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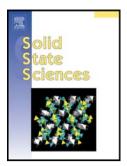
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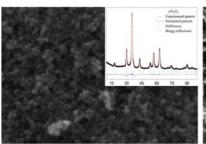
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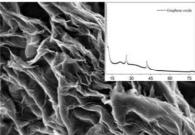
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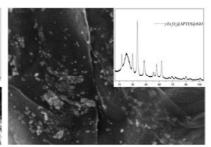
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# SYNTHESIS OF SILANIZED MAGHEMITE NANOPARTICLES ONTO REDUCED GRAPHENE SHEETS COMPOSITES

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#### **Abstract**

Novel  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES@rGO composites are successfully synthesized by using graphene oxide and silanized maghemite nanoparticles. Graphene oxide and maghemite were obtained by Hummers and Massart methods, respectively. The silanization process was done to functionalize maghemite surface with a controllable quantity of amino groups. Then, by adding aqueous graphene oxide suspension, the bonding between graphene oxide and silanized maghemite nanoparticles was done in refluxing conditions. Afterwards, chemical reduced graphene oxide reaction was realized by addition of hydrazine solution. The characterization of  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES@rGO composites was studied by X-ray Diffraction, Fourier Transformed Infrared Spectroscopy, thermogravimetric analysis and scanning electron microscopy.

KEYWORDS: Maghemite, APTES, Graphene, surface density, composite.

#### 1- Introduction

The development of nanocomposite systems formed by nanostructured materials with their own chemical and physical properties is a growing research area in nanotechnology because of their potential technological applications in biosensors[1], supercapacitors[2], and other nanodevices[3]. Recently, graphene has been the focus of much research given their unique structural, electrical, thermal and mechanical properties [4-10]. Graphene shows magnetoresistive properties linked to its organic nature, that can be used as a base for a magnetoresistive device [11,12]. Granular materials composed by magnetic nanoparticles dispersed in a matrix displays giant magnetoresistance, which is originated in a different physical property, the orientation of the magnetic supermoments of the superparamagnetic nanoparticles [13, 14]. Therefore, an interesting system can be prepared by attaching magnetic nanoparticles to molecular organic materials, particularly graphene layers [15, 17]. Maghemite nanoparticles (γ-Fe<sub>2</sub>O<sub>3</sub>) are commonly used due to their intrinsic stability against oxidation, and large specific magnetization [18,19]. This letter reports preparation and characterization of composites formed by γ-Fe<sub>2</sub>O<sub>3</sub> nanoparticles, functionalized with aminopropyltriethoxysilane (APTES). This process is used to anchor nanoparticles to other molecular structures [20-23]. After this functionalization, the obtained structure  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES is attached to reduced graphene oxide layers (rGO), named now γ-Fe<sub>2</sub>O<sub>3</sub>@APTES@rGO. In a former article, Il T. Kim et. al. obtained a new compound where graphene was decorated with γ-Fe<sub>2</sub>O<sub>3</sub> nanoparticles [17]. However, the main difference between the compound synthesized by Il T. Kim et. al. and the

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composite reported here is the presence of the propylsilane linking graphene with  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles. Such differences result in a quantitative control of the association between nanoparticles and graphene (the components of composite). The synthesis process is resumed as 1) silanizing  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> with 3-aminopropyltriethoxysilane (APTES); 2) coupling between  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES compounds and graphene oxide (GO); and 3) chemical reduction of graphene oxide preserving the  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> silane-coating and its bonding to the graphene sheet. The experimental results show that chemically reduced graphene oxide sheets (rGO) are linked to  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES nanoparticles.

#### 2- Materials and methods

Preparation of maghemite nanoparticles:  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles has been synthesized by a modified Massart procedure [24]. An aqueous mixture of FeCl<sub>3</sub> and Fe(NH<sub>4</sub>)<sub>2</sub>(SO<sub>4</sub>)<sub>2</sub> (molar ratio 2:1) is co-precipitated with concentrated NH<sub>4</sub>OH. After one hour of reaction, the precipitate is filtered and washed with distilled water until pH 7. The obtained product, composed by magnetite nanoparticles (Fe<sub>3</sub>O<sub>4</sub>) is dispersed into an acidic solution of HNO<sub>3</sub> and Fe(NO<sub>3</sub>)<sub>3</sub> (pH 2 and ionic strength of 14.6 mM) at 80°C. The oxidation of Fe<sub>3</sub>O<sub>4</sub> to  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> occurred under vigorous stirring for 4 h. Finally, the supernatant is removed and the solid resultant powder is washed with acetone.

Silanized maghemite nanoparticles: 250 mg of  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles were dispersed in ethanol (200 ml), then dispersion was mixed with APTES in a molar ratio of 1:1 ( $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES<sup>1</sup>) and 1:10 ( $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES<sup>10</sup>). The solution was kept for seven hours in refluxing conditions. Solid products ( $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES) were centrifuged and dried at 80°C.

*Graphene oxide synthesis:* GO was synthesized using Hummers method [25]. Synthetic graphite (2 g,100 mesh) was added to concentrated  $H_2SO_4$  (46 ml) and stirred at 0°C for 24 h. Afterwards,  $NaNO_3$  (200 mg) was added to the mixture while keeping temperature and stirring. For oxidation,  $KMnO_4$  (6 g) was added slowly keeping temperature below 35°C. Then, water (86 ml) was dropped at a rate of 1.5 ml/min. Finally, water (280 ml) and  $H_2O_2$  (20 ml, 30%) were added to terminate the reaction. The obtained solid was centrifuged, washed with HCl (10 %) and water.

Composite formation: GO (250 mg) was ultrasonicated in NaOH solution (500 ml, pH 10) and stirred for 5 h. Then,  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES (50 mg) and dicyclohexylcarbodiimide (2 mg) were added into GO solution and kept in refluxing condition for 24 h. Obtained solid ( $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES@GO) was magnetically separated and dried. The reduction process occurs when  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES@GO (50 mg) was suspended in an aqueous hydrazine solution (10 ml, 35%) under stirring and heating (90°C) for 2 h [26]. The final solid ( $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES@rGO) was washed with water to remove hydrazine residual and dried at 60°C.

#### Characterization

X-ray diffraction (XRD) patterns were collected at room temperature with a Rigaku diffractometer (Cu  $K_{\alpha}$  radiation,  $\lambda$ =1.54056 Å) in 2 $\theta$  range from 10 $^{\circ}$  to 110 $^{\circ}$  with scan range of 0.12 $^{\circ}$ /min. Fourier transformed infrared

(FT-IR) spectra were recorder with an IRAffinity-1 spectrophotometer, in 4000-500 cm<sup>-1</sup> range, using standard KBr pellet technique. Thermal stability of compounds was investigated using a TGA-50 Thermogravimetric Analyzer under static air atmosphere at a heating rate of 10 °C min<sup>-1</sup>. Scanning electron microscopy (SEM) images were obtained with a Carl Zeiss Supra 40 field emission SEM microscope. Magnetization was measured in a Quantum Design MPMS SQUID magnetometer.

#### 3- Results and discussion

Figure 1, (a) and (b), show XRD patterns of  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES@GO and  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES@rGO composites, respectively. As can be seen, the presence of  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> nanocrystalline was confirmed by the characteristic reflections associated with maghemite phase. In Figure 1(a), the peak around 11° is related to GO, but, after hydrazine reduction, this peak vanishes and a poorly ordered graphene signal near to 22° appears, Figure 1(b). The peak of 22° is associated to (002) rGO [27]. In  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES@GO XRD patterns, there are three peaks around 26.4°, 44.6°, 77.4° that are assigned to (002), (101), (110) reflections of graphite (PDF 89-8487), respectively. The presence of these peaks can refer to an incomplete graphene conversion. Nevertheless, the existence of graphite impurities does not affect the conversion of GO to rGO.

Magnetization vs applied field measurements of silanized nanoparticles (not shown here) gave a saturation magnetization between 49 and 61 emu/g, compatible with accepted values for maghemite [28, 29], below the bulk saturation value (about 76 emu/g [30])

In order to confirm silanized  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>, FT-IR and TGA measurements were performed. In Figure 2a are shown the characteristics absorption bands of the aminopropyl groups as well as the stretching vibration of Fe–O and Si–O bonds. So, it is infered that APTES is successfully bonded to  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>. In TGA results, two weight-loss stages were observed (Figure 2b). The first one was associated with the loss of H<sub>2</sub>O, while the second one is the result from aminopropyl group decomposition. The temperature ranges and weight loss percentages are summarized in Figure 2b inset. Based in second weight-loss stage, it is possible to determinate APTES surface density. For this, molecular formula of  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES was considered as H<sub>2</sub>N-((CH<sub>2</sub>)<sub>3</sub>)-SiO<sub>3</sub>-(Fe<sub>2</sub>O<sub>3</sub>)<sub>n</sub>, with *n* as the coated particle ratio. As a result, APTES surface density is 3.69 (n~8) and 4.93 (n~6) molecules/nm<sup>2</sup> for  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES<sup>1</sup> and  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES<sup>10</sup>, respectively. Additionally, in Figure 2b is observed that when APTES surface density increase, the decomposition temperature range increases. This could be associated with an increase in thermal stability [31,32].

Moreover, when analyzing the FT-IR spectra associated with  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES@rGO, Figure 2c, it can be observed the absence of N-H stretching and NH<sub>2</sub> bending mode of amino group as an effect of covalent bonds between  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES and GO sheets [33]. In TGA measurements of two composites (Figure 2d) are observed three weight-loss stages that are associated with the loss of H<sub>2</sub>O, residual functional groups and rGO sheets, respectively [34, 35]. As can be seen, the rGO loss is higher for the composite which has the largest

number of amino groups in the coating. This effect confirms that there is a direct dependence between the coated characteristics and the amounts of graphene sheets in the synthesized composite.

The morphology and structural features of  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES, GO sheets and composite were studied by using SEM. Figure 3a shows a SEM image of  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES in which is clear that discrete coated-nanoparticles have a spherical shape. As shown in this image,  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES are small with a size of about 10 nm, which can become a spacer to prevent a posterior restacking of individual graphene sheets. From the image of GO, Figure 3b, it can be seen that graphene layer are exfoliated to a large extent, with a wrinkled structure that provides a large rough surface as scaffold for further modification. After reaction between  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES and GO, and the posterior reduction process with hydrazine, the coated-nanoparticles are completely distributed on rGO sheets (Figure 3c) indicating the composite formation. In this representative SEM image of the composite, the graphene sheets exhibit a slightly wrinkled surface and the coated-nanoparticles appear as bright dots. Many  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES spheres are firmly anchored on both sides of the wrinkled graphene sheets. The graphene layers might contribute to hinder coated-nanoparticles aggregation.

#### 4- Conclusion

In summary, we have successfully synthesized a novel nanocomposite based on  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES nanoparticles and chemically reduced graphene oxide sheets. The results from XRD, FT-IR and TGA clearly indicate that coated-nanoparticles and graphene sheets are firmly tethered by covalent bonding. However, the graphene oxide quantity present in the composite can be influenced as a function on amount of amino groups that constitute the  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> coating.

#### 5- Acknowledgements

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Figure 1. XRD patterns for (a)  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES@GO and (b)  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES@rGO composites. In figure 1(a), the diffraction peaks can be assigned to  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>, GO(\*) and graphite(•). The inset in (a) shows the structural analysis of  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> from which was obtained the particle size (12.83 nm). In figure 1(b), the diffraction peaks can be associated to  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> and rGO( $\checkmark$ ). The inset in (b) shows XRD of graphite, GO and rGO.

Figure 2. FT-IR and TGA measurements for  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES and  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>@APTES@rGO compounds. FT-IR signal assignation, in Figure (a) and (c), was performed with data previously reported in the literature [31-33]. The absorption band associated with the Fe–O–Si bond cannot be seen because it appears near to 580 cm<sup>-1</sup> and it would overlap with the Fe–O vibrations of  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>.

Figure 3. SEM images of (a) γ-Fe<sub>2</sub>O<sub>3</sub>@ APTES, (b) GO sheets and (c) γ-Fe<sub>2</sub>O<sub>3</sub>@ APTES@rGO composite.

Figure 1.

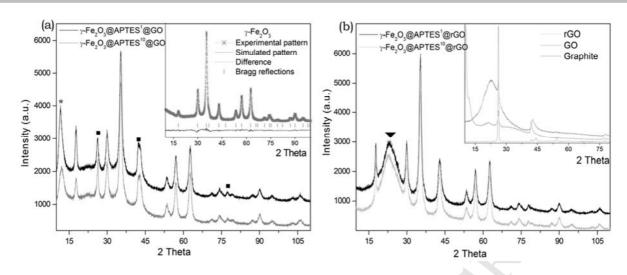


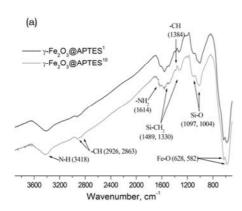
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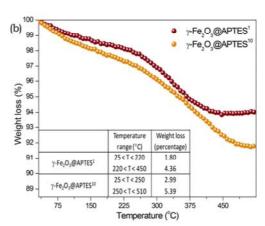


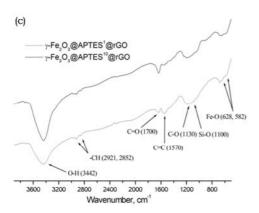
Figure 3.

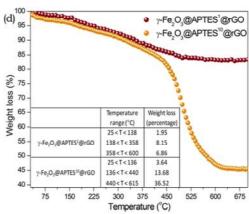


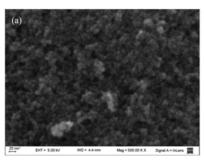


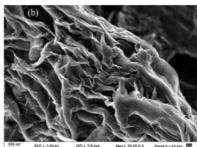


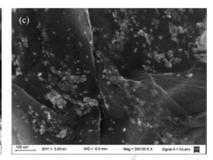














# • Novel γ-Fe<sub>2</sub>O<sub>3</sub>@APTES@rGO composites are synthesized.

- Silanization process of maghemite nanoparticles was done in controllable quantity.
- XRD, FT-IR, TGA and SEM techniques were used to confirm the composite existence.

