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Title: Electrodeposited Cu₂ZnSnS₄ thin films

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Abstract: Cu₂ZnSnS₄ (CZTS) thin films have been prepared using Electrochemical Atomic Layer Deposition (EC-ALD) and also by one-step conventional constant potential electrodeposition. Optimal deposition conditions were investigated using cyclic voltammetry (CV). Then, based on CVs results, CZTS films were grown employing EC-ALD deposition cycles using the sequence Au/S/Cu/S/Zn/S/Sn/S to form the desired quaternary compound. In parallel, conventional one-step electrodeposition was carried out at -0.85 V vs. Ag/AgCl over 1 hour. A thermal treatment in sulfur vapor was also investigated in an attempt to optimize the stoichiometry. The crystal structure of the films was characterized by XRD and micro Raman spectroscopy, while the morphology, thickness, topography and elemental composition were investigated using FIB-SEM and EDS.



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Dear Sir,

Please find enclosed a revised version of the manuscript "Electrodeposited $\text{Cu}_2\text{ZnSnS}_4$ thin films", by M. Valdes, M. Modibedi, M. Mathe, T. Hillie and M. Vazquez

We have modified the figure captions, as required.

I thank you in advance. Sincerely,

Dr. Marcela Vázquez

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5 Electrodeposited $\text{Cu}_2\text{ZnSnS}_4$ thin films
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Abstract

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4 $\text{Cu}_2\text{ZnSnS}_4$ (CZTS) thin films have been prepared using Electrochemical Atomic
5 Layer Deposition (EC-ALD) and also by one-step conventional constant
6 potential electrodeposition. Optimal deposition conditions were investigated
7 using cyclic voltammetry (CV). Then, based on CVs results, CZTS films were
8 grown employing EC-ALD deposition cycles using the sequence
9 Au/S/Cu/S/Zn/S/Sn/S to form the desired quaternary compound. In parallel,
10 conventional one-step electrodeposition was carried out at -0.85 V vs. Ag/AgCl
11 over 1 hour. A thermal treatment in sulfur vapor was also investigated in an
12 attempt to optimize the stoichiometry.

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14 The crystal structure of the films was characterized by XRD and micro Raman
15 spectroscopy, while the morphology, thickness, topography and elemental
16 composition were investigated using FIB-SEM and EDS.

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36 **Keywords:** electrodeposition; semiconductors; thin films; CZTS
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1. Introduction

The need to find new materials that can ensure mass production of photovoltaic panels has sparked interest in new alloys and composites. In this scenario, $\text{Cu}_2\text{ZnSn}(\text{S},\text{Se})_4$ (CZTS) has emerged as a serious substitute for $\text{CuInGa}(\text{Se},\text{S})_2$ (CIGSSe), where the indium and gallium are replaced by most abundant elements in the Earth's crust **therefore more economical** [1]. In particular, CZTS retains the fundamental properties for an ideal absorbent material in solar cells. These include a high absorption coefficient ($\alpha \sim 10^5 \text{ cm}^{-1}$) and direct band gap energy between 1 and 1.5 eV, which matches very well the solar spectrum [2].

Besides the impact that alternative materials can cause, it is important to achieve a **real cost reduction. Thus**, the techniques used to deposit thin films should involve low costs of equipment, allow depositions in large areas and be easily transferred to industrial scale. Electrodeposition meets all these characteristics and has been used for decades on an industrial scale for coatings with applications in several fields.

Different electroplating routes to obtain CZTS have been reported. So far the most commonly used methodology involves the electroplating of stacked layers of individual elements, followed by treatment in sulfur atmosphere to convert the precursor into CZTS [3-5]. This approach has led to high quality CZTS films yielding efficiencies close to 7% in solar cell devices [5]. **Other authors are reported to have chosen to prepare CTZ in a one step electrodeposition [6] or by co-electrodeposition [7] and later incorporated S by sulfurization during annealing.**

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Some authors have reported a direct electrodeposition of a Cu-Zn-Sn alloy followed by a thermal reaction in sulfur atmosphere [8, 9], while others include a sulfur source in the electrolyte to form a CZTS complete precursor solution [10-12]. The usual complication when using this method is to find a suitable potential window that allows the deposition of all ions in correct proportions, given that reduction potentials are far apart from each other. This can be tackled using complexing ions that shift the reduction potential of the noblest element, approaching it to values close to the more active ones. Citrate ions have been used as complexing agents, as reported by Pawar et al. [10, 13] and thiocyanate ions (CNS^-) by Iljina et al. [14].

Over the last few years, another method for electrodepositing thin films of semiconducting materials has been used with satisfactory results. Electrochemical atomic layer deposition (EC-ALD) is based on atomic layer deposition (ALD). Alternate underpotential deposition (UPD) of the elements in a cycle promotes thin-film formation by deposition of the elements one atomic layer at a time. The method relies on a series of electrochemical surface-limited reactions, which are used to deposit each atomic layer and was originally developed by Stickney and coworkers [15, 16]. Every deposition cycle gives a monolayer of the desired compound, so that the thickness of the film is governed by the number of deposition cycles [17]. The deposition of each individual element can be controlled separately, by carefully choosing conditions such as potentials, electrolyte composition and concentration, pH and deposition times. These conditions have to be adjusted depending on the desired deposit and the substrate being employed. Promising results for EC-ALD for related ternary compounds like CuInS_2 [18], CuInSe_2 [19, 20] and

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 $\text{Cu}_x\text{Sn}_y\text{S}_z$ [21] have been reported recently. However, the deposition of a quaternary compound is even more challenging.

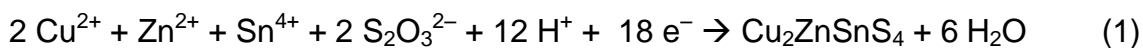
The purpose of this work is to evaluate diverse electrochemical alternatives to prepare thin films of $\text{Cu}_2\text{ZnSnS}_4$, namely EC-ALD and conventional, potentiostatic, one-step electrodeposition.

2. Experimental

2.1. Potentiostatic deposition of $\text{Cu}_2\text{ZnSnS}_4$ films

$\text{Cu}_2\text{ZnSnS}_4$ (CZTS) films were electrodeposited on top of fluorine-doped tin oxide (FTO, from Pilkington TEC Glass, ~ 8 Ohms/sq) electrodes ($20 \times 20 \times 3$ mm³). Prior to the deposition, both substrates were cleaned successively in detergent and isopropyl alcohol solution in an ultrasonic bath. A three-electrode cell, with a Pt mesh as counter electrode and a saturated calomel electrode (SCE) as reference electrode, was used. Electrodeposition was carried out employing a PGZ 402 Voltalab[®] potentiostat/galvanostat. The electrolytic bath consists of an acidic aqueous solution containing 0.02 mol/L CuCl_2 , 0.015 mol/L $\text{Zn}(\text{NO}_3)_2$, 0.02 mol/L SnCl_4 and 0.02 mol/L $\text{Na}_2\text{S}_2\text{O}_3$. Also, 0.2 mol/L sodium citrate ($\text{Na}_3\text{C}_6\text{H}_5\text{O}_7$) was used as complexing agent and the pH was adjusted between 4-4.5 using tartaric acid ($\text{C}_4\text{H}_6\text{O}_6$). CZTS thin films were electrodeposited in potentiostatic mode, applying a potential of -0.85 V vs. SCE (saturated calomel electrode). The deposition time was set at 1 hour and the solution was purged with nitrogen prior to the electrodeposition. After completing the electrodeposition the samples were rinsed with distilled water

1 and dried in air. The general equation for electrodeposition of CZTS can be
2 described as:
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11 To improve the crystallinity of the films, thermal treatments were carried out in a
12 three-temperature zone thermal reactor. First, the samples were placed in the
13 cold zone of the tube (room temperature). Then the second zone was heated to
14 500°C and in the third zone the sulfur powder (0.3 g) was allowed to reach
15 300°C (melting starts at around 200-250°C). At this point, the samples were
16 moved to the second zone in order to start the sulfurization process, as the
17 vaporized sulfur got in contact with the samples assisted by argon flux. After 1
18 hour the samples were cooled down by placing them back in the “cold zone”.
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31 32 33 34 2.2 EC-ALD deposition of $\text{Cu}_2\text{ZnSnS}_4$ films

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36 The CZTS films were prepared using an automated flow cell electrodeposition
37 system (Electrochemical ALD L.C., Athens, GA). The set-up includes a series of
38 solution reservoirs, pumps and valves, an electrochemical flow cell, a
39 potentiostat, and specialized “Sequencer” software. The reference electrode
40 was Ag/AgCl (3 mol/L KCl) and a gold wire acted as counter electrode. The
41 electrochemical flow cell was designed to promote laminar flow, with the cell
42 volume being 0.3 mL. Solutions were pumped at 50 mL min^{-1} . Pumps and
43 plumbing were contained in a nitrogen purged Plexiglas box. A schematic
44 representation of the equipment is shown in Figure 1.
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1 Substrates used were Au coated glass slides (TA134, 50 Å Ti, 1000 Å Au, EMF
2 Corp., New York). The exposed deposition area is 2.25 cm².

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4 Reagent grade chemicals were used to prepare the following electrolytes: 5 10⁻⁴
6 mol/L CuCl₂, 5 10⁻⁴ mol/L ZnSO₄, 1 10⁻³ mol/L SnCl₂ in HCl (pH=1) and 6 10⁻⁴
7 mol/L Na₂S₂O₃ in 0.1 mol/L KOH. In every case except for SnCl₂, 0.1 mol/L
8 Na₂SO₄ was used as supporting electrolyte.

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10 A potential vs. time profile illustrating the EC-ALD cycle chosen to deposit CZTS
11 is shown in Figure 2. UPD potentials for depositing each element were chosen
12 as described below and fixed at the following values, reported against Ag/AgCl:
13 -0.82 V to deposit S, +0.16 V to deposit Cu, -0.75 V to deposit Zn and -0.48 V to
14 deposit Sn. One cycle consisted of letting the solution of each element flow into
15 the cell for 30 s at the potential chosen for the deposition of the corresponding
16 element in the following order:

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31 S/Cu/S/Zn/S/Sn/S/Cu/S/Zn/S/Sn
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34 This process will be referred to as one CZTS EC-ALD cycle, and was repeated
35 100 times.
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41 2.3. Materials characterization

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44 The crystalline structure of the films was analyzed by X-ray diffraction in grazing
45 incidence configuration (GXRD) using a PANalytical X'Pert PRO diffraction
46 system employing Cu-K α radiation at 40 kV and 40 mA. The samples were
47 scanned between 15° and 75°, with a step size of 0.01° and with the X-ray
48 beam fixed at 3°. The crystallographic data for each phase were taken from the
49 literature [22].
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1 Raman spectroscopy measurements were performed using an Invia Reflex
2 confocal Raman microprobe using a 50x objective. Excitation was provided with
3 the 514 nm emission line of an Ar⁺ laser. In this configuration the laser power on
4 the sample was less than 2 mW measured with a silicon photodiode (Coherent
5 Inc.). For this condition, no thermal effects could be detected as a result of
6 recording the Raman spectra. Raman spectra were taken by averaging 3
7 acquisitions of 20 seconds each. Raman micro-mapping was performed
8 scanning a square zone (40 x 40 μm) in the sample and recording 25 spectra in
9 the x and y directions. For both axes the step between spectra was set in 20
10 μm.
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24 The morphology and the chemical composition of the films were registered with
25 a field emission scanning electron microscope (Carl Zeiss Supra 40 FESEM)
26 coupled with an X-ray microanalysis system (Oxford Instrument, INCA
27 processor). Cross-section images of the films were obtained using a Focused-
28 Ion Beam Scanning Electron Microscope (Augira Cobra FIB SEM).
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41 **3. Results and Discussion**

42 **3.1. Electrodeposition of Cu₂ZnSnS₄ films**

43 Figure 3 presents GXRD diffractograms of electrodeposited samples, before
44 and after the thermal treatment in sulfur. The peaks resulting from FTO (mostly
45 SnO₂) have been identified as F. The **pattern** also show some peaks that can
46 be attributed to the CuKβ radiation from the X rays that cannot be filtered due to
47 the incidence angle **(labeled as β)**. Further than these, the diffractogram
48 corresponding to the as-deposited sample (Figure 3a) shows peaks that can be
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1 ascribed to binary Cu compounds, mainly Cu_2S and Cu_2O . It is clear that only
2 after annealing can diffraction planes from the CZTS phase be identified in the
3 diffractogram. The major diffraction peaks appear at $2\theta = 28.5^\circ$, 47.9° and 56.2°
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5 attributable to the (112), (220), and (312) crystallographic planes of CZTS.
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7 These diffraction planes match the standard XRD pattern of the $\text{Cu}_2\text{ZnSnS}_4$
8 phase (PDF #26-0575). These results are in good agreement with those
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10 obtained by Pawar et al. [10, 13], who used a similar precursor bath but
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12 employed a Mo-coated substrate and performed the thermal treatment in Ar
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14 atmosphere. Similar results have also been shown by Cui et al. [11] and Iljina et
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16 al. [14], even when in these cases the annealing treatment was performed in an
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18 atmosphere of $\text{N}_2 + \text{H}_2\text{S}$ (5%). Binary secondary phases were detected using
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20 both Mo coated glass and ITO as substrates by Cui et al. [11].
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29 Secondary phases are frequently formed during the deposition of CZTS. Even
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31 more, in the case of electrodeposited CZTSe, recent mechanistic studies
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33 suggested that this compound was formed by reactions among the
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35 corresponding binary selenides: Cu_2Se , ZnSe , and SnSe_2 [12]. As regards
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37 CZTS, the most frequently found byproducts are binary compounds, such as
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39 Cu_xS in Cu-rich and ZnS in Zn-rich deposits, as well as ternary phases such as
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41 Cu_2SnS_3 and ZnSnO_3 . However, ZnS with a blend structure (PDF #26-0566),
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43 exhibits some planes that are quite close to CZTS (112) and (200) diffraction
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45 planes. Thus, Raman spectroscopy is a good complement to XRD and has
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47 been successfully used to analyze the structure and the phase purity of CZTS
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49 samples [23, 24]. Figure 4 presents Raman spectra of CZTS films, comparing
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51 as-deposited and annealed samples. As it can be seen, the spectrum of the as-
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53 deposited sample is dominated by bands that can be attributed to binary
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phases, mainly SnS₂ and Cu_{2-x}S. Again, as in XRD analysis, the presence of CZTS cannot be confirmed, as no band can be ascribed to this quaternary compound. Besides, a couple of bands could not be attributed to any expected byproduct. Instead, after sulfurization there are two signals with peaks at 334 and 286 cm⁻¹ that can be related to the two main vibrational modes of kesterite, both with A symmetry [25]. The spectrum in Figure 4 also shows weak contributions at 134, 255 and 362 cm⁻¹ that can be attributed to low intensity E and B symmetry modes. After annealing, the spectrum becomes less noisy and exhibits a decrease in the Full Width at Half Maximum (FWHM) of the main bands that can be taken as indicative of an improvement in the crystalline degree of the material.

Raman maps of electrodeposited and annealed films are presented in Figure 5 to show the homogeneity of the composition throughout on the surface. The two main vibrational modes of kesterite with A symmetry (at 334 and 286 cm⁻¹) are always present. This demonstrates that CZTS is evenly distributed along the whole area under analysis. The signal observed around 470 cm⁻¹, originated by Cu_{2-x}S, is present with irregular intensity in most of the spectra. In contrast, Raman spectra of annealed films with a monophasic composition have been reported by other authors using thiocyanate as complexing agents in the precursor bath [14] or an atmosphere of N₂ + H₂S (5%) during annealing [26]. However, in spite of the advantages of Raman spectroscopy mentioned above, there is limited information in the literature of Raman analysis for electrodeposited CZTS. Even in the few cases where the technique is employed, no maps are shown to indicate if the composition and in particular, the absence of binary phases was homogenous along the whole surface.

1 SEM images showing the morphology of as-deposited and sulfurized CZTS
2 films are presented in Figure 6. As-deposited samples are highly porous and
3 consist of small aggregates. This morphology is very similar to that reported
4 earlier by Jeon et al. [27]. The size of the grains seems to increase after
5 annealing in sulfur, which is consistent with the improvement in the crystalline
6 degree deduced from XRD and Raman results. The grain size of the film after
7 annealing is found to be larger than 1 μm . The morphology of the annealed
8 samples resembles platelets and secondary phases are easy to distinguish. Cui
9 et al. also noted some white spots situated at the surface of an electrodeposited
10 film, and attributed it to Cu_2S crystallites [11].
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24 The chemical composition of the CZTS films before and after the thermal
25 treatments was determined by EDS and is presented in Table 1. It can be
26 observed that the amount of copper in the film decreases while the sulfur
27 content increases with annealing, so that both tend to their respective
28 stoichiometric values. In contrast, the zinc and tin content seems to be much
29 less affected by the sulfurization treatment. Even after annealing in sulfur vapor,
30 the films seem to be slightly deficient in S and comparatively rich in Cu.
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44 3.2. CZTS films deposited by EC-ALD

45 Cyclic voltammograms were recorded for Au/glass substrates using Cu, S, Sn
46 and Zn precursor solutions, so as to have indicative values for UPD potentials
47 that will be later used to grow the stacks that lead to the quaternary compound.
48 The results obtained for Sn and Cu are shown in Figure 7. Figure 7a is a CV of
49 a clean Au substrate in the Cu^{2+} solution which displays a broad UPD region
50 around 0.16 V, just before with the onset of bulk Cu deposition peak. In the
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1 case of Sn, the UPD region can be seen at around -0.48 V (Figure 7b). The
2 potentials to deposit Zn and S were defined in a similar way (not shown), and
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4 resulted to be -0.8 V for Zn and -0.45 V for Sn.
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7 FIB-SEM images were captured to follow the formation of the film and to
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9 evaluate the thickness. The a cross-section of a sample deposited using 100
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11 cycles of EC-ALD can be clearly observed in Figure 8(a). The deposit presents
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13 some voids but the average thickness can be estimated in 150 nm as shown in
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15 Figure 8(b).
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19 The chemical composition of films deposited by EC-ALD was determined by
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21 EDS. A typical spectrum is presented in Table 2. It can be seen that all the
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23 elements are present in the deposit. EDS maps (not shown) demonstrate that
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25 they are evenly distributed along the surface.
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29 XRD/Raman
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32 33 34 **4. Conclusions** 35

36 Electrochemical processes, such as conventional electrodeposition and
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38 electrochemical atomic layer deposition (EC-ALD) are suitable to produce
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40 photovoltaic materials, and present themselves as interesting options that can
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42 replace high-vacuum techniques.
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46 Good quality semiconductor thin films can be obtained with one-step
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48 electrodeposition as was shown with SEM, XRD and Raman spectroscopy. A
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50 post deposition treatment in sulphur atmosphere (thermal annealing) is a key
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52 step to improve the properties of the films obtained with one-step
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54 electrodeposition. Even if binary undesirable phases appear, they could be
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56 etched with a quick immersion in KCN solution.
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In the case of **EC-ALD**, preliminary results show that 100 cycles produce a film of 150 nm of thickness, where EDS analysis shows that all the elements are present. A more exhaustive characterization of the film is currently in progress.

Acknowledgements

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Figure captions

Figure 1. Schematic representation of the equipment used to deposit CZTS using EC-ALD

Figure 2. Potential vs. time used to deposit CZTS using EC-ALD.

Figure 3. GXRD patterns of CZTS electrodeposited on FTO at $E = -0.85$ V during 1 h. a) as-deposited film, dashed line; b) films annealed in sulfur, full line.

Peaks labeled F denote diffraction planes from FTO while those labeled β are diffraction planes produced from unfiltered $\text{CuK}\beta$ radiation. b

Figure 4. Raman spectra of CZTS films electrodeposited on FTO at $E = -0.85$ V during 1 h for as deposited and annealed samples.

Figure 5. Micro-Raman mapping performed over a 40×40 μm region of CZTS films electrodeposited on FTO at $E = -0.85$ V during 1 h. The sample is later annealed in S vapor.

Figure 6. SEM images (10000 X) of CZTS films (a) as-deposited and (b) after sulfurization.

Figure 7. (a) Cyclic voltammogram in $5 \cdot 10^{-4}$ mol/L $\text{CuCl}_2 + 0.1$ mol/L Na_2SO_4 for a Au electrode, (b) Cyclic voltammogram in $1 \cdot 10^{-3}$ mol/L SnCl_2 in HCl ($\text{pH}=1$) Scan rate: 10 mV s^{-1} .

Figure 8. FIB-SEM cross images of CZTS films deposited by EC-ALD (100 cycles). (a) general view; (b) thickness estimation in detailed view.

Figure 9. EDS spectrum of CZTS films deposited by EC-ALD (100 cycles).

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Table 1

	Cu (at%)	Zn (at%)	Sn (at%)	S (at%)	Cu/(Zn+Sn)	S/(metal)
Stoichiometric	25.0	12.5	12.5	50	1	1
-0.85 V, as-deposited	52.2 ± 4.6	16.2 ± 1.7	7.2 ± 0.8	24.4 ± 1.6	2.2	0.3
-0,85 V, sulfurized	30.6 ± 1.7	15.0 ± 1.4	7.9 ± 2.3	46.5 ± 0.8	1.3	0.9

Table 1. Chemical composition obtained by EDS of different CZTS films. All the films were deposited at 1 hour.

Figure 1
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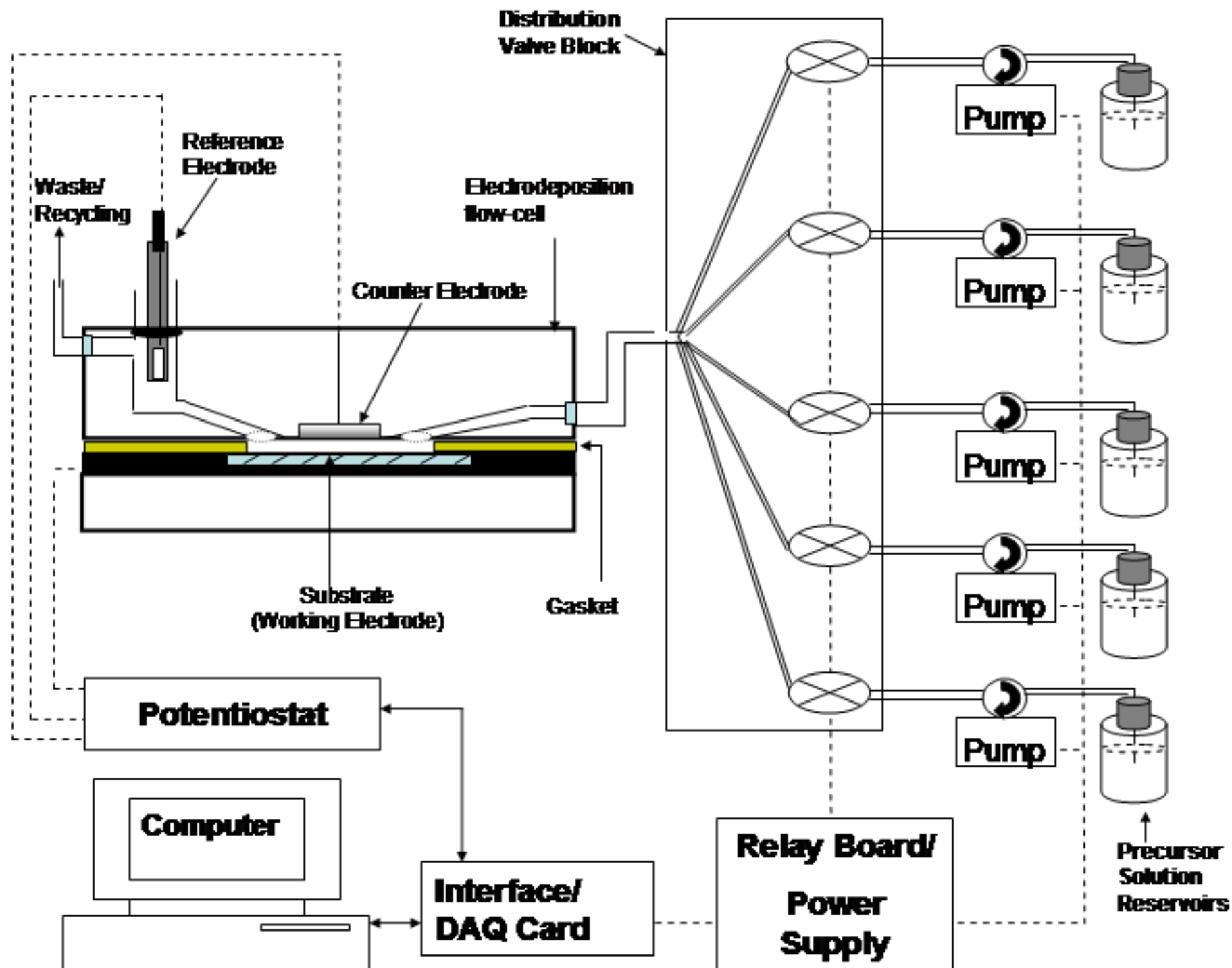


Figure 2
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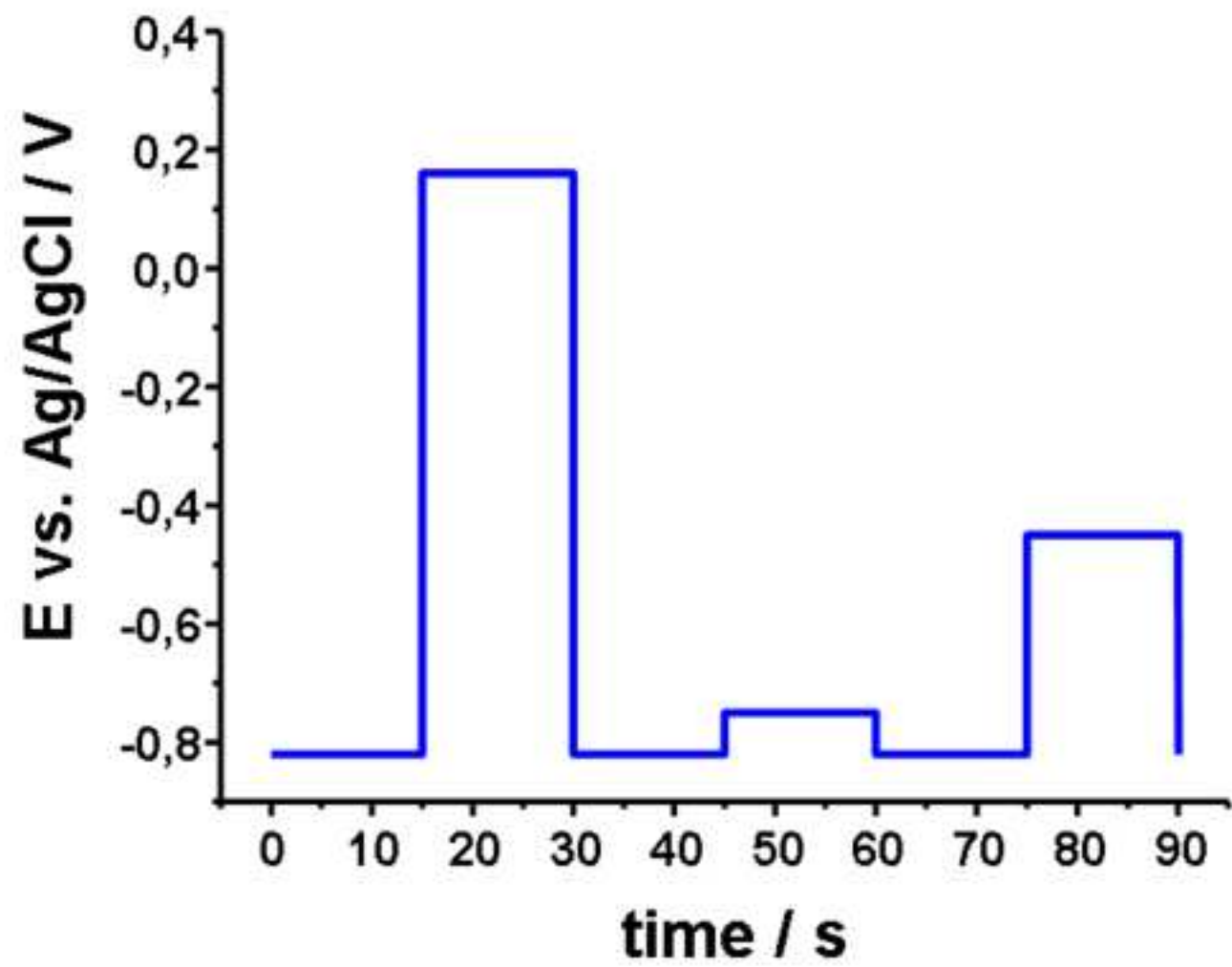


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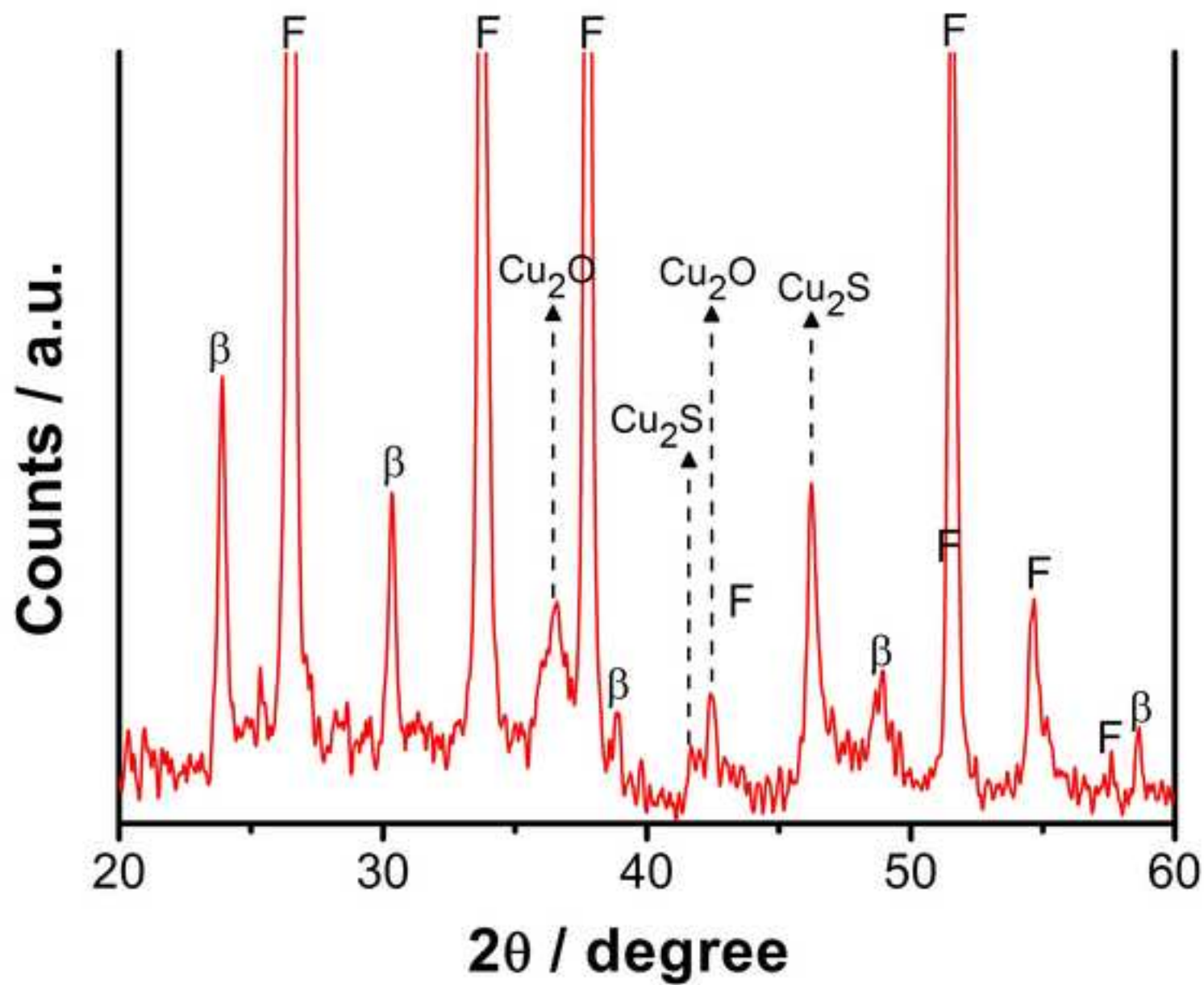


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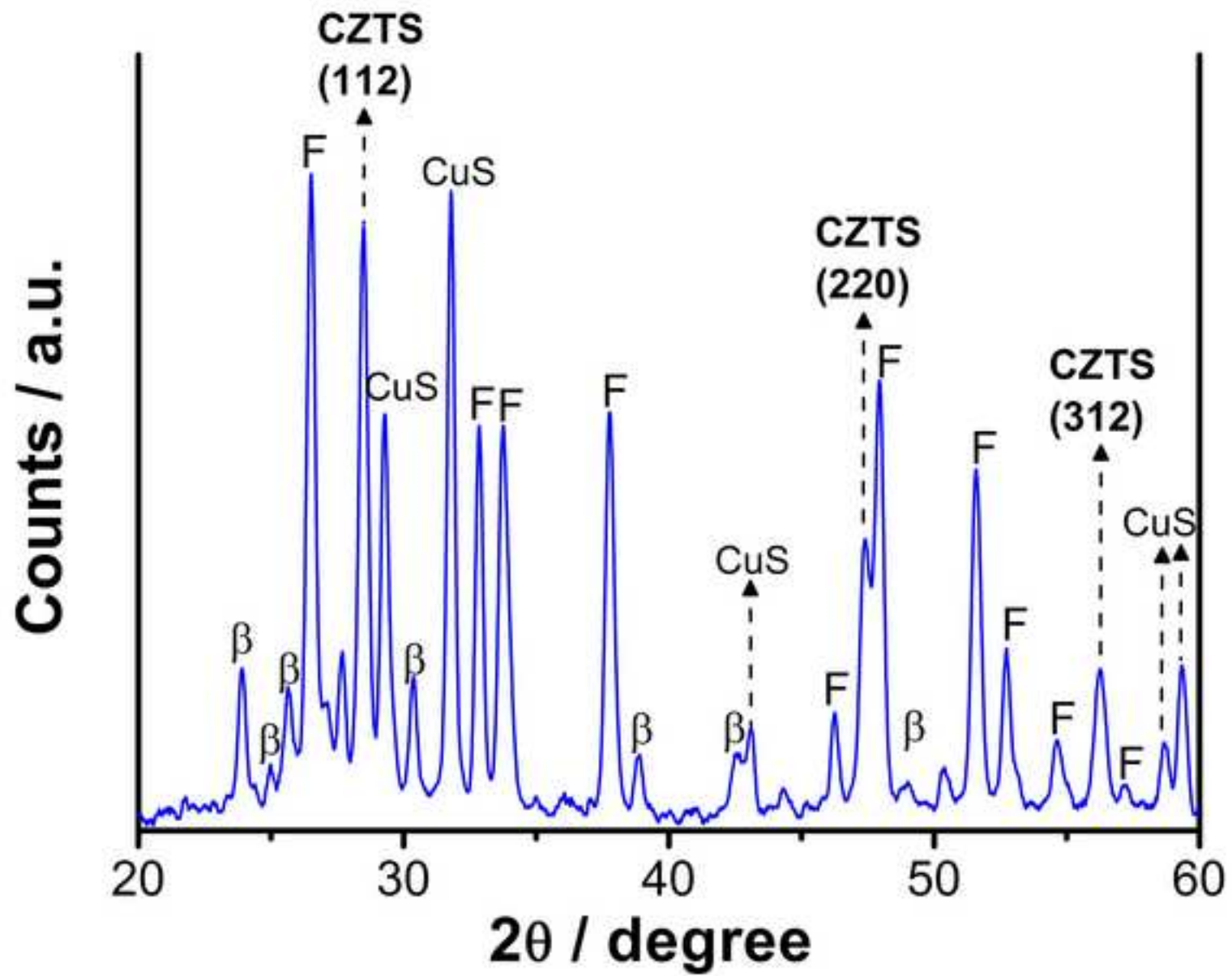


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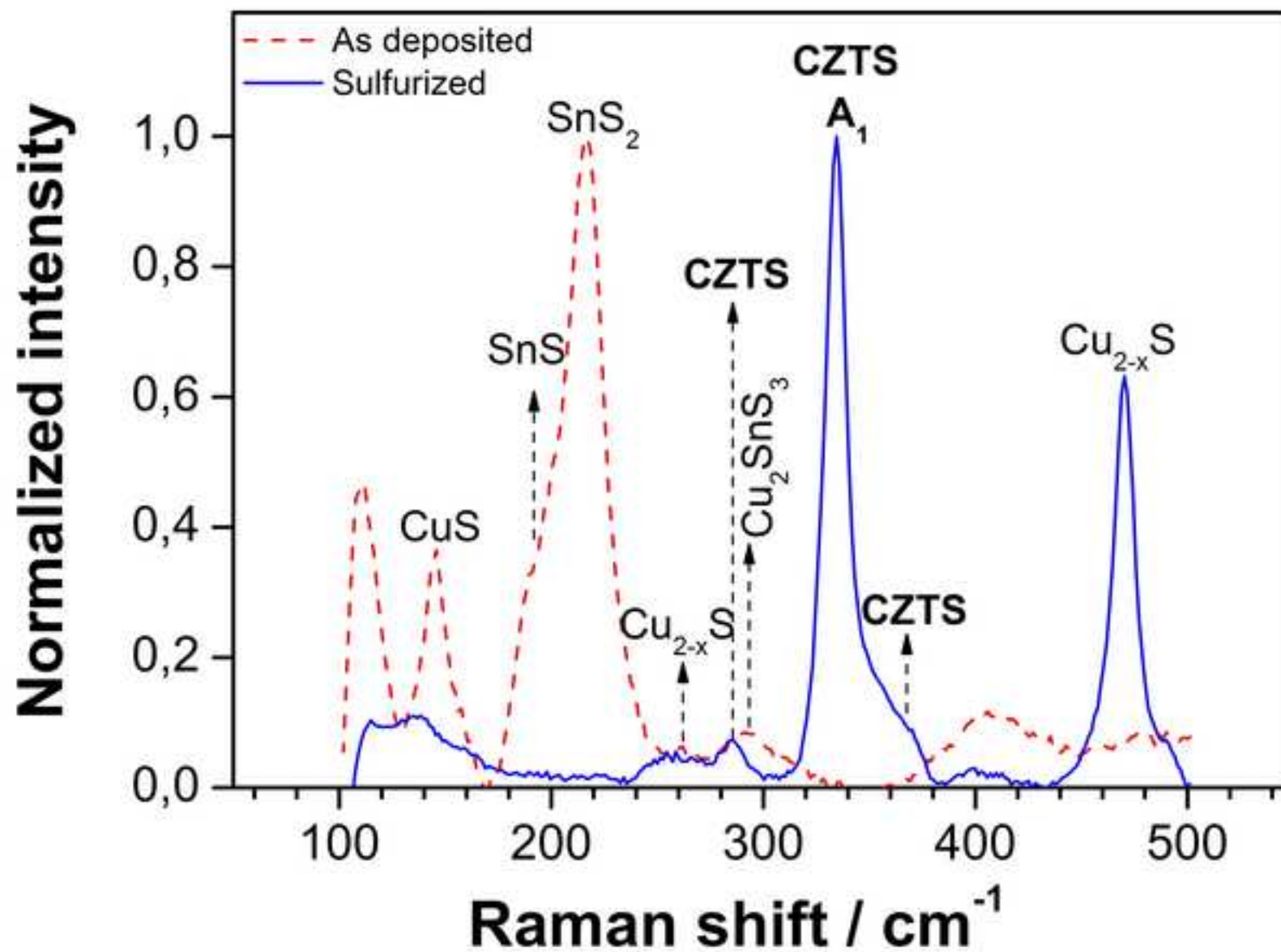


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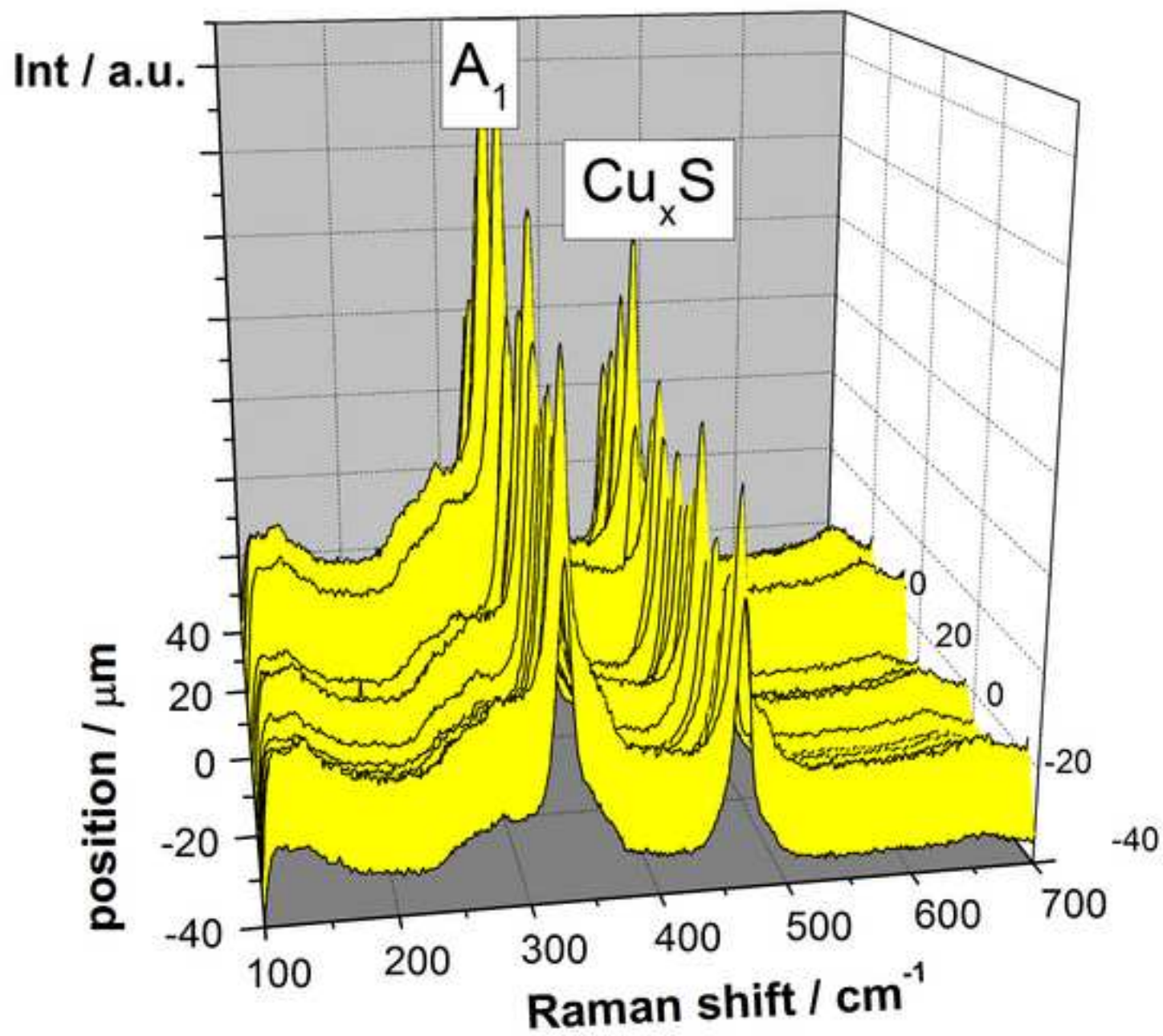
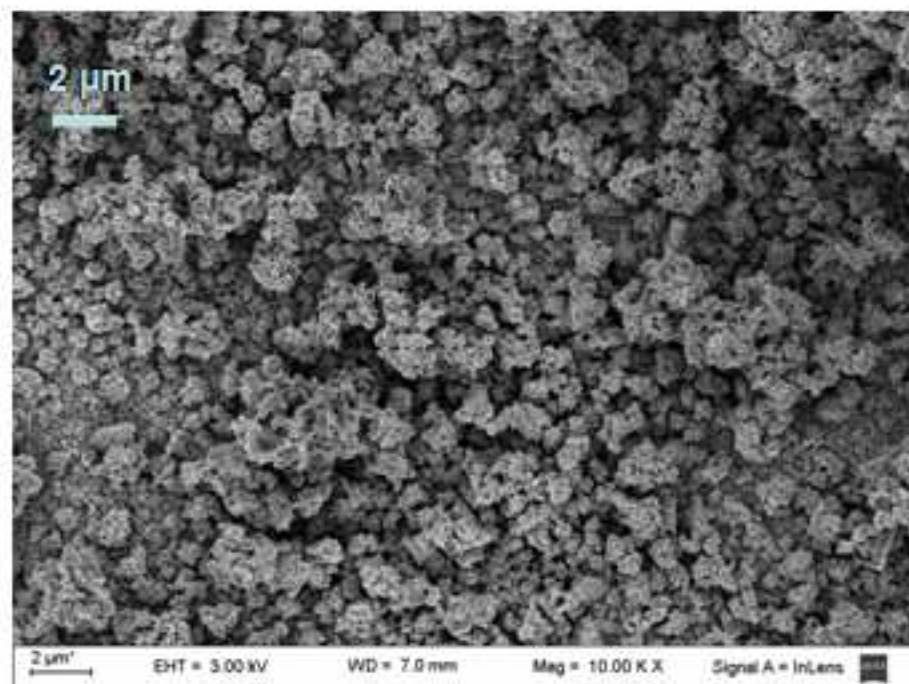
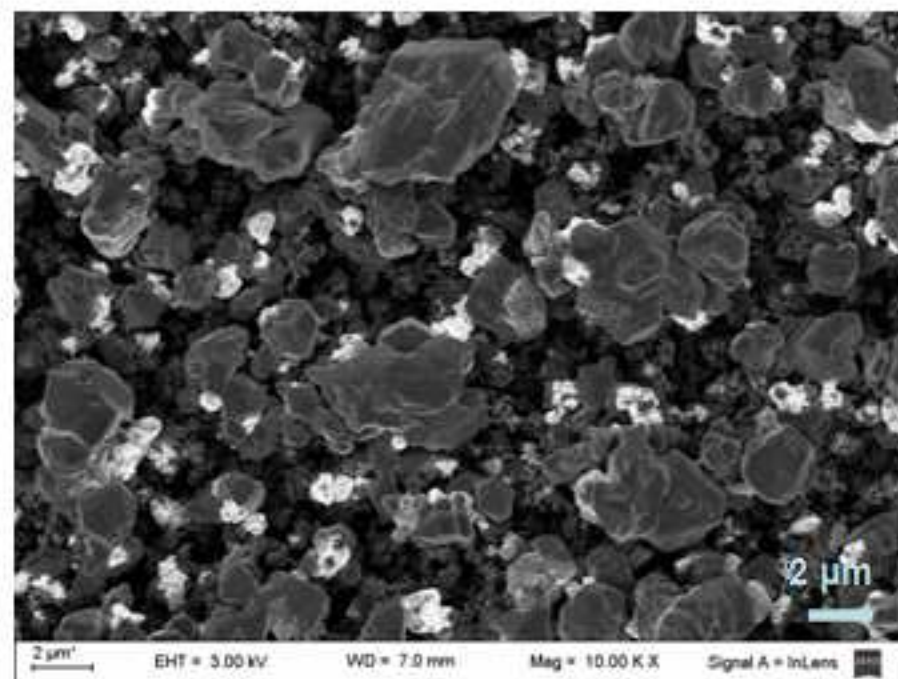


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(a)



(b)

Figure 7a

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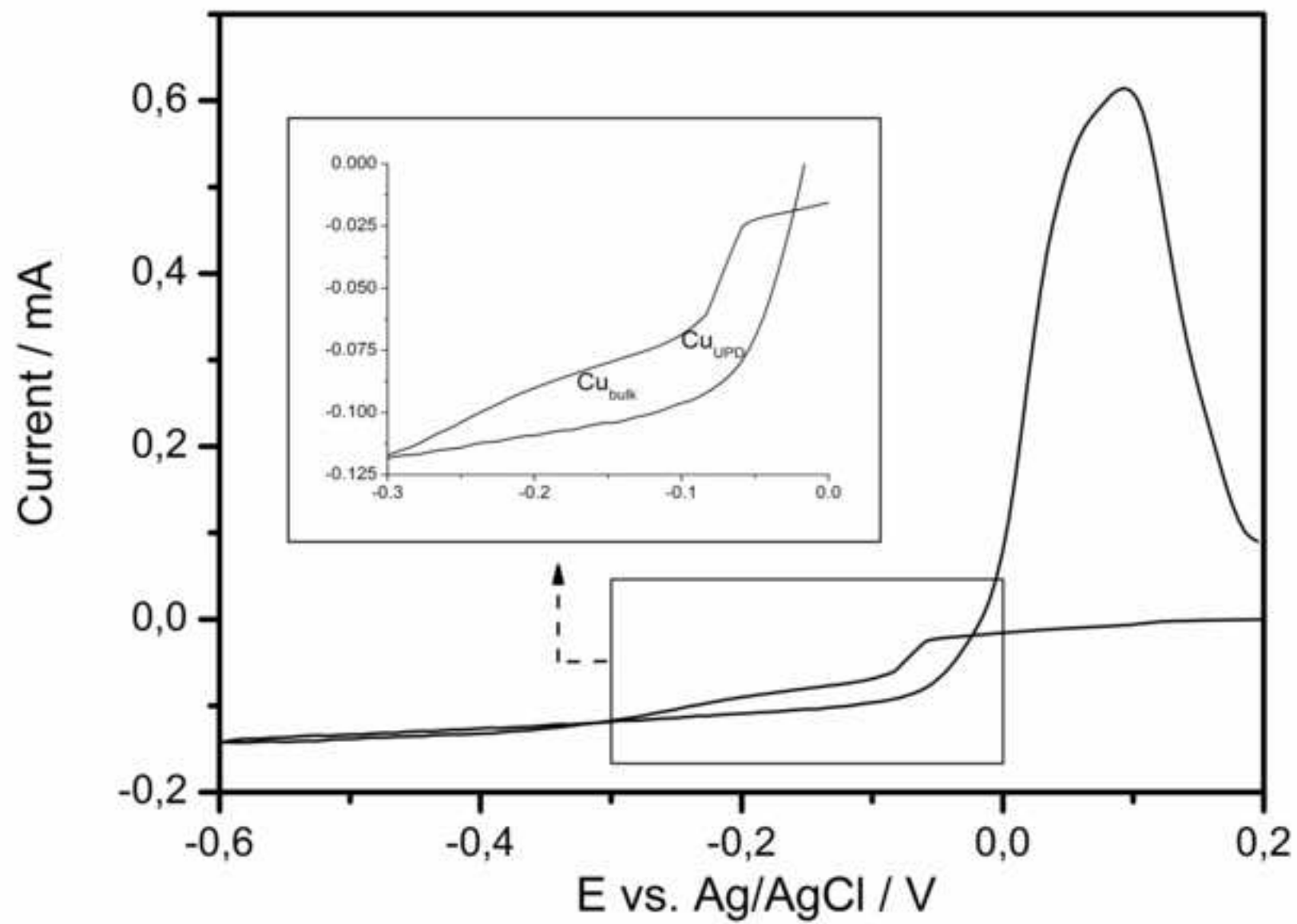


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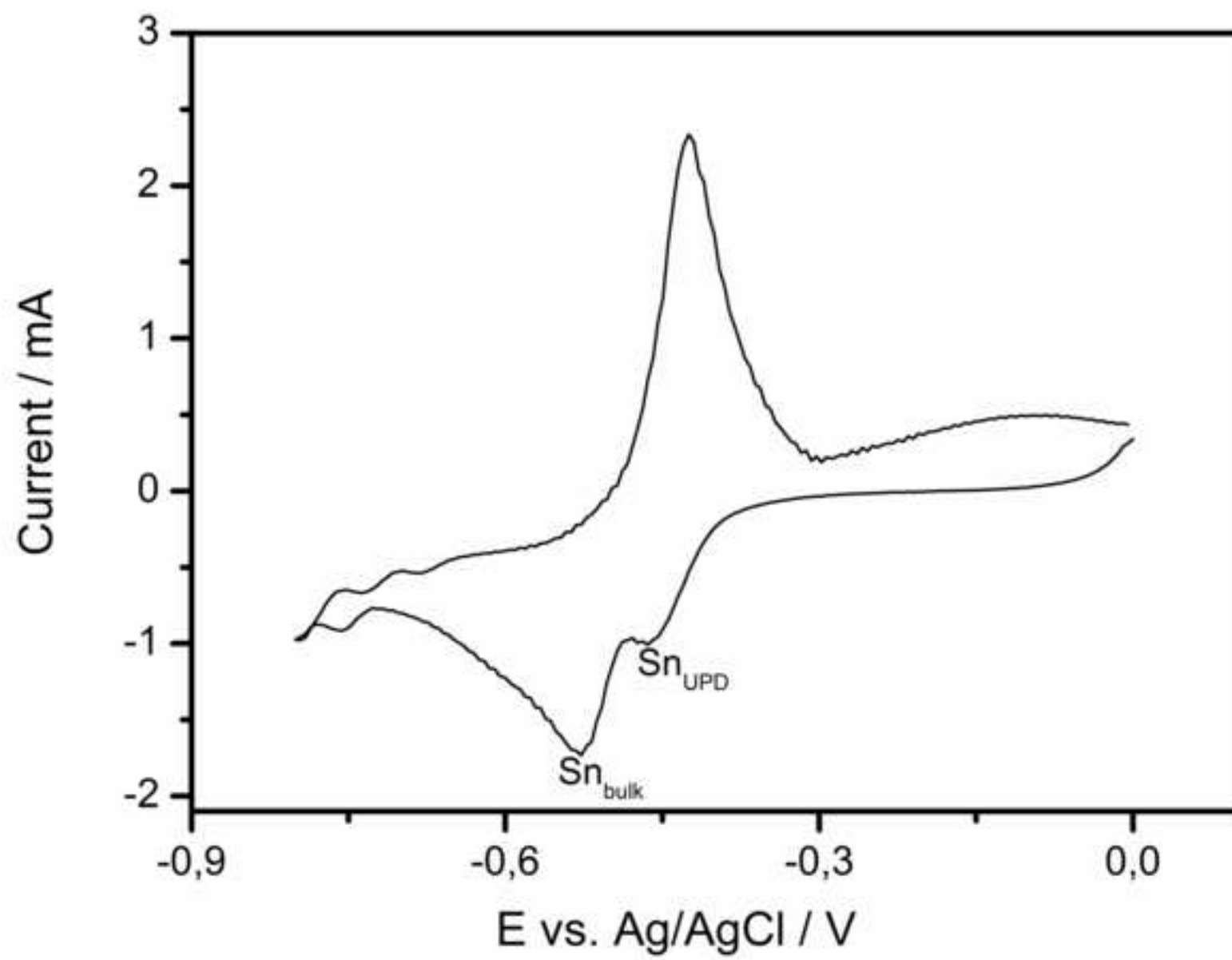


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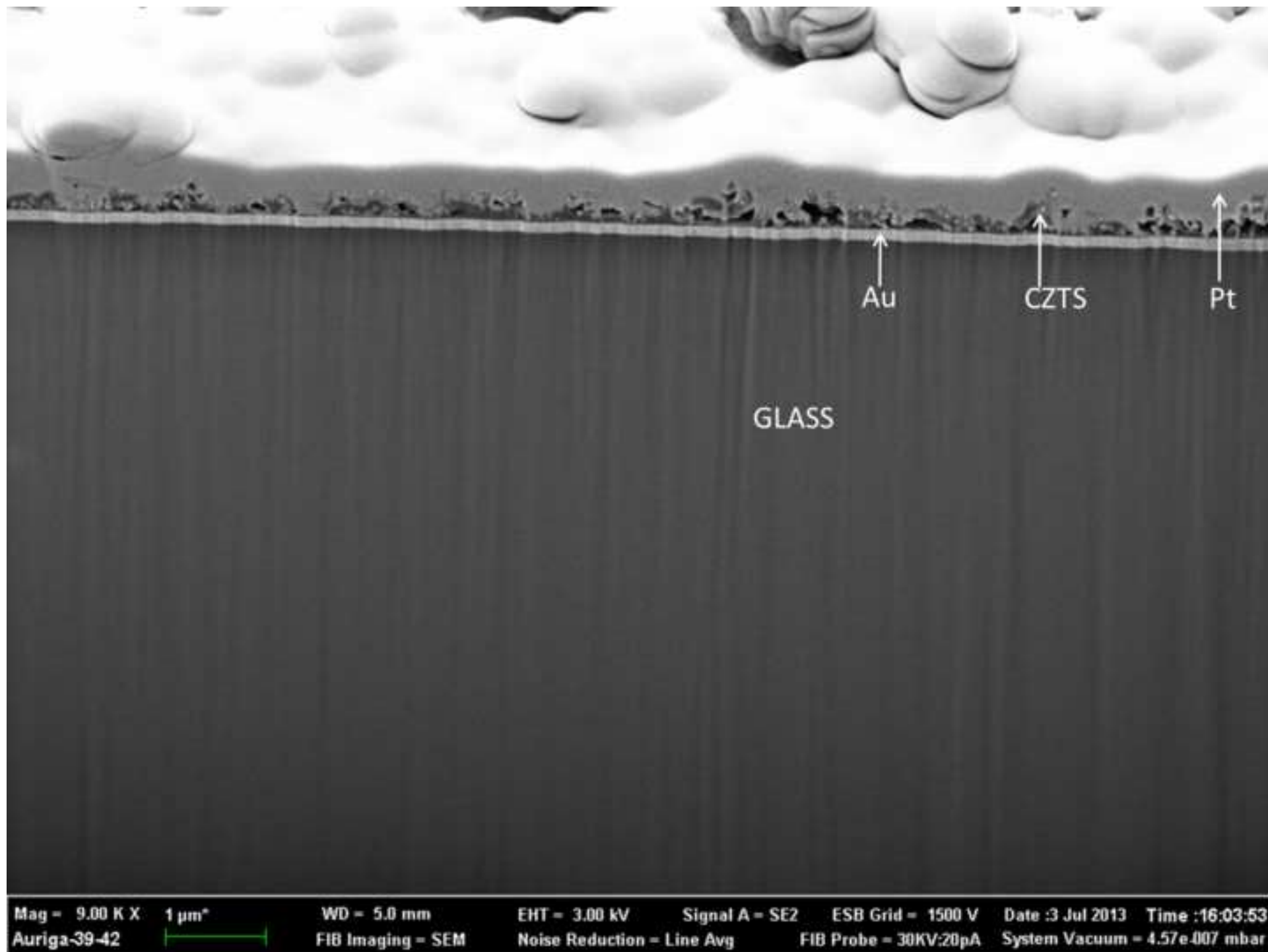


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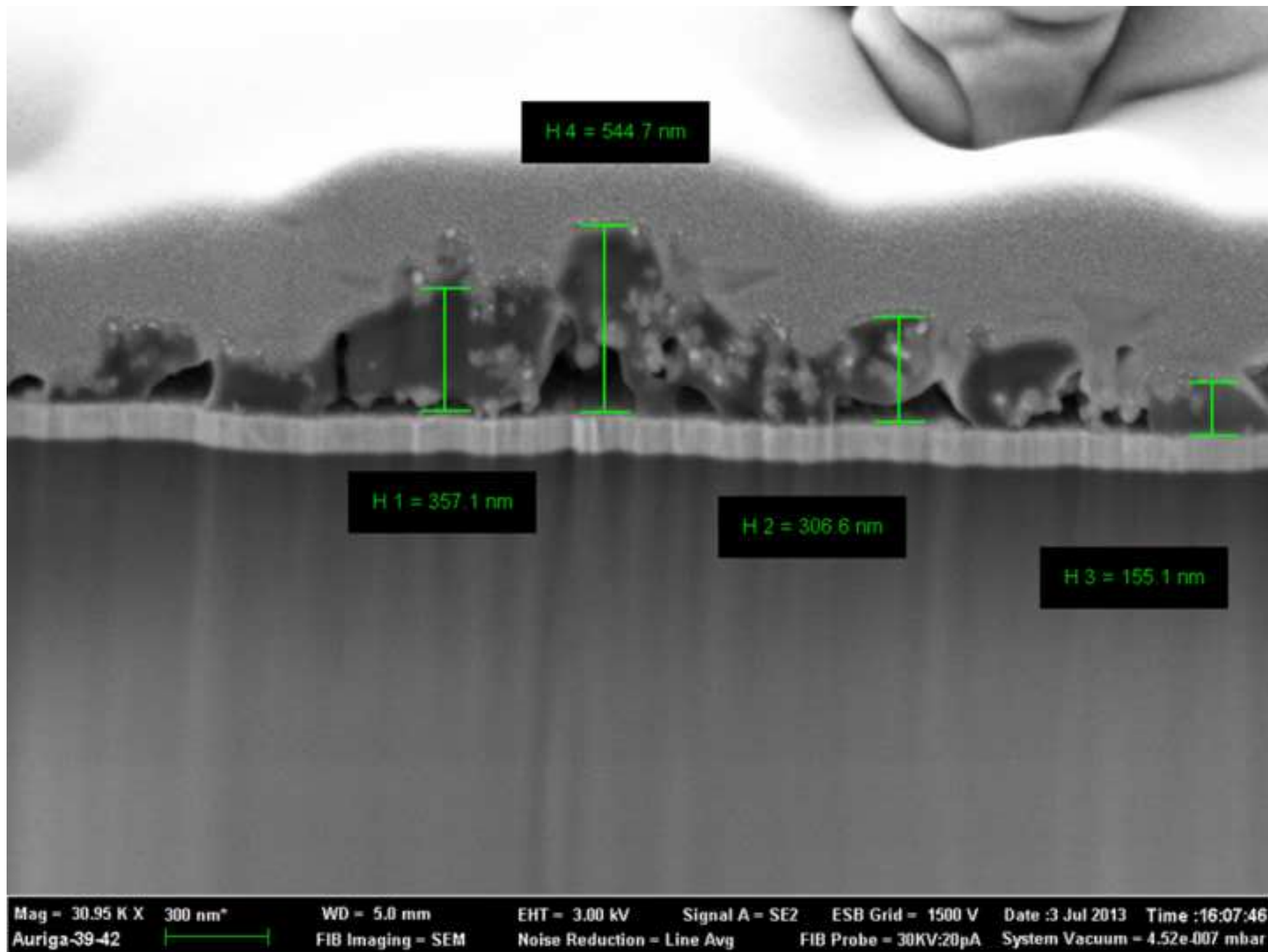


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