area under the P-
B_e	effective specimen thickness	•	CMOD record
$C_{\rm CMOD}$	specimen elastic compliance based on P-CMOD	W	specimen width
	record	δ	crack-tip opening displacement (CTOD)
C(T)	compact tension specimen	Δa	crack extension
DC(T)	disk-shaped compact specimens	η_p	non-dimensional parameter dependent of geo-
Ε	Young's modulus of elasticity		metry and loading type that relates the plastic
E'	effective Young's modulus of elasticity and for		contribution of J-integral to the work per unit
	plane strain conditions, the appropriate effective		uncracked ligament area in loading a cracked body
	elastic modulus is $E' = E/(1 - v^2)$	υ	Poisson's ratio
Н	distance between clamped grips	$\sigma_{ m UTS}$	ultimate tensile strength
J	<i>J</i> -Integral	$\sigma_{ m Y}$	effective yield strength
$J_{0.2\mathrm{BL}}$	resistance to crack extension at 0.2 mm crack ex-	$\sigma_{ m YS}$	0.2% offset yield strength
	tension offset to the blunting line of BSI standards		
$J_{0.2}$	resistance to crack extension at 0.2 mm crack ex-	Abbreviat	tions
	tension including blunting		
$J_{1.0}$	<i>J</i> -Integral value at $\Delta a = 1.0 \text{ mm}$	CMOD	crack mouth opening displacement
$J_{ m el}$	elastic component of J-Integral	HSLA	high strength low alloy steel
$J_{ m Ic}$	crack-extension resistance at 0.2 mm crack exten-	LLD, v	load line displacement
	sion offset to the blunting line of ASTM standards	NT	normalization technique
$J_{ m pl}$	plastic component of J-Integral	THS	tough high strength steel
K_{I}	elastic stress intensity factor	UC	unloading compliance method

procedures to measure fracture toughness using SENT specimens. Recently, the British Standard Institution (BSI) released the BS 8571:2014 standard [13] to determine fracture toughness for SENT specimens. All methods need that force (*P*), and crack mouth opening displacement (CMOD) are continuously measured during tests.

There are several single-specimen methodologies to determine J-R curves, normalization technique and elastic unloading compliance method are some of them and have been adopted in ASTM E1820 standard.

Initially, use of the normalization technique was proposed without the necessity of implementing specific instrumentation (i.e. fracture extensioneters). Thereby, *J*-R curves could be obtained through load vs. load-line displacement records (*P*-LLD) and initial and final physical crack lengths measured from fracture surfaces. This methodology is based on the load separation principle introduced by *Ernst et al.* [14]. Later, other researchers also investigated it [15–18]. They demonstrated that, in some cases, load can be separated in two multiplicative functions, one dependent on crack length – G(a/W) – and the other dependent on plastic displacement – $H(v_{pl}/W)$ – as follows:

$$P = G(a/W)H(v_{pl}/W).$$

According to this equation, a relationship between the three variables *P*, *a* and *v*, can be inferred during tests [19–22]. Dividing load by function *G*, *P* acquires the normalized form (P_N) and is defined in terms of plastic displacement or, in other words, in terms of the material plastic behavior:

$$P_N = \frac{P}{G(a/W)} = H(v_{pl}/W).$$
⁽²⁾

The function *H* is currently expressed as a function of four parameters (a, b, c and d) [23], see Eq. (6), and the appropriated constants for each specimen can be fitted through some data pairs of points of normalized load vs. normalized plastic displacement record $(P_N - v_{pl}/W)$. Then, using the fitted normalization function, the crack extension can be evaluated iteratively at all loading points for each specimen [24–26]. This methodology is standardized by ASTM (ASTM E1820 standard), which allows tests only in SE (B), C(T) and DC(T) specimens containing deep cracks (0.45 $\leq a/W \leq 0.7$).

In this investigation, the normalization technique was applied to SENT specimens ($0.40 < a_0/W < 0.55$) of two types of structural steels to obtain the corresponding *J*-R curves. Some modifications to the standard methodology were applied, as the use of *P*-CMOD record instead of *P*-LLD ones. The measured *J*-R curves were compared to those obtained from the same specimens by the unloading compliance technique.

2. Materials and methods

2.1. Materials

Two different structural steels were tested, a high strength low alloy steel (HSLA) and a tough high strength steel (named in this investigation as THS). For the first material, two specimens were tested and for the latter five specimens were used. To obtain the tensile mechanical properties for both materials, tensile tests according to ASTM E8/E8M-15a [27] were performed in cylindrical standardized specimens. Table 1 presents the tensile mechanical properties of the materials.

2.2. Fracture testing

Fracture toughness tests were performed on SENT specimen geometry according to BS 8571:2014 standard (C-type specimen, W/B = 0.5), in air, at room temperature and under displacement control. An Instron 1332 servo-hydraulic testing machine instrumented with \pm 250 kN load cell and a MTS 632.03F-31 fracture extensometer with 12 mm of nominal travel were used. The nominal dimensions of SENT specimens with integral knife edges can be seen in Fig. 1. All specimens were fatigue pre-cracked in a three-point bending device under a load ratio R = 0.1 at approximately 20 Hz. The maximum pre-cracking load for each specimen was calculated according to ASTM E1820-17a.

2.3. Crack length measurement

Crack lengths were evaluated through normalization technique and by elastic unloading compliance, the latter being used as a reference. A brief explanation of each technique follows, including the modifications employed.

2.3.1. Normalization method

Although use of the method was proposed on *P*-LLD records, in this investigation the normalization technique was applied on load vs. crack mouth opening displacement records (*P*-CMOD). Afterwards, on the *Discussion* section, the use of *P*-CMOD records will be justified. The application of the method was based on two standards procedures: Annex 15 of ASTM E1820-17a standard and BS8571:2014 standard, which were slightly modified. The main methodology including modifications is briefly described below.

Initially each P_i value up to, but not including, the maximum force P_{max} , of each experimental *P*-CMOD record is normalized using Eq. (3).

$$P_{Ni} = \frac{P_i}{WB\left(\frac{W-a_{bi}}{W}\right)^{\eta_p}},\tag{3}$$

where *i* refers to the *i*th loading point. In this work the value of η_p was calculated as given by Eq. (10) in BS 8571:2014 standard, and a_{bi} is the blunting corrected crack size at *i*th data point calculated by the following equation:

$$a_{bi} = a_0 + \frac{J_i}{2m\sigma_Y}.$$
(4)

The applicability of Eq. (4) to SENT specimens, as well as the use of an *m* factor in it, will be discussed later. For the corresponding CMOD, a normalization procedure is necessary to give a normalized plastic CMOD. The plastic component of CMOD is then normalized as:

$$CMOD'_{pl,i} = \frac{CMOD_{pl,i}}{W} = \frac{CMOD_i - P_i C_{CMOD,i}}{W},$$
(5)

where $\text{CMOD}_{\text{pl},i}$ is the plastic component of CMOD, CMOD_{*i*} is the total crack mouth opening displacement, and $C_{\text{CMOD},i}$ is the specimen elastic crack mouth opening displacement compliance based on the corrected blunted crack size a_{bi} .

The final P and CMOD data pair is normalized using the same equations as described above, with the final physical crack length measured from the fracture surface. Thereafter, the normalized data is plotted and a tangent line from the final load-displacement pair to the curve must be drawn. The data to the right of the tangency point shall be excluded for the fitting procedure. Normalized data with plastic normalized CMOD lower than 0.001 must be also excluded. Thus, through the following normalization function expressed by Eq. (6), remaining data can be fitted.

Table 1

Yield strength (σ_{YS}), ultimate tensile strength (σ_{UTS}) and the strain hardening exponent (*n*) of the studied structural steels (MPa).

Material	$\sigma_{ m YS}$	$\sigma_{\rm UTS}$	n
HSLA steel	396	500	8
THS steel	786	853	10



Fig. 1. SENT fracture specimen geometry. In detail, the profile of the notch.

$$P_{\rm N} = \frac{a + bCMOD'_{pl} + cCMOD'_{pl}}{d + CMOD'_{pl}},\tag{6}$$

where a, b, c and d are fitting constants. Finally, through the fitted normalization function, an iterative procedure is used adjusting crack lengths a_i to force P_{Ni} and CMOD'_{pli} defined by Eq. (6) to be equal to P_{Ni} and CMOD'_{pli} described by Eqs. (3) and (5), respectively. The a_i that best fit Eq. (6) to the experimental data is the instantaneous estimative of the physical crack length used for *J*-R curves determination.

2.3.2. Unloading compliance method

Crack lengths by unloading compliance were estimated through the compliance equation given by *Cravero and Ruggieri* [28] valid for clamped SENT specimen with H/W = 10 and integral knife edges:

$$\frac{a}{W} = 1.6485 - 9.1005u + 33.025u^2 - 78.467u^3 + 97.344u^4 - 47.227u^5,$$
(7)

where

$$u = \frac{1}{1 + \sqrt{B_e C_{\text{CMOD}} E'}}.$$
(8)

This equation is valid over the range $0.1 \le a/W \le 0.7$.

2.4. J-Integral calculations

J-Integral values were calculated according to BS 8571:2014 standard [13] as:

$$J = J_{el} + J_{pl}.$$



Fig. 2. Typical P-CMOD records for the tested materials in SENT geometry.

(9)

where

$$J_{el} = \frac{K_1^2 (1 - v^2)}{E},$$
(10)

 $K_{\rm I}$ is the stress intensity factor. The plastic component of J-Integral was calculated according to the following equation:

$$J_{pl} = \frac{\eta_p U_p}{B(W-a_0)},\tag{11}$$

With η_p calculated in terms of CMOD as given by Eq. (10) of this standard.

3. Results

3.1. P-CMOD records

Typical experimental load (*P*) vs. crack mouth opening displacement (CMOD) records for the tested materials are shown in Fig. 2. The unloading/reloading sequences corresponding to the unloading compliance method are clearly seen in these records.

3.2. Crack length measurements

The initial and final physical crack lengths were measured from the fracture surfaces using the 9-point average method as described in BS 7448-1:1991 standard [29]. Figs. 3 and 4 present typical fracture surfaces of the SENT specimens for HSLA and THS steel respectively. The fatigue pre-cracked region and the stable crack growth region are clearly seen, as well as the fatigue and the stable crack growth fronts, which were very regular in all cases.

Initial a_0/W ratio, final crack length, physical (Δa) and unloading compliance (Δa_{UC}) stable crack extensions, as well as the percentage difference between them for each specimen are shown in Table 2. None of the tests became invalid as a result of irregular fatigue crack front and/or crack growth.

3.3. Normalized functions

Each *P*-CMOD record was normalized using Eqs. (3)–(6). These data are plotted in Figs. 5 and 6, as well as the fitted curves (in red) obtained using only the dark points. In these figures, it is also possible to see the fitting coefficients for the normalization calibration function of these specific specimens.

3.4. Calculations of crack extension

Crack extensions by both methodologies were calculated as described in *Materials and Methods*. Figs. 7 and 8 present comparisons of crack extension values by normalization and by elastic unloading compliance for HSLA and THS steel specimens, respectively. To assist further analysis, the identity line was also plotted in both figures.

3.5. J-R curves

Figs. 9 and 10 show typical experimental $J - \Delta a$ pairs from each test by normalization technique (NT) and by elastic unloading compliance method (UC). The first one represents a typical *J*-R curve of the HSLA steel and the second one of the THS steel. Vertical auxiliary lines at 0.2 mm and at 20% of b_0 in Δa were additionally plotted in both figures. Blunting lines as proposed by BS 7448-4:1997 standard were also drawn (although the concept of blunting line is not used in the BS 8571:2014 standard).

Figs. 11 and 12 present all experimental $J - \Delta a$ pairs obtained through normalization technique and elastic unloading compliance method for HSLA and THS steel, respectively.



Fig. 3. Typical fracture surface of HSLA steel. FPC: fatigue pre-crack; SCG: stable crack growth; FR: final post-test brittle fracture.



Fig. 4. Typical fracture surface of THS steel. FPC: fatigue pre-crack; SCG: stable crack growth; FR: final post-test brittle fracture.

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	nitial (a_0	ι_0) and final (a	a _f) crack lengths,	, physical (Δa)	and unloading	g compliance ($\Delta a_{\rm UC}$)	stable crack	extensions of each tested s	pecimen
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Material	Specimen	<i>a</i> ₀ / <i>W</i> (mm)	<i>a</i> _f (mm)	$\Delta a \ (mm)$	$\Delta a_{\rm UC}$ (mm)	Diff. (%)
HSLA steel	01	0.525	6.71	1.49	1.45	2.68
	02	0.515	8.13	2.98	2.88	3.36
THS steel	01	0.412	6.34	2.16	2.10	2.78
	02	0.466	7.55	2.42	2.11	12.81
	03	0.427	6.60	2.34	2.27	2.99
	04	0.426	а	а	1.10	а
	05	0.415	6.47	2.31	2.39	3.46

^a After the test and prior to heat tinting this specimen was accidentally broken.



Fig. 5. Normalized load vs. normalized plastic crack mouth opening displacement for SENT 01 of HSLA steel.

3.6. Initiation fracture toughness determination

Strictly speaking, the standardized determination of initiation fracture toughness for SENT specimens can only be made through the BS 8571:2014 standard. In this standard the initiation value of fracture toughness of the material ($J_{0,2}$) is defined as the value of J at 0.2 mm of crack extension from the R-curve fit [13]. For comparison, similar toughness values were also determined through



Fig. 6. Normalized load vs. normalized plastic crack mouth opening displacement for SENT 02 of THS steel.



Fig. 7. Comparison of crack extensions obtained through normalization technique and elastic unloading compliance method for specimens of HSLA steel.

procedures described in ASTM E1820–17a (J_{Ic}) and BS 7448-4:1997 ($J_{0.2BL}$) standards, which include the concept of blunting line. As initiation fracture toughness is defined at the interception of *J*-R curves with an auxiliary line, a best fit procedure is necessary to obtain the fracture toughness values. In this investigation, the equation used to fit the qualified data was the one described in BS 8571:2014. Tables 3 and 4 present toughness values of both materials obtained by normalization technique and unloading



Fig. 8. Comparison of crack extensions obtained through normalization technique and elastic unloading compliance method for specimens of THS steel.



Fig. 9. Experimental $J - \Delta a$ pairs obtained by NT and UC for SENT 01 of HSLA steel.



Fig. 10. Experimental $J - \Delta a$ pairs obtained by NT and UC for SENT 02 of THS steel.



Fig. 11. Comparison of experimental $J - \Delta a$ pairs obtained through unloading compliance method (UC) and normalization technique (NT) for the SENT specimens of HSLA steel.



Fig. 12. Comparison of experimental $J - \Delta a$ pairs obtained through unloading compliance method (UC) and normalization technique (NT) for the SENT specimens of THS steel.

Table 3

Fractures toughness of the HSLA steel in SENT specimens calculated through normalization technique (NT) and elastic unloading compliance method (UC) according to BS 8571:2014 ($J_{0.2}$), ASTM E1820-17a (J_{Ic}) and BS 7448-4:1997 ($J_{0.2BL}$) [kJ/m²].

Specimen	BS 8571:2014		ASTM E1820-17a		BS 7448:1997	
	NT	UC	NT	UC	NT	UC
HSLA 01 HSLA 02 Mean	283 308 296 ± 18	298 292 295 ± 04	669 707 688 ± 27	697 803 750 ± 75	460 549 505 ± 63	494 528 511 ± 24

Table 4

Fractures toughness of the THS steel in SENT specimens calculated through normalization technique and elastic unloading compliance method according to BS 8571:2014 ($J_{0,2}$), ASTM E1820-17a (J_{1c}) and BS 7448-4:1997 ($J_{0,2BL}$) [kJ/m²].

Specimen	BS 8571:2014		ASTM E1820-17a		BS 7448:1997	BS 7448:1997	
	NT	UC	NT	UC	NT	UC	
THS 01	576	482	1326	1188	941	761	
THS 02	524	505	1266	1238	824	800	
THS 03	620	562	1306	1355	943	928	
THS 04	603	492	1280	1354	914	817	
THS 05	662	491	1393	1314	1026	878	
Mean	597 ± 51	506 ± 32	$1314~\pm~50$	$1290~\pm~74$	930 ± 72	837 ± 66	

compliance according to the three mentioned standards: BS 8571:2014, ASTM E1820-17a, and BS 7448-4:1997.

4. Discussion

4.1. Normalization method applied to SENT specimens

The applicability of normalization method in SENT geometry makes this methodology very attractive because of the lower sensitivity of elastic unloading compliance of SENT geometries compared to the SE(B) and C(T) ones, still lower for shallow cracks.



Fig. 13. a/W vs. E'BC_{CMOD} for SE(B), C(T), pin-loaded SENT, and clamped SENT geometries.

This fact can be better observed in Fig. 13, which compares the solutions for a/W vs. E'BC from CMOD for SE(B), C(T), pin-loaded SENT, and clamped SENT geometries.

Besides the mechanical properties, environment and temperature, the material response during fracture tests depends on the triaxiality level at the crack tip, which means that it is dependent on specimen geometry, crack size (a/W), loading mode, etc. Different configurations can lead to different crack tip constraint levels and plastic deformation zones/patterns [30]. High-constraint



Fig. 14. J-R curves by normalization method calculated using *m* factor from Shen & Tyson, from ASTM E1820-17a standard, and by elastic compliance.

specimens, such as deeply cracked SE(B), C(T) and DC(T) specimens, limit plasticity to a region near the crack tip, at the remaining ligament. On the other hand, low-constraint specimens, as SENT and shallow cracked SE(B), are capable of deforming plastically more than high-constraint specimens and they do not restrict the plasticity to a limited region at the crack tip. When SENT specimens are loaded, they promote different plastic deformation patterns which include plastic deformation within and outside the remaining ligament. However, the use of CMOD allows us to accurately measure the displacement associated with the fracture process avoiding other plastic deformation contributions. Besides that, the η_{pl} factor that relates the plastic contribution of *J*-Integral to the work per



Fig. 15. Correction procedure for the initial oscillations in *J*-R curves measured through the normalization method. (a) Original *J*-R curve - common initial oscillations. (b) Corrected *J*-R curve.

unit uncracked ligament area must be derived from CMOD record as reported in some researches [30–32]. In this context, and as described earlier, the normalization method was applied based on *P*-CMOD records.

The crack length estimation procedure of the normalization method needs a blunting corrected crack size (a_{bi}). The concept of blunting line is not considered in the BS 8571:2014 standard and was introduced as [33]:

$$J = 2m\sigma_{\rm Y}\Delta a,\tag{12}$$

where *m* is a constraint factor dependent on *a/W*, the stress state, and σ_{YS}/σ_{UTS} among others. For bending geometries, ASTM E1820-17a standard uses m = 1 in the blunting line slope and in the normalization method procedure [4], while BS 7448-4:1997 uses m = 3.75/2. It is important to note that there is no agreement in the literature about the value of *m* [33,34]. Instead, Shen & Tyson proposed the following equations to estimate the *m* value for clamped SENT geometries [35]:

$$m = A_1 \frac{a}{W} + A_2,\tag{13}$$

where

$$\begin{cases} A_1 = -0.1293 + 0.1152n - 0.00986n^2 + 0.000263n^3 \\ A_2 = 3.08670 - 0.2970n + 0.01940n^2 - 0.000427n^3 \end{cases}$$
(14)

and *n* is the Ramberg-Osgood strain hardening exponent. In our case, Eq. (13) led to a blunting line slope that is very close to that used in BS 7448-4:1997 standard. Additionally, as can be seen in Fig. 14, the obtained normalization *J*-R curve (open circles) was very close to the unloading compliance one (dark squares). In opposition, the normalization *J*-R curve obtained by using ASTM E1820-17a blunting line (m = 1) (dark circles) did not match the unloading compliance well, as can also be seen in the same figure.

Another feature observed during the application of this technique is the initial oscillation of the *J*-R curve. Fig. 15a shows these oscillations for a SENT specimen of THS steel. These oscillations were observed in all *J*-R curves measured by the normalization method. They occur because of the crack length estimation through this methodology is very sensitive to the quality of the four parameters function (Eq. (6)), even more so for small amounts of plastic displacements [36]. A suggested procedure to correct this problem is to use the blunting line representing the crack tip blunting and shift all displaced $J - \Delta a$ data to this line. Fig. 15b presents the corrected *J*-R curve according to this procedure. All measured *J*-R curves presented in this investigation (Figs. 9–12) were corrected using this procedure. It is important to note that these oscillations occurred only in the initial part of the curves for small values of plastic displacement and out of the qualification region for the *J*-R curve determination (see Fig. 15a).

4.2. Crack extension

As can be seen in Figs. 7 and 8, the crack extension values obtained through normalization and elastic unloading compliance are in good agreement, falling close to the identity line. As can be seen in Table 2, unloading compliance underestimated final crack extension in some cases, but always within the limit imposed by ASTM E1820-17a and BSI 7448-4 standards [4,5].

4.3. Initiation fracture toughness

Normalization and unloading compliance methodologies led to similar initiation fracture toughness values. The maximum difference observed was approximately 20%. Tables 3 and 4 present these values. By comparing the crack initiation toughness mean values we can observe that the values obtained by the normalization method for the HSLA steel were in general smaller than values obtained by the elastic unloading compliance method. Besides, the opposite behavior was verified for the THS steel.

The experimental *J*-R curves obtained by both methodologies are in close agreement, as can be seen in Figs. 11 and 12. The crack propagation parameter $J_{1.0}$ presented similar values in all cases, as well as the corresponding slopes, $(dJ/da|_{1.0})$. This is another evidence that the normalization method using *P*-CMOD record for SENT specimens can be applied to measure *J*-R curves and to determine the initiation fracture toughness of the material.

Although only *J*-R curves were considered in this work, the proposed methodology with small modifications could be applied also to determination of CTOD-R curves, that are also considered in the BS 8571:2014 standard.

5. Concluding remarks

The normalization method proved to be a good alternative for *J*-R curve determination using SENT specimens with $0.4 < a_0/W < 0.55$ of two different structural steels. Crack length estimations and *J*-R curves by normalization and unloading compliance were in close agreement.

The following aspects need to be taken into account when the normalization method is applied to SENT specimens:

- The use of P-CMOD records allows us to accurately measure the displacement associated with the fracture process, avoiding other plastic deformation contributions;
- A blunting line slope based on the Shen & Tyson expression was introduced. This value was close to that given by BS 7448-4:1997.

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Appendix A. Supplementary material

Supplementary data associated with this article can be found, in the online version, at http://dx.doi.org/10.1016/j.engfracmech. 2018.06.033.

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