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Theoretical study of new potential semiconductor surfaces performance for dye sensitized solar cell usage: $\text{TiO}_2\text{-B}$ (001), (100) and $\text{H}_2\text{Ti}_3\text{O}_7$ (100)



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ABSTRACT

Hydrogen titanate ($\text{H}_2\text{Ti}_3\text{O}_7$) and $\text{TiO}_2\text{-B}$ polymorph are potential surfaces identified experimentally in the last years, which need to be analyzed. To study their performance as surfaces for dye sensitized solar cells (DSSC), a set of dye adsorption configurations were evaluated on them, as model dye the small and organic catechol molecule was used. We have calculated adsorption geometry, energy, electronic transfer from dye to semiconductor adsorbent and frontier orbitals by means of density functional theory (DFT). Results show that vacancy-like defected $\text{H}_2\text{Ti}_3\text{O}_7$ (100) and $\text{TiO}_2\text{-B}$ (100) surfaces present favorable adsorption energies. Finally, an adequate energy level alignment make both surfaces prone to be adequate for direct electron transfer upon excitation, from catechol to the conduction band of the semiconductors, with bands located in the Visible region of the electromagnetic spectrum. Additionally, the band structure alignment indicates an increase in the open circuit voltage, in reference to I_2/I_3^- redox pair potential. All these characteristics make hydrogen titanate ($\text{H}_2\text{Ti}_3\text{O}_7$) and $\text{TiO}_2\text{-B}$ polymorph promising for DSSC applications.

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1. Introduction

The $\text{TiO}_2\text{-B}$ polymorph was synthesized by Marchand et al. [1], its chemical structure presents bronze pattern Na_xTiO_2 , presenting continuous channels and lower density than the others TiO_2 polymorphs, i.e. anatase, rutile, brookite. It can be obtained easily via soft chemical routes by proton exchange followed by dehydration of titanates in layers, as Feist et al. have reported [2,3]. Different nanostructural morphologies of $\text{TiO}_2\text{-B}$ have been synthesized as nanoparticles [4], nanowires [5–7], nanoribbons [8] and nanotubes [9]. However, regarding its performance in DSSC, as far as we know, only few studies have been reported [4,8,10] with comparable results.

Only some theoretical studies have been carried out regarding $\text{TiO}_2\text{-B}$ polymorph. Structural, electronic and vibrational properties have been calculated by first principles DFT [11]. Vittadini et al.

have studied the reconstruction and structural stability of $\text{TiO}_2\text{-B}$ surfaces [12]. Adsorption of water on $\text{TiO}_2\text{-B}$ (001) and (100) has been reported by Liu et al. [13]. The presence of $\text{TiO}_2\text{-B}$ during synthesis of titania nanostructures, showing a stability similar to that for anatase at nanoscale level, together with few results reported about this polymorph, motivate us to study these surfaces.

Nanostructured TiO_2 can be found in several forms, like sheets, rods and scrolls and it can interconvert into each other changing the chemical setting, temperature and/or pressure [14–16]. These transformations can create nanostructures with different size, shape, structure, and also variable compositions [17]. These compositions can vary from stoichiometric TiO_2 to metal-ion and proton-exchanged titanates [14]. One interesting reversible interconversion is between $\text{Na}_2\text{Ti}_3\text{O}_7$ nanowire and $\text{H}_2\text{Ti}_3\text{O}_7$ nanoscrolls resulting from changes of pH in aqueous dispersions [18,19]. TiO_2 in phase anatase and protonated titanates $\text{H}_2\text{Ti}_3\text{O}_7$ -type are directly related structurally and rearrangement from one specie to other is possible changing local water partial pressure [15]. In literature, we can find few experimental researches regarding $\text{H}_2\text{Ti}_3\text{O}_7$ structure and properties [20,21], as well as theoretical reports [22,23]. To the best of our knowledge, this compound is gaining interest and it has not been studied yet as a surface.

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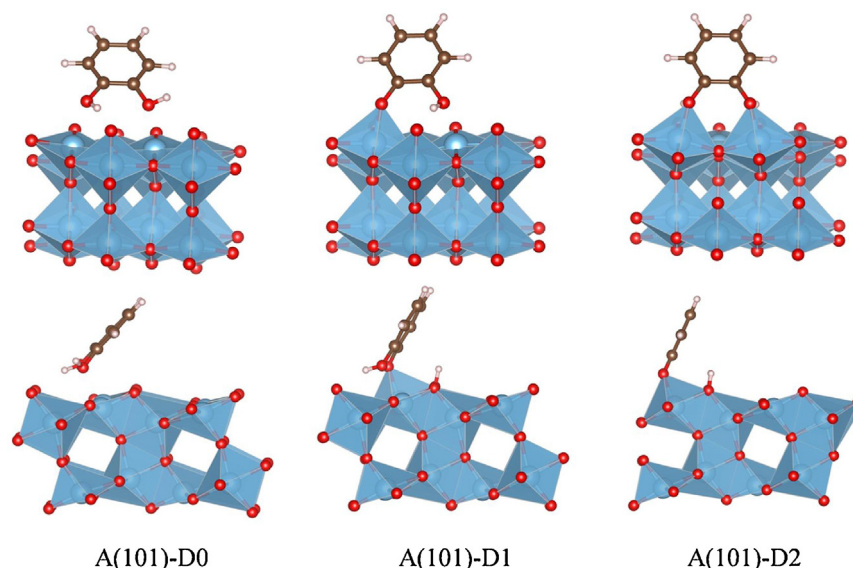


Fig. 1. Catechol adsorption on anatase (101), molecular adsorption D0, mono-dentate adsorption D1 and bi-dentate adsorption D2. Front and side views.

The aim of this work is to study the performance of these new TiO_2 -B and $\text{H}_2\text{Ti}_3\text{O}_7$ surfaces in dye sensitized solar cells (DSSC) [24], given that TiO_2 polymorphs are nontoxic, cheap and photostable semiconductors. These DSSCs are promising material for harvesting solar energy, very attractive due to its low cost, high efficiency, simple fabrication process and for being environmentally friendly [25–28].

As testing dye, we have used catechol, which is the smallest molecular sensitizer and presents two hydroxyl groups for attaching to the semiconductor surface, these features make it an excellent model to study the surfaces performance in DSSC [29–34]. As it was made previously for several authors for anatase and rutile polymorphs [35–46].

2. Computational methods

In all calculations, Density Functional Theory (DFT) [47] implemented in the Vienna Ab-initio Simulation Package (VASP) was used [48]. The projector augmented wave (PAW) method [49,50] was employed to account for the electron–ion core interaction, using the Perdew–Burke–Ernzerhof (PBE) as the generalized gradient approximation (GGA) [51] for the exchange–correlation term. In order to treat the strong on-site Coulomb interaction of localized electrons in transition metal oxides, which is not properly described by the GGA (and LDA) approach and affects the electronic structure, the GGA + U method proposed by Dudarev et al. [52] was used. Setting a U-parameter of 4 eV for 3d level of Ti in TiO_2 -B polymorph and a U-parameter of 5 eV in hydrogen titanate [53], the multiple occupation of d-orbitals is penalized, and thus, the band gap width takes values closer than the experimental ones, attenuating the underestimation of this width, as well as the electron delocalization.

The space groups for TiO_2 -B and $\text{H}_2\text{Ti}_3\text{O}_7$ are C2/m [11,21] and surfaces were modeled by cutting the unit cells in the specific directions to obtain the different faces, and then a vacuum region along the non-periodical direction was applied, of about 10 Å which is large enough to avoid significant interactions between different images of the slab. We have selected the following surfaces for catechol adsorption: $\text{H}_2\text{Ti}_3\text{O}_7$ (100), TiO_2 -B (001) and (100). During adsorption, a fully slab relaxation was allowed for the first two system, for TiO_2 -B (100) the superficial two layers were allowed to move and the bottom three were kept frozen in the bare

surface positions. The Brillouin-zone was sampled using $4 \times 4 \times 1$ Monkhorst-Pack K-point mesh [54]. A cutoff energy of 400 eV was used for the plane-wave basis set. Model-dye adsorption energy was calculated using the following equation:

$$E_{\text{ads}} = -(E_{\text{catechol/surface}} - E_{\text{catechol}} - E_{\text{surface}})$$

being $E_{\text{catechol/surface}}$ the energy of the system dye/semiconductor, E_{catechol} stands for the energy of isolated dye molecule calculated in a cubic box of $10 \times 10 \times 10 \text{ Å}^3$ and E_{surface} represents the energy of the bare – non-interacting – semiconductor surface. A positive E_{ads} implies a favorable energy process, while a negative implies the opposite, in order to compare our results with those reported in literature.

Density of states curves (DOS) were calculated to evaluate the electron transfer from the dye valence band to the semiconductor conduction band in the different systems under study, and they were plotted to check the band gap width estimation. HOMO and LUMO frontier orbitals were drawn for the more promising adsorption systems. In order to understand and evaluate appropriately how these new surfaces perform as adsorbents for DSSCs, our results are compared with those for the well-known catechol/anatase (101) system [37,41,46].

Finally, for the most stable geometries we performed further vibrational analysis by getting phonons at Γ – point. In all the cases, we obtained real frequencies demonstrating that all the geometries correspond to local minima, see supporting information for Table of vibrational frequencies.

3. Results and discussion

3.1. Dye-semiconductor systems

3.1.1. Catechol on anatase (101)

Although this system is widely studied, we have carried out this calculation as a fine tuning of the mechanics employed. We have taken into account three different possible adsorption sites, namely D0, D1 and D2 which correspond to molecular adsorption, mono-dentate adsorption with dissociation of one H atom and bi-dentate adsorption with dissociation of both H atoms from catechol hydroxyl groups, respectively, as it can be seen in Fig. 1.

It was found that the more probable adsorption site for catechol, energetically is D2, followed by D1 and at last D0, with adsorption

Table 1

$\text{Ti}_{\text{surf}}-\text{O}_{\text{dye}}$ and $\text{O}_{\text{surf}}-\text{H}_{\text{dye}}$ bond lengths in Å, catechol adsorption energy and energy gap from valence band to conduction band in eV for the three sites studied in catechol/anatase (101) system.

	Ti–O (Å)	O–H (Å)	E ads (eV)	VB–CB (eV)
D0	2.51 2.35		0.431	2.911
D1	1.94 2.33	0.972	0.810	2.101
D2	1.91 1.91	0.972 0.974	0.903	1.604

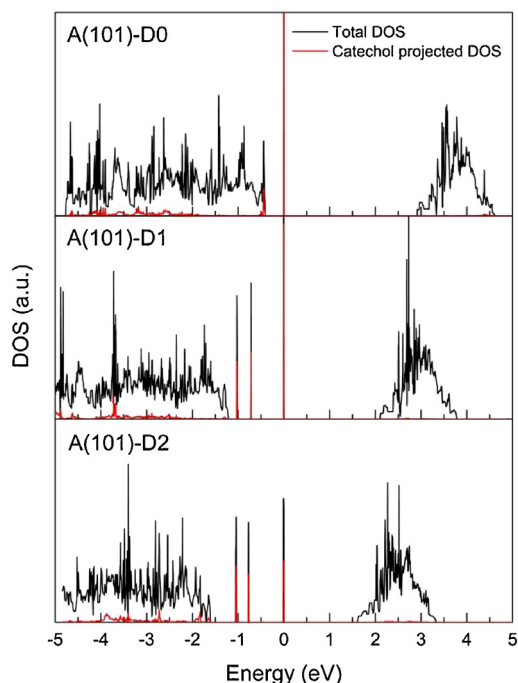


Fig. 2. Projected density of states for total system and catechol, for D0, D1 and D2 adsorption configuration on anatase (101).

energies of 0.903, 0.810 and 0.431 eV, respectively. This tendency agrees with data reported in literature [37,41,46], as seen in Table 1. The smaller electronic gap from valence band to conduction band also corresponds to D2 adsorption configuration, having a value of 1.604 eV. These two aspects, energetic and electronic, make the D2 configuration the more appropriate for dye adsorption, which is the same conclusion reached in previous cited literature.

In Fig. 2, the DOS curves are shown, the catechol projected DOS is colored in red. In the three cases, valence band energy states correspond to model-dye and they are located at the Fermi level. Also, it can be seen in Fig. 2 that while the molecule is being adsorbed, the VB–CB gap is shortened, from 2.911 eV to 1.604 eV, being best located in the visible region of the electromagnetic spectrum.

3.1.2. Catechol on $\text{H}_2\text{Ti}_3\text{O}_7$ (100)

The same adsorption configurations are studied for hydrogen titanate surface, i.e. D0, D1, D2 and a new one called C, which consists in the bonding of two oxygen atoms from catechol with one titanium surface atom (chelating bond) and dissociation of both H atoms from hydroxyl groups of the adsorbate. These systems are energetically unfavorable with the exception of the one corresponding to molecular adsorption, D0. However, when DOS curves are calculated and plotted we found that the energy gap from VB to CB is high, around 2.536 eV, making this system not feasible for DSSC usage.

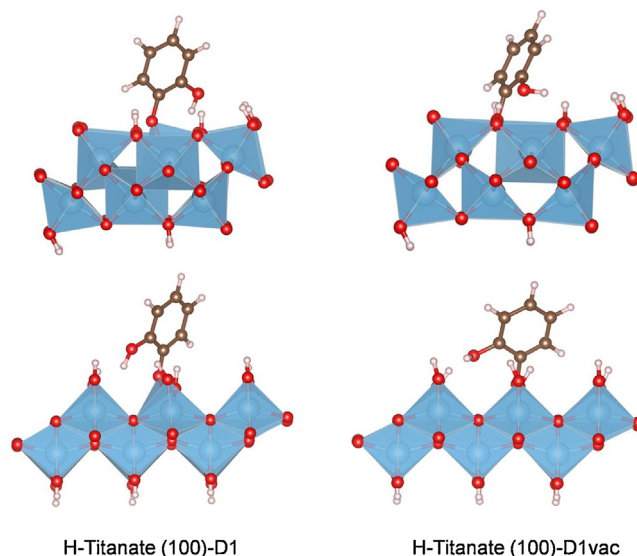


Fig. 3. Catechol adsorption on $\text{H}_2\text{Ti}_3\text{O}_7$ (100), D1 mono-dentate adsorption and D1-vac mono-dentate adsorption with dye oxygen atom located in surface O-vacancy. Front and side views.

Table 2

$\text{Ti}_{\text{surf}}-\text{O}_{\text{dye}}$ and $\text{O}_{\text{surf}}-\text{H}_{\text{dye}}$ bond lengths in Å, catechol adsorption energy and energy gap from valence band to conduction band in eV for the two sites studied in catechol/ $\text{H}_2\text{Ti}_3\text{O}_7$ (100) system.

	Ti–O (Å)	O–H (Å)	E ads (eV)	VB–CB (eV)
D1	1.95	0.974	1.096	1.624
D1-vac	2.06 2.11	0.972	1.750	2.354

In order to turn the surface more reactive, an oxygen vacancy is generated [20] and two catechol adsorption modes are studied. D1, which corresponds to the mono-dentate adsorption, a H atom from the dye is dissociated and the other one is kept bonded to the adsorbate. D1-vac corresponds to the adsorption of catechol forcing its oxygen atom to set in the superficial O-vacancy, bonding equally to two Ti atoms from the semiconductor surface. These configurations are shown in Fig. 3, and data regarding their bond lengths, adsorption energies and VB–CB energy gaps are summarized in Table 2.

Giving that these two configurations are optimum for catechol adsorption, it is interesting to calculate their respective DOS curves. In Fig. 4 the DOS curves are shown (the catechol projected DOS is in red). Despite energetically D1-vac configuration is more probable to happen, D1 configuration has the shorter energetic gap from VB to CB, better suitable for DSSC applications [55,56].

3.1.3. Catechol on $\text{TiO}_2\text{-B}$ (001)

We have considered four dye adsorption configurations for $\text{TiO}_2\text{-B}$ polymorph (001) surface, namely, D0, D2, C1 and C2. The last two configurations are described by the bonding of two dye-oxygen atoms to one superficial titanium atom, the difference between C1 and C2 is where the dissociated H atoms are located on the surface, as it is shown in Fig. 5. In Table 3 structural information as well as energetic data are listed for each configuration. None of these are energetically probable to occur.

In Fig. 6 we can see that the gap from VB to CB for configuration D2 is 0.919 eV, and for C1 is 1.384 eV, which is better than the 1.604 eV found for anatase (101) and better than 1.624 eV found for hydrogen titanate, very convenient for DSSC applications since it involves both the visible and the infrared region of the electromag-

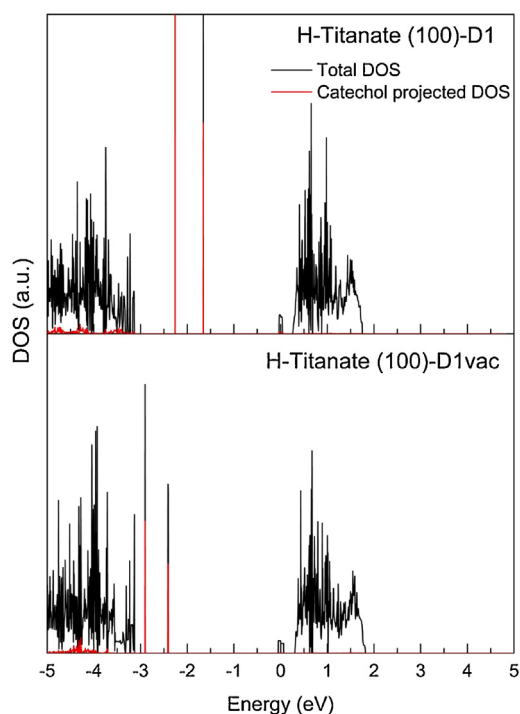


Fig. 4. Total and projected DOS curves for catechol adsorption on $\text{H}_2\text{Ti}_3\text{O}_7$ surface containing an oxygen vacancy.

netic spectrum [55,56]. However, its dye adsorption energy makes this system not appropriate for our aim. The reason for this behavior is the low surface energy of facet (001) of TiO_2 -B polymorph, which clearly reduces its chemical reactivity, which compromises the adsorption of the dye.

3.1.4. Catechol on TiO_2 -B (100)

For the case of catechol adsorbed on TiO_2 -B polymorph (100) surface, we have studied the same four configurations that in the previous case (D0, D2, C1 and C2), as seen in Fig. 7.

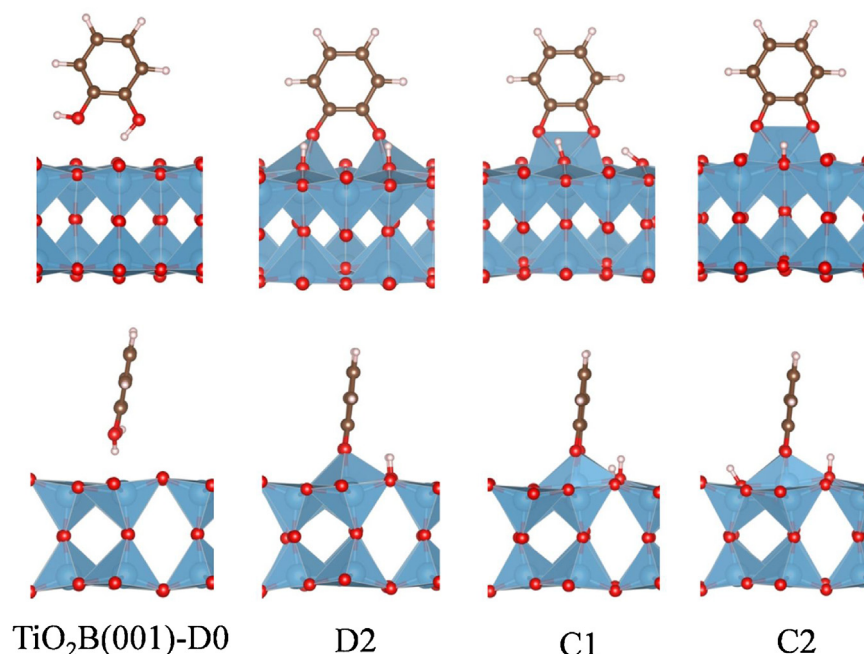


Fig. 5. Catechol adsorption on TiO_2 -B (001) surface, molecular adsorption D0, bi-dentate adsorption D2 and chelating adsorption C1 and C2. Front and side zoomed views.

Table 3

$\text{Ti}_{\text{surf}}\text{—O}_{\text{dye}}$ and $\text{O}_{\text{surf}}\text{—H}_{\text{dye}}$ bond lengths in Å, catechol adsorption energy and energy gap from valence band to conduction band in eV for the four sites studied in catechol/ TiO_2 -B (001) system.

	Ti—O (Å)	O—H (Å)	E ads (eV)	VB—CB (eV)
D0	2.73 2.89		−0.111	1.197
D2	1.92 1.92	0.976 0.976	−0.489	0.919
C1	2.01 2.08	0.978 0.989	−1.488	1.384
C2	2.02 2.02	0.980 0.979	−1.693	1.571

Table 4

$\text{Ti}_{\text{surf}}\text{—O}_{\text{dye}}$ and $\text{O}_{\text{surf}}\text{—H}_{\text{dye}}$ bond lengths in Å, catechol adsorption energy and energy gap from valence band to conduction band in eV for the four sites studied in catechol/ TiO_2 -B (100) system.

	Ti—O (Å)	O—H (Å)	E ads (eV)	VB—CB (eV)
D0	2.464 2.226		0.395	2.931
D2	1.879 1.879	0.980 0.980	1.068	2.938
C1	1.942 1.972	0.978 0.979	0.285	0.637
C2	1.961 1.956	0.978 0.972	0.804	2.918

In Table 4, structural and energetic information is listed for the four adsorption configurations. All cases are probable to occur energetically, following the sequence D2, C2, D0 and C1 with adsorption energies of 1.068, 0.804, 0.395 and 0.285 eV, respectively. Nevertheless, only the C1 case has a convenient energy gap between VB and CB for DSSC applications, but it worths to notice that its adsorption energy is minor than that on anatase (101) and hydrogen titanate containing an O vacancy. The VB—CB gaps have been plotted in Fig. 8, total DOS and catechol projected DOS are shown.

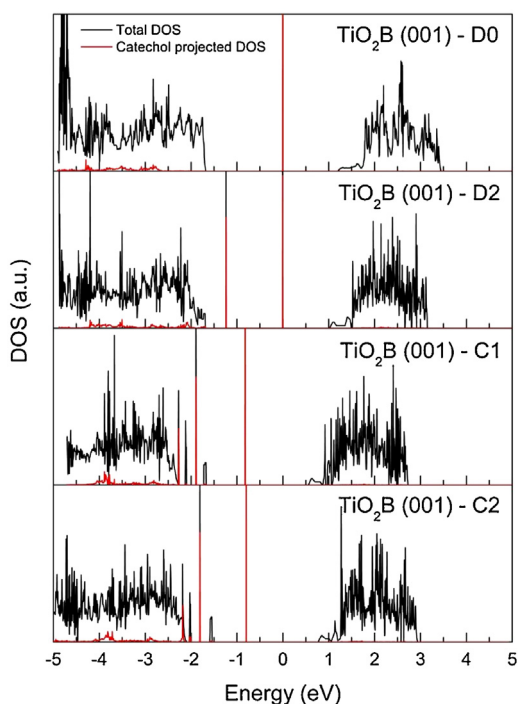


Fig. 6. Total and projected DOS curves for D0, D2, C1 and C2 catechol adsorption on $\text{TiO}_2\text{-B}$ (001).

3.2. Frontier orbitals

For an effective sensitization performance, it is very important that the HOMO (highest occupied molecular orbital) of the dye is in the semiconductor gap and the LUMO (lowest unoccupied molecular orbital) is above the conduction band minimum. Thus, the electron-hole separation is achieved in the hybrid interface by electron photo-injection from the dye molecule HOMO to the surface conduction band, or by first excitation of an electron from HOMO to LUMO of the dye follow by electron injection in the surface conduction band. Naturally, catechol on anatase (101) presents this behavior, being classified as DSSC type-II system, in which a direct charge-transfer occurs from dye's HOMO to the semiconductor's CB, that should present a band absorption in the visible region of

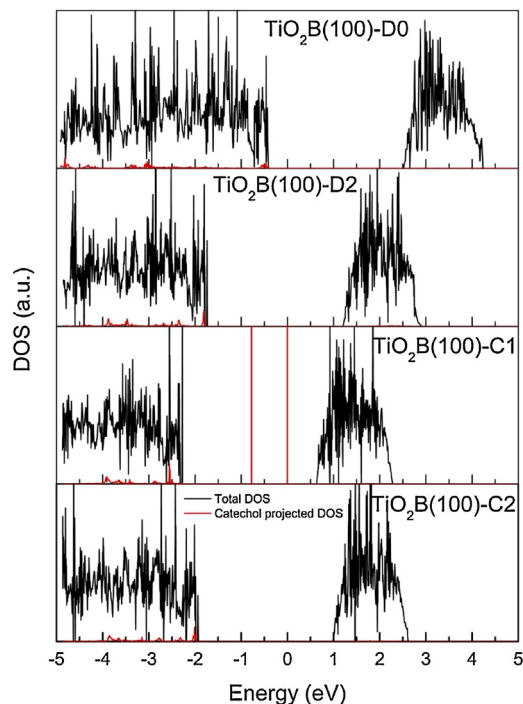


Fig. 8. Total DOS and catechol projected DOS for D0, D2, C1 and C2 adsorption on $\text{TiO}_2\text{-B}$ (100).

the electromagnetic spectrum [55,56]. From our studied systems, catechol on hydrogen titanate containing an oxygen vacancy and catechol at C1 location on $\text{TiO}_2\text{-B}$ (100) are the appropriate ones for DSSC applications, due to their favorable adsorption energies and because they fulfill the frontier orbitals condition. In Figs. 9 and 10, as a comparison, front and side view, for HOMO and LUMO, respectively, corresponding to systems dye/anatase (101), dye/hydrogen titanate (100) and dye/ $\text{TiO}_2\text{-B}$ (100)-C1 are plotted. It is clear that HOMO is composed from π -states of the dye, and LUMO for Ti-d states of the surface, as expected.

It can be observed in Fig. 9 that HOMO frontier orbital of dye/H-titanate-D1 involves the catechol oxygen atom and one superficial titanium atom, which would make more probable the electron transfer from adsorbate to the semiconductor surface. In D1-vac

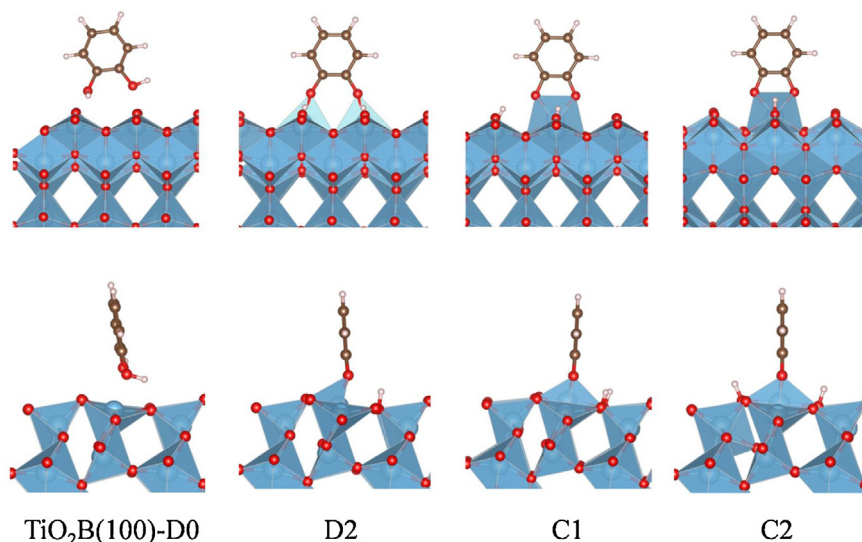


Fig. 7. Catechol adsorption on $\text{TiO}_2\text{-B}$ (100) surface, molecular adsorption D0, bi-dentate adsorption D2 and chelating adsorption C1 and C2. Front and side zoomed views.

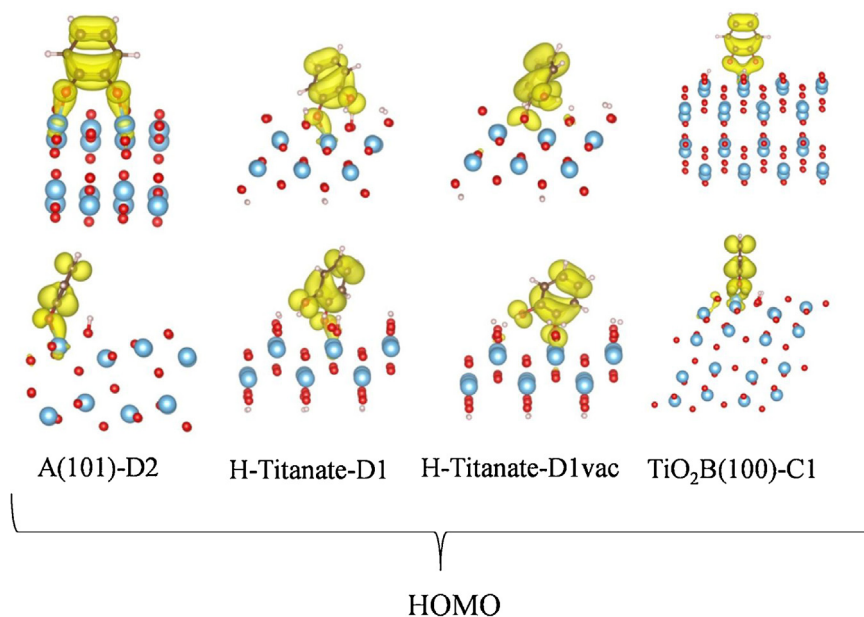


Fig. 9. HOMO frontier orbitals for optimum dye/semiconductor systems. Front and side view are shown. Some bonds were omitted for sake of clarity.

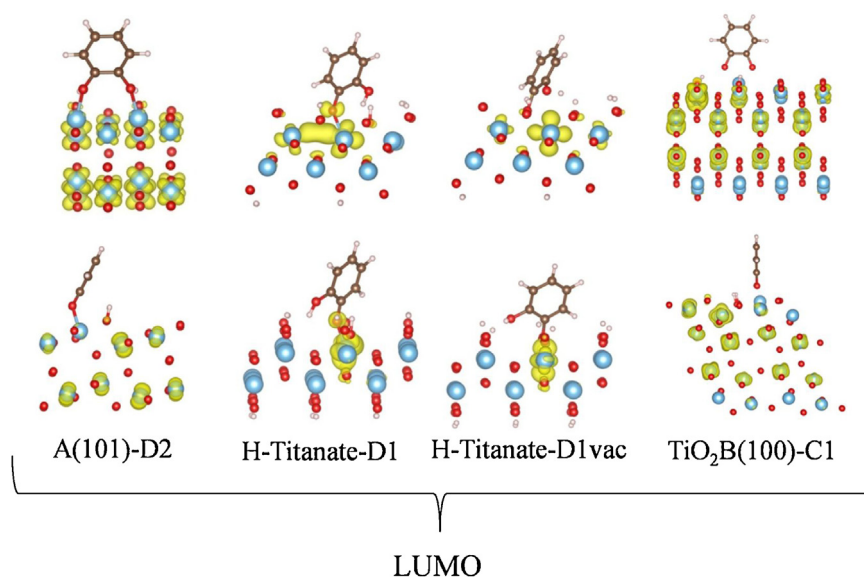


Fig. 10. LUMO frontier orbitals for optimum dye/semiconductor systems. Front and side view are shown. Some bonds were omitted for sake of clarity.

adsorption case, HOMO frontier orbital involves not only the dye oxygen p-states but also titanium d-states from the surface, given that dye oxygen atom occupies now the semiconductor vacancy, turning this surface more stable. However, this oxygen atom coming from the dye can be interpreted now as part of the surface, thus making more difficult the electron transfer process. In the case of dye/TiO₂B(100)-C1 the HOMO orbital overlap is small making the system less appropriate for DSSC applications in comparison with hydrogen titanate surfaces.

Finally, and utilizing the work function information it is possible to perform a first approach to the band alignment of all the reported surfaces as can be observed in Fig. 11. Additionally, taking as a reference the I₂/I₃[−] redox pair potential E_{redox} , it is possible to estimate the open circuit potential V_{OC} according to [57]:

$$V_{\text{OC}} = \frac{1}{q_e} \left(E_{\text{CB}} + k_B T \ln \left(\frac{n_c}{N_{\text{CB}}} \right) - E_{\text{redox}} \right)$$

Where n_c and N_{CB} corresponds to the electrons in the conduction band and the total number of electronic states in the CB, respectively and k_B is the Boltzmann constant. Taking the $\frac{n_c}{N_{\text{CB}}}$ ratio as constant for all the surfaces [57], it is possible to draw the scheme presented in Fig. 11. According to this, the V_{OC} ordering is the following: $V_{\text{OC-TNT(100)}} > V_{\text{OC-TiO}_2(\text{B})(001)} > V_{\text{OC-Anatase(101)}} > V_{\text{OC-TiO}_2(\text{B})(100)}$.

4. Conclusions

The performance of surfaces of new compounds found experimentally –TiO₂-B and H₂Ti₃O₇– in DSSC was analyzed by testing their adsorbant properties via dye model catechol adsorption. Structural and electronic properties were calculated using DFT theory, implementing Hubbard U term. It was found that not all these semiconductor surfaces are prone to catechol adsorption and electron transfer from dye to surface. TiO₂-B (001) is not energetically optimum for dye adsorption due to its low (001) surface energy,

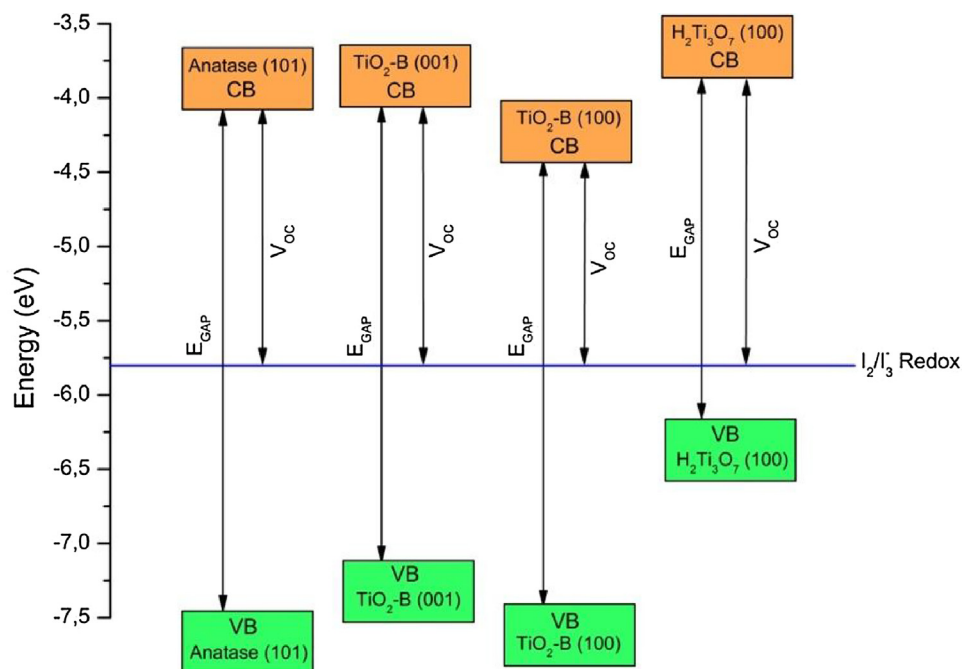


Fig. 11. Band alignment for all the studied polymorphs and titanates, utilizing as reference the I_2/I_3^- redox pair potential.

although the gap from valence band to conduction band is appropriate for electron transfer and well positioned in the visible region of the electromagnetic spectrum. Surface TiO_2 -B (100) behaves in the opposite way, showing favorable adsorption energies for D0, D2, C1 and C2 configurations. In particular, TiO_2 -B (100)-C1 configuration is energetically possible and involves a considerable small gap for electron transfer. $H_2Ti_3O_7$ (100) surface, containing an oxygen vacancy is the more suitable one for DSSC applications due to its optimum dye adsorption energy and small gap between valence band and conduction band. Frontier orbitals plots endorse this conclusion showing the dye/surface orbitals overlapping in HOMO allowing an easy electron transfer. Finally, the band alignment study allows as to conclude that the H-titanate surfaces should develop the higher open circuit voltage in comparison to the rest of the studied TiO_2 polymorphs.

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