

# Theoretical prediction of atomic and electronic structure of neutral $\text{Si}_6\text{O}_m$ ( $m=1-11$ ) clusters

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(Received 14 October 2008; accepted 20 January 2009; published online 7 April 2009)

In this paper we found the most stable structures of silicon-oxide clusters of  $\text{Si}_6\text{O}_m$  ( $m=1-11$ ) by using the genetic algorithm. In this work the genetic algorithm uses a semiempirical energy function, MSINDO, to find the best cluster structures of  $\text{Si}_6\text{O}_m$  ( $m=1-11$ ). The best structures found were further optimized using the density functional theory. We report the stable geometries, binding energies, lowest unoccupied molecular orbital-highest occupied molecular orbital gap, dissociation energies for the most favorable fragmentation channels and polarizabilities of  $\text{Si}_6\text{O}_m$  ( $m=1-11$ ). For most of the clusters studied here we report structures not previously found using limited search approaches on common structural motifs. © 2009 American Institute of Physics.

[DOI: 10.1063/1.3080549]

## I. INTRODUCTION

Silica is a very important material, one of the most abundant materials on earth.<sup>1</sup> Its nanoparticles are interesting because of their importance in technological applications, such as microelectronics<sup>2</sup> and optics,<sup>3</sup> its uses, ranging from glass to catalysis,<sup>4</sup> and the very important optical fiber communications.<sup>5</sup> Since silica is one of the materials that shows novel properties in the nanoscale range, the study and understanding of the structure, bonding, stability, and properties of their clusters and nanoparticles is very important and necessary.<sup>6</sup> Since El-Shall and co-workers<sup>7</sup> observed bright blue photoluminescence in silica, different authors tried to justify the experimental observation by large concentration of oxygen defects related to the size of the lowest unoccupied molecular orbital-highest occupied molecular orbital (LUMO-HOMO) gap.<sup>8</sup> Silica clusters show light absorption<sup>8</sup> and photoluminescence<sup>9</sup> because their particular structures are very different from the conventional bulk material. Another interest is silica as a molecular building material, since the combination of its bulk stability and the advantage of its abundance put it in a very good position to be exploited via synthetic methods.<sup>10</sup>

Cages and tubes of  $\text{SiO}_2$  are an integral constituent part of the extended periodic systems known as zeolites.<sup>11</sup> Silicon oxides have a fundamental role in the growth of SiNWs because it has been proven experimentally that the yield of SiNWs will be greatly enhanced if silicon oxide is present during the synthesis.<sup>12</sup>

The structure and properties of silicon oxides clusters have been studied from both experimental<sup>13-16</sup> and theoretical points of view.<sup>17-19</sup> Photoelectron spectra of some silicon oxide anions have been studied by Wang *et al.*,<sup>14-16</sup> providing information about electronic structures of the clusters.

Chelikowsky<sup>17</sup> found that buckled rings of  $\text{Si}_4\text{O}_4$  and  $\text{Si}_5\text{O}_5$  are more stable than planar ones. Song and Choi<sup>20</sup> compared the stability of elongated and compact  $\text{SiO}_2$  clusters with 12-46 molecules using plane-wave density-functional theory (DFT) and provided a very good discussion about effects of size, morphology, and different types of silica rings and their electronic and optical properties. Nayak *et al.*<sup>21</sup> studied structure and properties of  $(\text{SiO}_2)_n$  ( $n=1-6$ ) and  $\text{Si}_3\text{O}_n$  ( $n=1, 3, 4$ ) employing *ab initio* methodology, while Zhang *et al.*<sup>19</sup> and Chu *et al.*<sup>22</sup> provided a detailed theoretical study of structures of  $\text{Si}_n\text{O}_m$  ( $n, m=1-8$ ) by the DFT-B3LYP (Ref. 23) method. Zhang *et al.*<sup>24</sup> demonstrated that the two- (2MR) and three-membered ring (3MR) hybrid clusters are energetically more favorable than the pure 2MR or 3MR clusters indicating their suitability like blocks of construction for new materials.

Wang *et al.*<sup>25</sup> studied the segregation of  $\text{Si}_n\text{O}_n$  in different motifs because of the great interest in the observation of high-yield silicon nanowire (SiNW) growth from SiO vapor.<sup>12,26,27</sup> Experiments showed that SiNWs can be obtained by thermal evaporation or laser ablation of Si powder mixed with  $\text{SiO}_2$  or SiO,<sup>12,26,27</sup> but there is no SiNWs growth from pure  $\text{SiO}_2$  materials.<sup>12</sup> Reber *et al.*<sup>28</sup> demonstrated that SiO units assemble to form bigger clusters, with silicon rich agglomerations at the center and oxygen rich structures at the periphery.

In spite of the large number of calculations of properties and electronic of oxide clusters, the understanding of their structures is still an open subject. In this article we propose to search for the structures of  $\text{Si}_6\text{O}_m$  in the energy hypersurface employing our modified genetic algorithms (MGAC) package,<sup>29-31</sup> which uses the parallel implementation of genetic algorithms (PGA),<sup>31</sup> a stochastic method. PGA is, in conjunction with basin-hopping Monte Carlo<sup>32,33</sup> and simulated annealing,<sup>34</sup> one of the best approaches to explore and find global and secondary minima of complex energy potential surfaces. This simulation technique permits to handle

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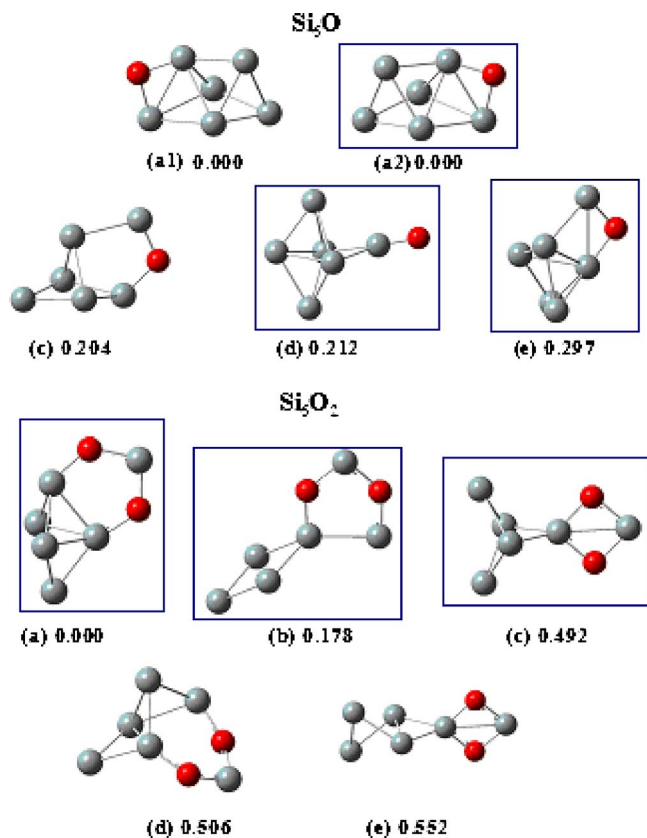


FIG. 1. (Color online) Optimized structures for the five lower-energy isomers for  $\text{Si}_6\text{O}$  and  $\text{Si}_6\text{O}_2$  clusters at the B3LYP/6-311G(2d) level of theory. BE in eV based on a Si atomic energy of  $-7847.211$  eV and O atomic energy of  $-2036.020$  eV. The structures similar to those reported in the literature are enclosed by frames.

with clusters of large size. It is not forthcoming to determine isomers of more than ten atoms with traditional methodology, such as DFT, *ab initio*, etc. PGA provides not only the global minima, but also other isomers close to the minima.

Current computational limitations make it unfeasible to use *ab initio* or DFT methods in extensive genetic algorithms (GA) searches, therefore we adopted the same recipe that proved to be successful in silicon clusters prediction.<sup>30,35</sup> We use a semiempirical molecular orbital program, MSINDO,<sup>36–38</sup> to evaluate the energy of the clusters in the global search using the GA, while a DFT method to refine those more promising ones by performing a local optimization of the best structures found by the GA. This global search assures that  $\text{Si}_6\text{O}_m$  isomers are produced from a full exploration of the MSINDO/DFT potential energy, without the limitations of DFT relaxed structures obtained by adding  $m$  oxygen atoms, one by one, to different  $\text{Si}_6$  isomers. In an attempt to understand better the electronic properties in the formation of these clusters,<sup>39,40</sup> we studied two series: (i)  $\text{Si}_6\text{O}_m$  ( $m=2,3,4,5,6$ ), which can be thought as small  $\text{Si}_n$  clusters plus small oxide fragments and (ii)  $\text{Si}_6\text{O}_m$  ( $m=9,10,11$ ), clusters rich in oxygen, which show that the more stable ones are those for which the silicon atoms are more coordinated to oxygen ones.

## II. METHODOLOGY

In order to determine the most stable isomers, a stochastic method which uses PGA has been used. Bazterra *et al.*<sup>31</sup>

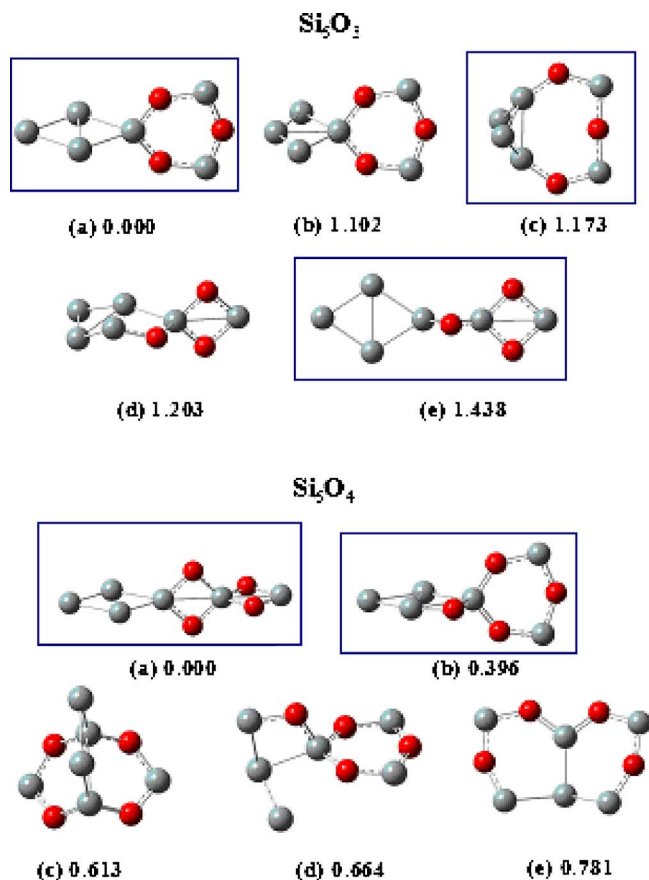


FIG. 2. (Color online) Optimized structures for the five lower-energy isomers for  $\text{Si}_6\text{O}_3$  and  $\text{Si}_6\text{O}_4$  clusters at the B3LYP/6-311G(2d) level of theory. BE in eV based on a Si atomic energy of  $-7847.211$  eV and O atomic energy of  $-2036.020$  eV. The structures similar to those reported in the literature are enclosed by frames.

implemented the MGAC package using parallel techniques. The coding method used here allows making it very portable as well as easy to maintain and upgrade. The cluster is represented by a genome of dimension  $3N$ , where  $N$  is the number of atoms in the cluster. Moreover, any genetic operator, mating, crossover, mutation, etc., applied to this genome produces a valid individual, i.e., a possible structure for the desired cluster size. The GA operations of mating, crossover, and also the “cut and slice” operator introduced by Johnston and Roberts<sup>41</sup> are used to evolve one generation into the next one. The best individuals among the population, 50% in our case, are directly copied into the next generation, and the other 50% is replaced applying the genetic operators. This technique is also known as elitism. The criteria for fitness probability, selection of the individuals, and genetic operators are discussed in detail in Ref. 29. GA procedure was run several times to guarantee that the resulting structures are independent of the initial population and statistically significant. We ran the MGAC about five times per  $\text{Si}_6\text{O}_m$  system, employing from 60 to 200 generations with 60 individuals. All the energy calculations were done with the MSINDO code.<sup>36–38</sup> The isomers selected by MGAC/MSINDO calculations were fully optimized using the DFT method with B3LYP exchange correlation functional, employing 6-311G(2d) basis sets. Vibration frequencies were calculated for the optimized structures to check that no imaginary fre-

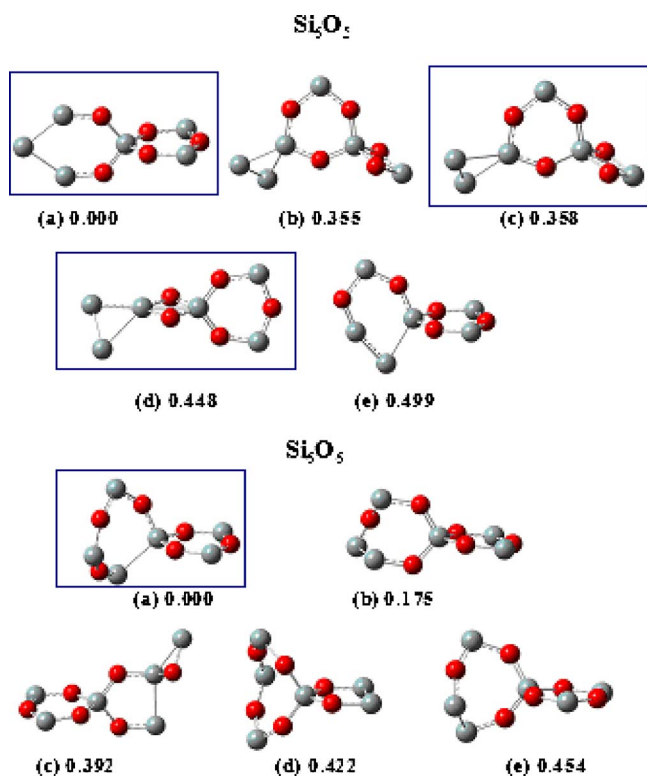


FIG. 3. (Color online) Optimized structures for the five lower-energy isomers for  $\text{Si}_6\text{O}_5$  and  $\text{Si}_6\text{O}_6$  clusters at the B3LYP/6-311G(2d) level of theory. BE in eV based on a Si atomic energy of  $-7847.211$  eV and O atomic energy of  $-2036.020$  eV. The structures similar to those reported in the literature are enclosed by frames.

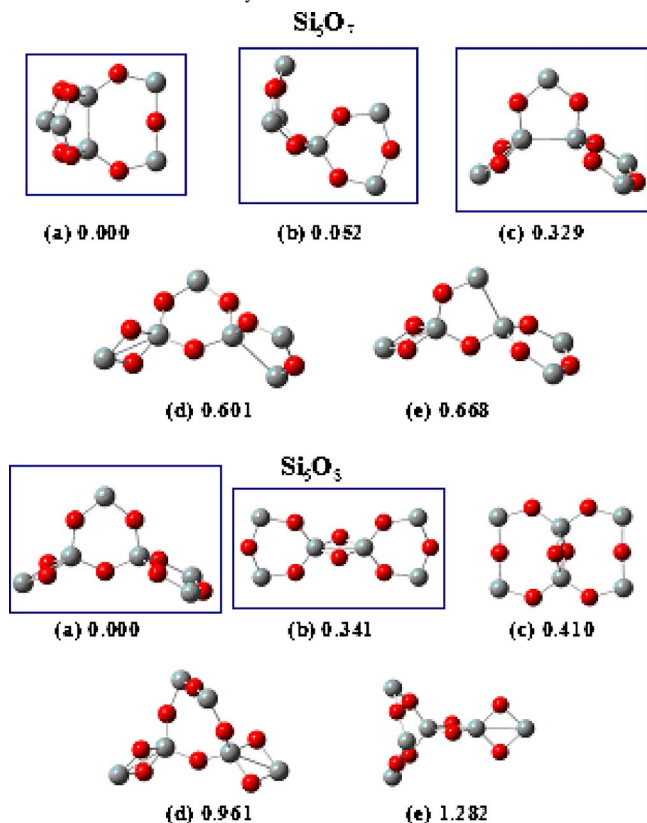


FIG. 4. (Color online) Optimized structures for the five lower-energy isomers for  $\text{Si}_6\text{O}_7$  and  $\text{Si}_6\text{O}_8$  clusters at the B3LYP/6-311G(2d) level of theory. BE in eV based on a Si atomic energy of  $-7847.211$  eV and O atomic energy of  $-2036.020$  eV. The structures similar to those reported in the literature are enclosed by frames.

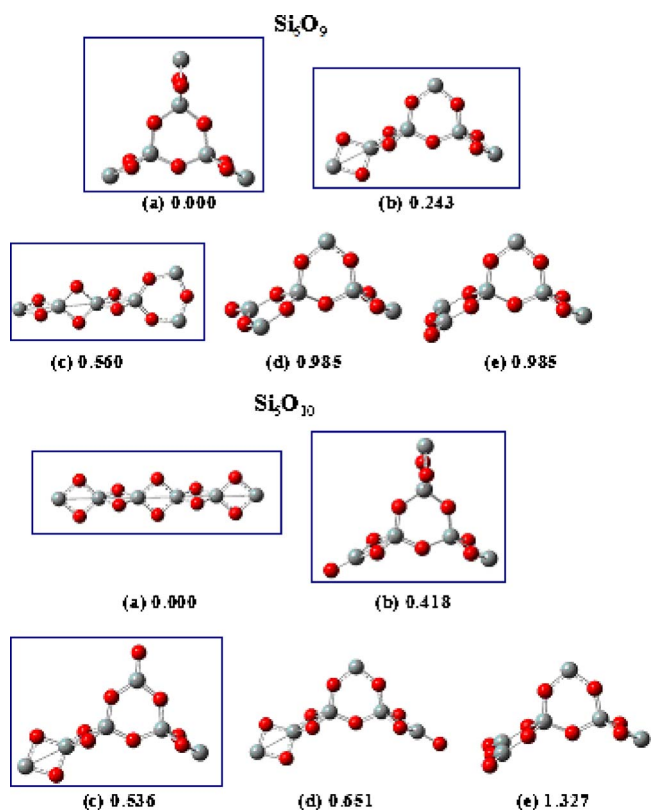


FIG. 5. (Color online) Optimized structures for the five lower-energy isomers for  $\text{Si}_6\text{O}_9$  and  $\text{Si}_6\text{O}_{10}$  clusters at the B3LYP/6-311G(2d) level of theory. BE in eV based on a Si atomic energy of  $-7847.211$  eV and O atomic energy of  $-2036.020$  eV. The structures similar to those reported in the literature are enclosed by frames.

quencies are present. All these calculations were done using the package of programs GAUSSIAN 98.<sup>42</sup>

### III. RESULTS AND DISCUSSION

Figures 1–6 show the stable structures of the five lowest-energy isomers for  $\text{Si}_6\text{O}_m$  ( $m=1-11$ ) clusters obtained by the MGAC/MSINDO prediction and followed by the

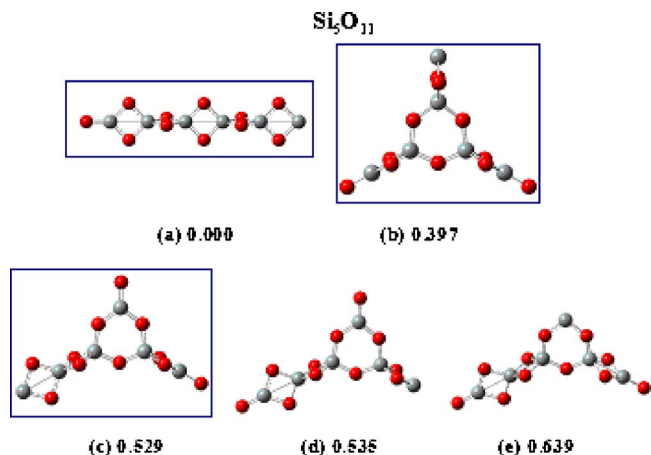


FIG. 6. (Color online) Optimized structures for the five lower-energy isomers for  $\text{Si}_6\text{O}_{11}$  clusters at the B3LYP/6-311G(2d) level of theory. BE in eV based on a Si atomic energy of  $-7847.211$  eV and O atomic energy of  $-2036.020$  eV. The structures similar to those reported in the literature are enclosed by frames.

TABLE I. rms between this work and the work of Chelikowsky (Ref. 17) 2MR, 3MRS, and Si<sub>3</sub>O<sub>2</sub> reported by Wang *et al.* (Ref. 14) geometry parameters, for Si<sub>6</sub>O<sub>m</sub> (*m*=2–11) clusters.

System	Fragment	rms		System	Fragment	rms	
		Distance (Å)	Angle (deg)			Distance (Å)	Angle (deg)
Si <sub>6</sub> O <sub>2</sub> - <i>a</i>	Si <sub>3</sub> O <sub>2</sub>	0.048	3.5	Si <sub>6</sub> O <sub>6</sub> - <i>c</i>	Si <sub>3</sub> O <sub>3</sub>	0.023	3.3
Si <sub>6</sub> O <sub>2</sub> - <i>b</i>	Si <sub>3</sub> O <sub>2</sub>	0.020	3.4		Si <sub>3</sub> O <sub>2</sub>	0.032	4.9
Si <sub>6</sub> O <sub>2</sub> - <i>c</i>	Si <sub>2</sub> O <sub>2</sub>	0.006	0.4	Si <sub>6</sub> O <sub>6</sub> - <i>d</i>	Si <sub>4</sub> O <sub>4</sub>	0.020	10.6
Si <sub>6</sub> O <sub>2</sub> - <i>e</i>	Si <sub>2</sub> O <sub>2</sub>	0.015	0.8		Si <sub>3</sub> O <sub>2</sub>	0.090	3.3
				Si <sub>6</sub> O <sub>6</sub> - <i>e</i>	Si <sub>3</sub> O <sub>3</sub>	0.022	3.8
Si <sub>6</sub> O <sub>3</sub> - <i>a</i>	Si <sub>3</sub> O <sub>3</sub>	0.013	2.5				
Si <sub>6</sub> O <sub>3</sub> - <i>b</i>	Si <sub>3</sub> O <sub>3</sub>	0.007	2.2	Si <sub>6</sub> O <sub>7</sub> - <i>b</i>	Si <sub>3</sub> O <sub>3</sub>	0.020	3.5
Si <sub>6</sub> O <sub>3</sub> - <i>d</i>	Si <sub>2</sub> O <sub>2</sub>	0.022	1.4		Si <sub>4</sub> O <sub>4</sub>	0.018	9.9
Si <sub>6</sub> O <sub>3</sub> - <i>e</i>	Si <sub>2</sub> O <sub>2</sub>	0.023	1.3	Si <sub>6</sub> O <sub>7</sub> - <i>c</i>	Si <sub>2</sub> O <sub>2</sub>	0.020	1.3
					Si <sub>3</sub> O <sub>3</sub>	0.018	3.3
Si <sub>6</sub> O <sub>4</sub> - <i>a</i>	Si <sub>2</sub> O <sub>2</sub> -first	0.030	2.2		Si <sub>3</sub> O <sub>2</sub>	0.061	2.8
	Si <sub>2</sub> O <sub>2</sub> -second	0.032	1.9	Si <sub>6</sub> O <sub>7</sub> - <i>d</i>	Si <sub>2</sub> O <sub>2</sub>	0.027	1.7
Si <sub>6</sub> O <sub>4</sub> - <i>b</i>	Si <sub>3</sub> O <sub>3</sub>	0.018	3.4		Si <sub>3</sub> O <sub>3</sub>	0.029	4.1
Si <sub>6</sub> O <sub>4</sub> - <i>c</i>	Si <sub>3</sub> O <sub>2</sub>	0.028	5.3		Si <sub>3</sub> O <sub>2</sub>	0.017	3.7
Si <sub>6</sub> O <sub>4</sub> - <i>d</i>	Si <sub>3</sub> O <sub>3</sub>	0.013	2.6	Si <sub>6</sub> O <sub>7</sub> - <i>e</i>	Si <sub>2</sub> O <sub>2</sub>	0.029	1.8
					Si <sub>3</sub> O <sub>3</sub>	0.012	2.8
Si <sub>6</sub> O <sub>5</sub> - <i>a</i>	Si <sub>3</sub> O <sub>3</sub>	0.025	3.6		Si <sub>3</sub> O <sub>2</sub>	0.031	5.1
Si <sub>6</sub> O <sub>5</sub> - <i>b</i>	Si <sub>2</sub> O <sub>2</sub>	0.031	1.9				
	Si <sub>3</sub> O <sub>3</sub>	0.028	3.5	Si <sub>6</sub> O <sub>8</sub> - <i>a</i>	Si <sub>2</sub> O <sub>2</sub>	0.028	1.8
Si <sub>6</sub> O <sub>5</sub> - <i>c</i>	Si <sub>2</sub> O <sub>2</sub>	0.030	1.9		Si <sub>3</sub> O <sub>3</sub> -first	0.036	4.5
	Si <sub>3</sub> O <sub>3</sub>	0.029	3.2		Si <sub>3</sub> O <sub>3</sub> -second	0.025	3.6
Si <sub>6</sub> O <sub>5</sub> - <i>d</i>	Si <sub>2</sub> O <sub>2</sub>	0.026	1.8	Si <sub>6</sub> O <sub>8</sub> - <i>b</i>	Si <sub>2</sub> O <sub>2</sub>	0.035	2.9
	Si <sub>3</sub> O <sub>3</sub>	0.028	3.7		Si <sub>3</sub> O <sub>3</sub> -first	0.027	3.3
Si <sub>6</sub> O <sub>5</sub> - <i>e</i>	Si <sub>3</sub> O <sub>3</sub>	0.013	2.5		Si <sub>3</sub> O <sub>3</sub> -second	0.027	3.3
				Si <sub>6</sub> O <sub>8</sub> - <i>c</i>	Si <sub>2</sub> O <sub>2</sub>	0.032	2.7
Si <sub>6</sub> O <sub>6</sub> - <i>a</i>	Si <sub>3</sub> O <sub>3</sub>	0.009	2.8	Si <sub>6</sub> O <sub>8</sub> - <i>d</i>	Si <sub>2</sub> O <sub>2</sub> -first	0.028	1.8
Si <sub>6</sub> O <sub>6</sub> - <i>b</i>	Si <sub>3</sub> O <sub>3</sub>	0.021	3.1		Si <sub>2</sub> O <sub>2</sub> -second	0.027	1.8
	Si <sub>3</sub> O <sub>2</sub>	0.070	4.2	Si <sub>6</sub> O <sub>8</sub> - <i>e</i>	Si <sub>2</sub> O <sub>2</sub> -first	0.034	2.7
					Si <sub>2</sub> O <sub>2</sub> -second	0.032	1.8
					Si <sub>4</sub> O <sub>4</sub>	0.024	11.8
Si <sub>6</sub> O <sub>9</sub> - <i>a</i>	Si <sub>2</sub> O <sub>2</sub> -first	0.032	2.0	Si <sub>6</sub> O <sub>10</sub> - <i>d</i>	Si <sub>2</sub> O <sub>2</sub> -first	0.035	2.0
	Si <sub>2</sub> O <sub>2</sub> -second	0.032	2.0		Si <sub>2</sub> O <sub>2</sub> -second	0.038	2.7
	Si <sub>2</sub> O <sub>2</sub> -third	0.032	2.0		Si <sub>2</sub> O <sub>2</sub> -third	0.036	2.5
	Si <sub>3</sub> O <sub>3</sub>	0.046	5.1		Si <sub>3</sub> O <sub>3</sub>	0.041	4.6
Si <sub>6</sub> O <sub>9</sub> - <i>b</i>	Si <sub>2</sub> O <sub>2</sub> -first	0.037	2.7	Si <sub>6</sub> O <sub>10</sub> - <i>e</i>	Si <sub>2</sub> O <sub>2</sub>	0.033	2.0
	Si <sub>2</sub> O <sub>2</sub> -second	0.032	1.9		Si <sub>3</sub> O <sub>3</sub> -first	0.045	4.3
	Si <sub>2</sub> O <sub>2</sub> -third	0.031	2.2		Si <sub>3</sub> O <sub>3</sub> -second	0.040	4.9
	Si <sub>3</sub> O <sub>3</sub>	0.038	4.3				
Si <sub>6</sub> O <sub>9</sub> - <i>c</i>	Si <sub>2</sub> O <sub>2</sub> -first	0.038	2.8	Si <sub>6</sub> O <sub>11</sub> - <i>a</i>	Si <sub>2</sub> O <sub>2</sub> -first	0.036	2.4
	Si <sub>2</sub> O <sub>2</sub> -second	0.037	2.6		Si <sub>2</sub> O <sub>2</sub> -second	0.040	2.7
	Si <sub>2</sub> O <sub>2</sub> -third	0.034	2.0		Si <sub>2</sub> O <sub>2</sub> -third	0.040	2.7
	Si <sub>3</sub> O <sub>3</sub>	0.030	3.7		Si <sub>2</sub> O <sub>2</sub> -fourth	0.039	2.7
Si <sub>6</sub> O <sub>9</sub> - <i>d</i>	Si <sub>3</sub> O <sub>3</sub> -first	0.037	4.7		Si <sub>2</sub> O <sub>2</sub> -fifth	0.037	2.1
	Si <sub>3</sub> O <sub>3</sub> -second	0.037	3.9	Si <sub>6</sub> O <sub>11</sub> - <i>b</i>	Si <sub>2</sub> O <sub>2</sub> -first	0.036	2.5
	Si <sub>2</sub> O <sub>2</sub>	0.031	1.8		Si <sub>2</sub> O <sub>2</sub> -second	0.036	2.5
Si <sub>6</sub> O <sub>9</sub> - <i>e</i>	Si <sub>3</sub> O <sub>3</sub> -first	0.037	4.6		Si <sub>2</sub> O <sub>2</sub> -third	0.039	2.3
	Si <sub>3</sub> O <sub>3</sub> -second	0.037	3.9		Si <sub>3</sub> O <sub>3</sub>	0.049	5.3
	Si <sub>2</sub> O <sub>2</sub>	0.031	1.9	Si <sub>6</sub> O <sub>11</sub> - <i>c</i>	Si <sub>2</sub> O <sub>2</sub> -first	0.038	2.1
					Si <sub>2</sub> O <sub>2</sub> -second	0.039	2.8
Si <sub>6</sub> O <sub>10</sub> - <i>a</i>	Si <sub>2</sub> O <sub>2</sub> -first	0.035	2.0		Si <sub>2</sub> O <sub>2</sub> -third	0.037	2.5
	Si <sub>2</sub> O <sub>2</sub> -second	0.038	2.6		Si <sub>3</sub> O <sub>3</sub>	0.048	4.7
	Si <sub>2</sub> O <sub>2</sub> -third	0.039	2.6	Si <sub>6</sub> O <sub>11</sub> - <i>d</i>	Si <sub>2</sub> O <sub>2</sub> -first	0.036	2.4
	Si <sub>2</sub> O <sub>2</sub> -fourth	0.038	2.6		Si <sub>2</sub> O <sub>2</sub> -second	0.040	2.9
	Si <sub>2</sub> O <sub>2</sub> -fifth	0.035	2.0		Si <sub>2</sub> O <sub>2</sub> -third	0.039	2.3
Si <sub>6</sub> O <sub>10</sub> - <i>b</i>	Si <sub>2</sub> O <sub>2</sub> -first	0.036	2.5		Si <sub>3</sub> O <sub>3</sub>	0.047	4.7
	Si <sub>2</sub> O <sub>2</sub> -second	0.035	2.1	Si <sub>6</sub> O <sub>11</sub> - <i>e</i>	Si <sub>2</sub> O <sub>2</sub> -first	0.036	2.5
	Si <sub>2</sub> O <sub>2</sub> -third	0.035	2.1		Si <sub>2</sub> O <sub>2</sub> -second	0.040	2.9

TABLE I. (Continued.)

System	Fragment	rms		System	Fragment	rms	
		Distance (Å)	Angle (deg)			Distance (Å)	Angle (deg)
Si <sub>6</sub> O <sub>10</sub> -c	Si <sub>3</sub> O <sub>3</sub>	0.047	5.2	Si <sub>2</sub> O <sub>2</sub> -third	0.036	2.4	
	Si <sub>2</sub> O <sub>2</sub> -first	0.035	2.0		Si <sub>3</sub> O <sub>3</sub>	0.042	4.7
	Si <sub>2</sub> O <sub>2</sub> -second	0.035	2.8				
	Si <sub>2</sub> O <sub>2</sub> -third	0.036	2.2				
	Si <sub>3</sub> O <sub>3</sub>	0.046	4.6				

B3LYP/6-311G(2d) scheme for their final local minimization and their corresponding binding energies (BE) per atom, with respect to the most stable structure for each system. Energies in eV are based on a Si atomic energy of  $-7847.211$  eV and an O atomic energy of  $-2036.020$  eV. The isomers enclosed into frames are those that have been previously reported in the literature.<sup>39,40</sup> The ordering of the energies for the Si<sub>6</sub>O<sub>m</sub> isomers might be different if they are evaluated by MSINDO method because semiempirical methods do not include correlation.

Although we found new structures such as the Si<sub>6</sub>O-a1 isomer, which is the mirror structure of the Si<sub>6</sub>O-a2 isomer, in most cases the structures agree with those reported by Zang *et al.*<sup>39</sup> Si-rich clusters have a tendency to form two fragments, one made of pure silicon atoms and the other made of silicon oxides, Si<sub>2</sub>O<sub>2</sub> (2MR), Si<sub>3</sub>O<sub>3</sub> (3MR), and Si<sub>4</sub>O<sub>4</sub> (4MR, 2MR).<sup>40</sup> On the contrary, for O-rich clusters the isomers prefer the hybrid combinations of 2MR and 3MR.

We inform in Table I the rms between the geometrical parameters of our predicted 2MR, 3MR, 4MR, and those reported by Chelikowsky.<sup>17</sup> The rms of bond distances and bond angles are about  $0.02$  Å and  $3^\circ$ , respectively, in motifs 2MR or 3MR. The greater distortions can be found in the angles, as it happens, for example, for the 3MR present in the system Si<sub>6</sub>O<sub>11</sub>-b ( $5.3^\circ$ ) and for the 2MR of the systems Si<sub>6</sub>O<sub>8</sub> and Si<sub>6</sub>O<sub>11</sub>-d ( $2.9^\circ$ ). In both cases, all the oxygen atoms that form the fragment in question show hypervalent

bonds. These distortions are smaller than those which run from  $9.9^\circ$  to  $11.8^\circ$  for the 4MR present in Si<sub>6</sub>O<sub>6</sub>-d, Si<sub>6</sub>O<sub>7</sub>-b, and Si<sub>6</sub>O<sub>8</sub>-e clusters. Si-rich clusters, Si<sub>6</sub>O<sub>2</sub>, Si<sub>6</sub>O<sub>4</sub>, Si<sub>6</sub>O<sub>6</sub>, and Si<sub>6</sub>O<sub>7</sub>, are formed by Si<sub>3</sub>O<sub>2</sub> molecules. We include in Table I the rms between the geometrical parameters of this motif and that reported by Wang *et al.*<sup>14</sup> In this case the rms of bond distances and bond angles are about  $0.05$  Å and  $4^\circ$ . To sum up, we could say that the motif of 2MR and 3MR are the ones that preserve their distances and their angles in the different systems. At the same time we point out that every time that the system counts with the necessary number of oxygen and silicon atoms to form the motifs 2MR and 3MR, these combinations appear in the cluster. This finding was highlighted by Zhang *et al.*,<sup>24</sup> “These 2MR and 3MR hybrid clusters would be considered as energetically favorable structural models of silica nanoclusters.”

In Fig. 7, we reported the moment of total inertia per its maximum component in a function of the number of oxygen atoms for the most stable isomer of each cluster. It is observed that the structures are mainly prolate (rugby football, or cigar shaped), except for the isomer Si<sub>6</sub>O<sub>7</sub>-a, which is more compact, and the isomer Si<sub>6</sub>O<sub>9</sub>-a, which is more oblate (saucer or disk-shaped). For  $m \geq 4$ , we can see that the most stable structures present Si atoms with four-coordination by oxygen atom, except for the systems Si<sub>6</sub>O<sub>6</sub> and Si<sub>6</sub>O<sub>7</sub>, which present three-coordination by oxygen atom and one-

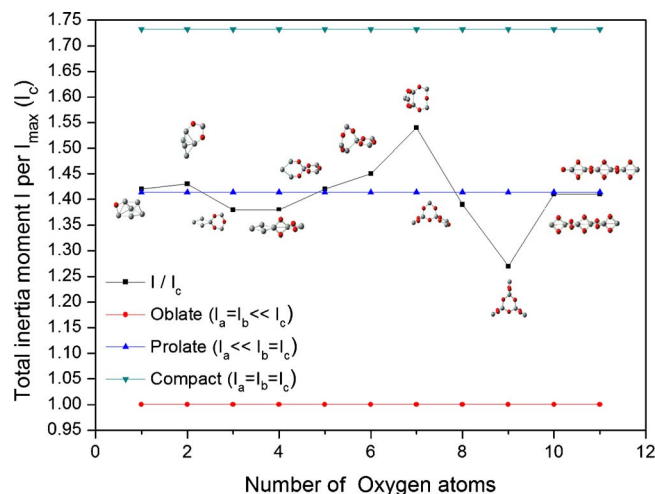


FIG. 7. (Color online) The moment of total inertia  $I$  per its maximum component  $I_c$  in function of the number  $m$  of oxygen atoms for the most stable isomer for each Si<sub>6</sub>O<sub>m</sub> system.

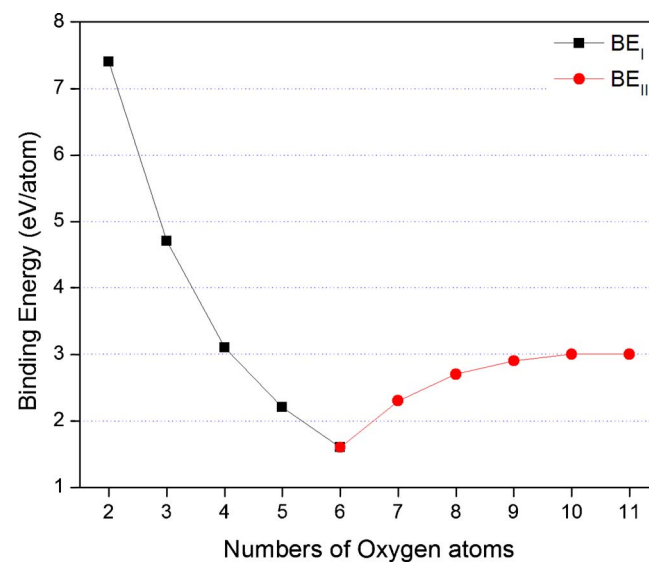
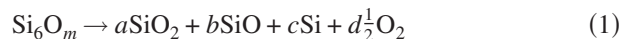


FIG. 8. (Color online) BE per oxygen atom of the lowest-energy isomers for Si<sub>6</sub>O<sub>m</sub> ( $m=2-11$ ) clusters [see Eq. (7)].

coordination by silicon atom. For silicon-oxide systems, silicon atoms prefer to form tetrahedral  $sp^3$  hybrid bonds to neighboring oxygen atoms, as it is mentioned by Zhang and Fan<sup>40</sup> We also study the BE/atom for the most stable clusters found in this work as a function of the number of oxygen atoms.

$\text{Si}_6\text{O}_m$  clusters might be dissociated into silicon oxides, silicon atoms, and oxygen.



requires an energy,

$$\begin{aligned} \text{BE} = & E(\text{Si}_6\text{O}_m) - aE(\text{SiO}_2) - bE(\text{SiO}) - cE(\text{Si}) \\ & - d\frac{1}{2}E(\text{O}_2). \end{aligned} \quad (2)$$

We are interested in the preferred dissociation channel, that is, the determination of  $a$ ,  $b$ ,  $c$ , and  $d$  values that, for each  $m$ , minimize BE. Conservation of the number of silicon and oxygen atoms requires  $a+b+c=6$  and  $2a+b+d=m$ , respectively. Furthermore, defining the two positive quantities

$$E_x = 2E(\text{SiO}_2) + E(\text{Si}) - E(\text{SiO}), \quad (3)$$

$$E_y = E(\text{SiO}) + \frac{1}{2}E(\text{O}_2) - E(\text{SiO}_2), \quad (4)$$

we can rewrite Eq. (2) as

$$\begin{aligned} \text{BE} = & \text{Si}_6\text{O}_m + (6-c)E_x + (m-d)E_y - 6E(\text{Si}) \\ & - m\frac{1}{2}E(\text{O}_2). \end{aligned} \quad (5)$$

Since  $E_x$  and  $E_y$  are positive, maximization of BE requires minimization of  $c$  and  $d$  subject to the  $b$ ,  $c$ , and  $d > 0$ . Our results are

$$\begin{aligned} m < 6 & \rightarrow a = 0 \quad b = m \quad c = 6 - m \quad d = 0, \\ 6 < m < 12 & \rightarrow a = m - 6 \quad b = 12 - m \quad c = 0 \quad d = 0, \end{aligned} \quad (6)$$

$$12 < m \rightarrow a = 6 \quad b = 0 \quad c = 0 \quad d = m - 12.$$

In Fig. 8, we display the tendencies for the proposed dissociation channels for the most stable clusters found in this work as a function of the number of oxygen atoms.

$$\text{BE}_I = [E(\text{Si}_6\text{O}_m) - mE(\text{SiO}) - (6-m)E(\text{Si})]/m \quad m \leq 6,$$

$$\begin{aligned} \text{BE}_{II} = & [E(\text{Si}_6\text{O}_m) - (m-6)E(\text{SiO}_2) \\ & - (12-m)E(\text{SiO})]/m \quad 6 < m \leq 12, \end{aligned} \quad (7)$$

where  $E$  is the total energy of the cluster or the atom indicated between brackets. These plots agree with our results above, see Eq. (6). So, it is interesting to note that for Si-rich clusters it is easy to dissociate them into SiO pieces and Si atoms ( $\text{BE}_I$ ). On the contrary, the behavior is different for O-rich clusters because they prefer dissociation into several pieces of SiO and  $\text{SiO}_2$  ( $\text{BE}_{II}$ ).  $\text{BE}_I$  diminishes with  $m$ , indicating the increment in the number of (SiO) pieces, and  $\text{BE}_{II}$  grows with  $m$ , showing that the number  $\text{SiO}_2$  pieces while the number of SiO pieces diminishes accordingly. This is additional evidence that SiO is more stable than  $\text{SiO}_2$ . Figure 8 shows that the best  $\text{Si}_6\text{O}_6$  isomer is the less stable among these silicon-oxide systems, as we can see from the  $\text{BE}_I$

curve. This observation was also reported by Zang *et al.*<sup>39</sup>

We study some fragmentation pathways and the corresponding energies for the most stable structures of the series  $\text{Si}_6\text{O}_m$  ( $m=1-11$ ) produced by MGAC/MSINDO/B3LYP methodology. Basically we choose those fragmentation channels formed by motifs that frequently appear in silicon-oxides studies ( $\text{Si}_2\text{O}_2$ ,  $\text{Si}_3\text{O}_3$ ,  $\text{Si}_3\text{O}_2$ , and  $\text{Si}_4\text{O}_4$ ) and core silicon clusters  $\text{Si}_n$  ( $n=2-5$ ). Furthermore, some fragmentation products contain SiO and  $\text{SiO}_2$  molecules.

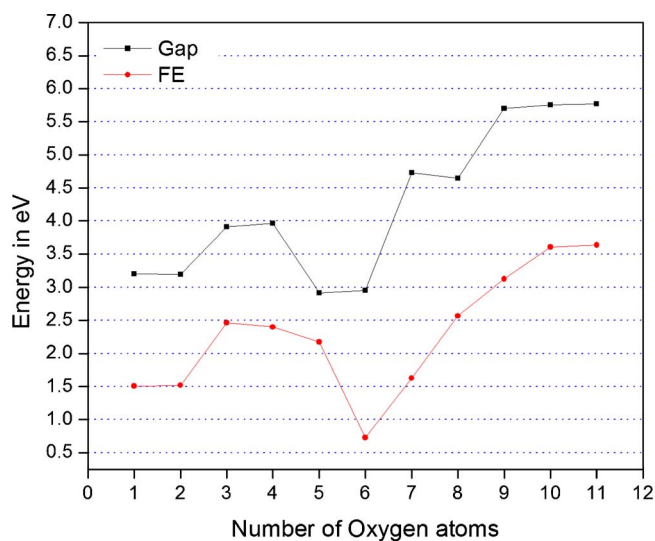
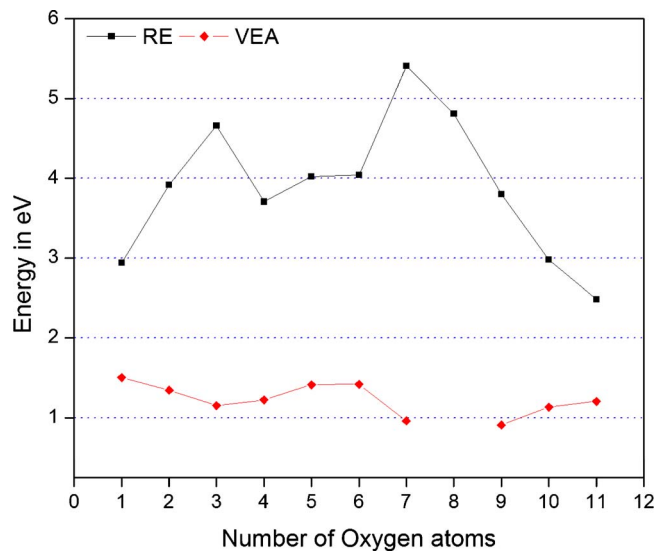
The energy for the fragmentation pathway  $\text{Si}_6\text{O}_m \rightarrow \text{Si}_k\text{O}_l + \text{Si}_{6-k}\text{O}_{m-l}$  is defined as  $\text{FE} = E(\text{Si}_k\text{O}_l) + E(\text{Si}_{6-k}\text{O}_{m-l}) - E(\text{Si}_6\text{O}_m)$  and the results are shown in Table II. The geometry of some of the fragmentation products,  $\text{Si}_k\text{O}_l$  and/or  $\text{Si}_{6-k}\text{O}_{m-l}$ , employed to evaluate the FEs were reported previously by Reber *et al.* and Lu *et al.*<sup>28,51</sup> For  $\text{Si}_6\text{O}_{10}$  and  $\text{Si}_6\text{O}_{11}$  we propose alternative fragmentation pathways to be taken into account and our minimum is discussed in the following:

- (1)  $\text{Si}_6\text{O}$ ,  $\text{Si}_6\text{O}_2$ , and  $\text{Si}_6\text{O}_3$ : It is interesting to note that Si-rich clusters segregate in small Si cores,  $\text{Si}_5$ ,  $\text{Si}_4$  and  $\text{Si}_3$  and SiO,  $\text{Si}_2\text{O}_2$  and  $\text{Si}_3\text{O}_3$  oxides, respectively. These channels agree with studies reported by Zang *et al.*<sup>39</sup> Wang *et al.*<sup>43</sup> also found that  $\text{Si}_n\text{O}$  has a tendency to segregate in  $\text{Si}_n + \text{SiO}$ . They also found recently<sup>25</sup> that some silicon monoxide clusters prefer to form nearly side polygonal Si clusters attached to monoxide silicon clusters. We found a similar behavior for Si-rich clusters,  $\text{Si}_6\text{O}$ ,  $\text{Si}_6\text{O}_2$ ,  $\text{Si}_6\text{O}_3$ , and some  $\text{Si}_6\text{O}_4$ , displayed in Fig. 1.
- (2)  $\text{Si}_6\text{O}_4$ : In this case the most favorable fragmentation channel is  $\text{Si}_6\text{O}_4 \rightarrow \text{Si}_3 + \text{Si}_3\text{O}_4$ , but its FE is only 0.04 eV smaller than the corresponding to the dissociation  $\text{Si}_6\text{O}_4 \rightarrow \text{Si}_3\text{O}_3 + \text{Si}_3\text{O}$ .
- (3)  $\text{Si}_6\text{O}_5$  and  $\text{Si}_6\text{O}_6$ : Both cases contain the  $\text{Si}_3\text{O}_3$  as one fragmentation product in the most favorable pathway. However, the energy fragmentation corresponding to the channel with a SiO fragment,  $\text{Si}_6\text{O}_5 \rightarrow \text{Si}_5\text{O}_4 + \text{SiO}$  and  $\text{Si}_6\text{O}_6 \rightarrow \text{Si}_5\text{O}_5 + \text{SiO}$ , are higher by only 0.383 and 0.894 eV relative to the minimum FE, 2.177 and 0.729 eV, respectively. This observation is important because SiO is a very stable system.<sup>25</sup>
- (4)  $\text{Si}_6\text{O}_7$  to  $\text{Si}_6\text{O}_9$ : The most favorable pathway is  $\text{Si}_6\text{O}_m \rightarrow \text{SiO} + \text{Si}_5\text{O}_{m-1}$ .  $\text{Si}_6\text{O}_7$  and  $\text{Si}_6\text{O}_8$  show also a favorable dissociation into  $\text{Si}_2\text{O}_2 + \text{Si}_4\text{O}_5$  and  $\text{Si}_2\text{O}_2 + \text{Si}_4\text{O}_6$ , respectively.
- (5)  $\text{Si}_6\text{O}_{10}$  and  $\text{Si}_6\text{O}_{11}$ : In both cases, the new fragmentation pathways proposed here present energies higher than those reported by Zang *et al.*<sup>39</sup> However our most stable channels are those reported by Zang *et al.*<sup>39</sup>

Figure 9 shows a similar behavior for the LUMO-HOMO gap and the fragmentation energy against the number of oxygen atoms for the most stable isomer of the series  $\text{Si}_6\text{O}_m$  ( $m=1-11$ ), except  $\text{Si}_6\text{O}_5$ . Clusters with high minimum fragmentation energy are particularly stable while large LUMO-HOMO gaps are indicative of chemical inertness.<sup>44</sup>  $\text{Si}_6\text{O}_6$  is the less stable isomer for both criteria. The minimum energy fragmentation contains  $\text{Si}_2\text{O}_2$ ,  $\text{Si}_3\text{O}_3$ , and SiO clusters, so these fragmentation products are considered to be

TABLE II. Fragmentation Energies (FE) and fragmentation channels for the most stable Si<sub>6</sub>O<sub>m</sub> ( $m=1-11$ ) clusters found by the MGAC/MSINDO/B3LYP method.

Si <sub>6</sub> O <sub>m</sub>	→Si <sub>k</sub> O <sub>l</sub>	+Si <sub>6-k</sub> O <sub>m-l</sub>	FE (eV)	Si <sub>6</sub> O <sub>m</sub>	→Si <sub>k</sub> O <sub>l</sub>	+Si <sub>6-k</sub> O <sub>m-l</sub>	FE (eV)
Si <sub>6</sub> O	Si <sub>5</sub> <sup>a</sup>	SiO	1.512	Si <sub>6</sub> O <sub>7</sub>	SiO	Si <sub>5</sub> O <sub>6</sub> <sup>b</sup>	1.629
	Si <sub>4</sub> <sup>a</sup>	Si <sub>2</sub> O <sup>b</sup>	2.793		Si <sub>2</sub> O <sub>2</sub>	Si <sub>4</sub> O <sub>5</sub> <sup>b</sup>	2.088
	Si <sub>3</sub>	Si <sub>3</sub> O <sup>b</sup>	3.838		Si <sub>3</sub> O <sub>3</sub>	Si <sub>3</sub> O <sub>4</sub> <sup>b</sup>	2.196
Si <sub>6</sub> O <sub>2</sub>	Si <sub>4</sub> <sup>a</sup>	Si <sub>2</sub> O <sub>2</sub>	1.520	Si <sub>4</sub> O <sub>4</sub>	Si <sub>2</sub> O <sub>3</sub> <sup>b</sup>	4.149	
	Si <sub>3</sub>	Si <sub>3</sub> O <sub>2</sub>	3.466	Si <sub>3</sub> O <sub>2</sub>	Si <sub>3</sub> O <sub>2</sub>	5.122	
	Si <sub>5</sub> <sup>a</sup>	SiO <sub>2</sub> <sup>b</sup>	3.806	SiO <sub>2</sub>	Si <sub>5</sub> O <sub>5</sub> <sup>b</sup>	5.435	
Si <sub>6</sub> O <sub>3</sub>	SiO	Si <sub>5</sub> O <sub>2</sub>	2.466	Si <sub>6</sub> O <sub>8</sub>	SiO	Si <sub>5</sub> O <sub>7</sub> <sup>b</sup>	2.569
	Si <sub>3</sub>	Si <sub>3</sub> O <sub>3</sub>	2.642		Si <sub>2</sub> O <sub>2</sub>	Si <sub>4</sub> O <sub>6</sub> <sup>b</sup>	3.327
	Si <sub>4</sub> <sup>a</sup>	Si <sub>2</sub> O <sub>3</sub> <sup>b</sup>	3.297		Si <sub>3</sub> O <sub>3</sub>	Si <sub>3</sub> O <sub>5</sub> <sup>b</sup>	4.447
	Si <sub>3</sub> O	Si <sub>3</sub> O <sub>2</sub>	4.223		SiO <sub>2</sub>	Si <sub>5</sub> O <sub>6</sub> <sup>b</sup>	4.817
Si <sub>6</sub> O <sub>4</sub>	Si <sub>3</sub>	Si <sub>3</sub> O <sub>4</sub> <sup>b</sup>	2.403	Si <sub>4</sub> O <sub>4</sub>	Si <sub>2</sub> O <sub>4</sub> <sup>b</sup>	6.593	
	Si <sub>3</sub> O	Si <sub>3</sub> O <sub>3</sub>	2.443	Si <sub>3</sub> O <sub>2</sub>	Si <sub>3</sub> O <sub>6</sub> <sup>b</sup>	7.510	
	Si <sub>3</sub> O <sub>2</sub>	Si <sub>3</sub> O <sub>2</sub>	3.639	Si <sub>6</sub> O <sub>9</sub>	SiO	Si <sub>5</sub> O <sub>8</sub> <sup>b</sup>	3.124
	Si <sub>2</sub>	Si <sub>4</sub> O <sub>4</sub>	3.915		Si <sub>2</sub> O <sub>2</sub>	Si <sub>4</sub> O <sub>7</sub> <sup>b</sup>	4.612
	Si <sub>4</sub> <sup>a</sup>	Si <sub>2</sub> O <sub>4</sub> <sup>b</sup>	4.636		SiO <sub>2</sub> <sup>b</sup>	Si <sub>5</sub> O <sub>7</sub> <sup>b</sup>	4.742
Si <sub>6</sub> O <sub>5</sub>	Si <sub>3</sub> O <sub>3</sub>	Si <sub>3</sub> O <sub>2</sub>	2.177	Si <sub>4</sub> O <sub>5</sub> <sup>b</sup>	Si <sub>2</sub> O <sub>4</sub> <sup>b</sup>	5.443	
	SiO	Si <sub>5</sub> O <sub>4</sub> <sup>b</sup>	2.560	Si <sub>3</sub> O <sub>3</sub>	Si <sub>3</sub> O <sub>6</sub> <sup>b</sup>	5.821	
	Si <sub>4</sub> O <sub>4</sub>	Si <sub>2</sub> O <sup>b</sup>	2.774	Si <sub>4</sub> O <sub>4</sub>	Si <sub>2</sub> O <sub>5</sub> <sup>b</sup>	10.423	
	Si <sub>2</sub>	Si <sub>4</sub> O <sub>5</sub> <sup>b</sup>	2.992	Si <sub>6</sub> O <sub>10</sub>	SiO	Si <sub>5</sub> O <sub>9</sub> <sup>c</sup>	3.606
	Si <sub>3</sub>	Si <sub>3</sub> O <sub>5</sub> <sup>b</sup>	3.867		SiO <sub>2</sub> <sup>b</sup>	Si <sub>5</sub> O <sub>8</sub> <sup>b</sup>	4.475
Si <sub>4</sub> <sup>a</sup>	Si <sub>2</sub> O <sub>5</sub> <sup>b</sup>	8.693	Si <sub>3</sub> O <sub>5</sub> <sup>b</sup>		Si <sub>3</sub> O <sub>5</sub> <sup>b</sup>	4.713	
Si <sub>6</sub> O <sub>6</sub>	Si <sub>3</sub> O <sub>3</sub>	Si <sub>3</sub> O <sub>3</sub>	0.729	Si <sub>4</sub> O <sub>6</sub> <sup>b</sup>	Si <sub>2</sub> O <sub>4</sub> <sup>b</sup>	4.845	
	Si <sub>2</sub> O <sub>2</sub>	Si <sub>4</sub> O <sub>4</sub>	1.623	Si <sub>3</sub> O <sub>4</sub> <sup>b</sup>	Si <sub>3</sub> O <sub>6</sub> <sup>b</sup>	4.850	
	SiO	Si <sub>5</sub> O <sub>5</sub> <sup>b</sup>	1.646	Si <sub>6</sub> O <sub>11</sub>	SiO	Si <sub>5</sub> O <sub>10</sub>	3.638
	Si <sub>3</sub> O <sub>2</sub>	Si <sub>3</sub> O <sub>4</sub> <sup>b</sup>	2.270		Si <sub>4</sub> O <sub>7</sub> <sup>b</sup>	Si <sub>2</sub> O <sub>4</sub> <sup>b</sup>	4.816
	SiO <sub>2</sub> <sup>b</sup>	Si <sub>5</sub> O <sub>4</sub> <sup>b</sup>	4.975		Si <sub>5</sub> O <sub>8</sub> <sup>b</sup>	SiO <sub>3</sub> <sup>b</sup>	7.223
				Si <sub>4</sub> O <sub>6</sub> <sup>b</sup>	Si <sub>2</sub> O <sub>5</sub> <sup>b</sup>	7.360	

<sup>a</sup>Geometries taken from Ref. 30.<sup>b</sup>Geometries taken from Ref. 28.<sup>c</sup>Geometries taken from Ref. 51.FIG. 9. (Color online) LUMO-HOMO gaps and minimum fragmentation energy FE for Si<sub>6</sub>O<sub>m</sub> ( $m=1-11$ ) clusters.FIG. 10. (Color online) Energy required to remove one oxygen RE and the VEA for Si<sub>6</sub>O<sub>m</sub> ( $m=1-11$ ) clusters.

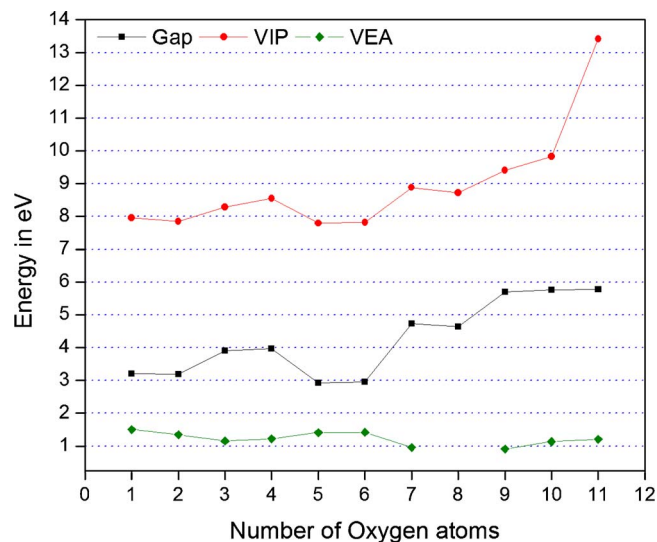


FIG. 11. (Color online) LUMO-HOMO gap, the vertical ionization potential VIP and the VEA for  $\text{Si}_6\text{O}_m$  ( $m=1-11$ ) clusters.

relatively stable clusters. These results are in good agreement with the experimental data reported by Rebert *et al.*<sup>28</sup>

We define RE as the energy required to remove one oxygen atom, that is  $\text{RE} = E(\text{Si}_6\text{O}_{m-1}) + \frac{1}{2}E(\text{O}_2) - E(\text{Si}_6\text{O}_m)$ . RE is either the energy required to remove one oxygen atom from  $\text{Si}_6\text{O}_m$  or the incremental energy on adding one oxygen atom to the  $\text{Si}_6\text{O}_{m-1}$  system. Because of that, the peaks observed in Fig. 10 are an indication of the cluster stability.  $\text{Si}_6\text{O}_3$  for the Si-rich clusters and  $\text{Si}_6\text{O}_7$  for the O-rich clusters are the most stable systems. For these systems the vertical electron affinity (VEA) reaches its corresponding minima, plotted also in Fig. 10, against the number of oxygen atoms. These peaks have a correspondence with the minimum ones presented in the VEAs.

The vertical ionization potential (VIP), VEA, and LUMO-HOMO gap for the most stable  $\text{Si}_6\text{O}_m$  ( $m=1-11$ ) isomers are plotted in Fig. 11 as a function of  $m$ . The GAP is between 3.19 and 3.96 eV for Si-rich systems ( $\text{Si}_6\text{O}_m, m=1-4$ ) and 2.92 and 5.77 eV for O-rich systems, and the VIPs are approximately 8–10 eV for  $\text{Si}_6\text{O}_{1-10}$  and greater (14 eV) for the  $\text{Si}_6\text{O}_{11}$  cluster. VEAs vary from 0.9 to 1.5 eV. These results indicate that  $\text{Si}_6\text{O}_m$  structures obtained from the MGAC/MSINDO/B3LYP production are stable against losing or adding electrons. We conclude that  $\text{Si}_6\text{O}_3$  and  $\text{Si}_6\text{O}_4$  are the most stable isomers among Si-rich clusters and  $\text{Si}_6\text{O}_{9-11}$  is the most stable among O-rich clusters. Figure 11 shows a similar small LUMO-HOMO gap for  $\text{Si}_6\text{O}_5$  and  $\text{Si}_6\text{O}_6$ , the clusters with O/Si close to one.

$\text{Si}_6\text{O}_8$  was not included on Figs. 10 and 11 because the B3LYP-VEA is negative for our convention. So we evaluated the affinity for this system, allowing the geometrical relaxation (adiabatic electron affinity, AEA). B3LYP did not find an appropriate state for the anion. Instead, the B3PWN functional gave the right sign (AEA=1.027 eV) (Ref. 45) indicating that B3LYP cannot represent properly the  $\text{Si}_6\text{O}_8$  anion.

We report in Table III principal modes for the most stable  $\text{Si}_6\text{O}_m$  ( $m=1-11$ ) clusters produced in this work. They are identified, in the same table, with the vibration frequencies of the corresponding fragmentation products measured by Anderson and Ogden.<sup>13</sup> It offers an additional evidence of the presence of 2MR and 3MR fragmentation products.

Figure 12 shows the HOMO and the LUMO for the best isomer of each system, which presents 2MR, 3MR, or 4MR motif. It is interesting to highlight that the HOMOs and LUMOs are located preferably in those regions of the cluster where those motifs are not present, i.e., outside of the motifs. This is the case of the systems  $\text{Si}_6\text{O}_9$ ,  $\text{Si}_6\text{O}_{10}$ , and  $\text{Si}_6\text{O}_{11}$ .

TABLE III. Vibration frequencies for  $\text{Si}_6\text{O}_m$  ( $m=1-11$ ) clusters and some of their fragmentation products. We report the frequencies for principal modes of the most stable isomer of each cluster.

$\text{Si}_6\text{O}_m^a$	Vibration frequencies		Fragmentation product <sup>b</sup>	$\text{Si}_6\text{O}_m^a$	Vibration frequencies		Fragmentation product <sup>b</sup>
	Principal modes				Principal modes		
$\text{Si}_6\text{O}$	851.5		$(\text{Si}_6)$	$\text{Si}_6\text{O}_8$	1018.1	1008.4 <sup>d</sup>	$(\text{SiO}_2)$
	466.7	460.0 <sup>c</sup>			1038.0	1079.0	
$\text{Si}_6\text{O}_2$	903.1	1008.4 <sup>a</sup>	$(\text{SiO}_2)$	$\text{Si}_6\text{O}_9$	1007.7	691.1 <sup>a</sup>	$(\text{SiO}_2)$
	957.7	957.6 <sup>a</sup>			1100.9	957.6 <sup>a</sup>	
$\text{Si}_6\text{O}_3$	977.3	972.6 <sup>d</sup>	$(\text{Si}_3\text{O}_3)$	$\text{Si}_6\text{O}_{10}$	972.6 <sup>d</sup>	972.6 <sup>d</sup>	$(\text{Si}_3\text{O}_3)$
	794.7	766.6 <sup>a</sup>			1101.6	776.5	
$\text{Si}_6\text{O}_4$		766.3 <sup>d</sup>	$(\text{Si}_2\text{O}_2)$	$\text{Si}_6\text{O}_{11}$	870.2	801.3 <sup>a</sup>	$(\text{Si}_2\text{O}_2)$
	1012.7	1008.4 <sup>a</sup>			935.3	1008.4 <sup>a</sup>	
$\text{Si}_6\text{O}_5$	1010.2	1008.4 <sup>a</sup>	$(\text{SiO}_2)$	776.5	766.6 <sup>a</sup>	$(\text{SiO}_2)$	
	942.5	954.8 <sup>a</sup>		801.6	801.3 <sup>a</sup>		
$\text{Si}_6\text{O}_6$	975.9	957.6 <sup>a</sup>	$(\text{Si}_3\text{O}_3)$	801.6	801.3 <sup>a</sup>	$(\text{Si}_2\text{O}_2)$	
		972.6 <sup>d</sup>		804.7 <sup>d</sup>	804.7 <sup>d</sup>		
$\text{Si}_6\text{O}_7$	935.6	1008.4 <sup>a</sup>	$(\text{SiO}_2)$	932.8	1008.4	$(\text{SiO}_2)$	
	1000.4						

<sup>a</sup>This work.

<sup>b</sup>Indicated in  $\text{Si}_6\text{O}_m \rightarrow \text{Si}_k\text{O}_l + \text{Si}_{6-k}\text{O}_{m-l}$ .

<sup>c</sup>Geometry taken from Ref. 30.

<sup>d</sup>Vibration frequencies from the mass spectra reported in Ref. 13 for the fragmentation products.



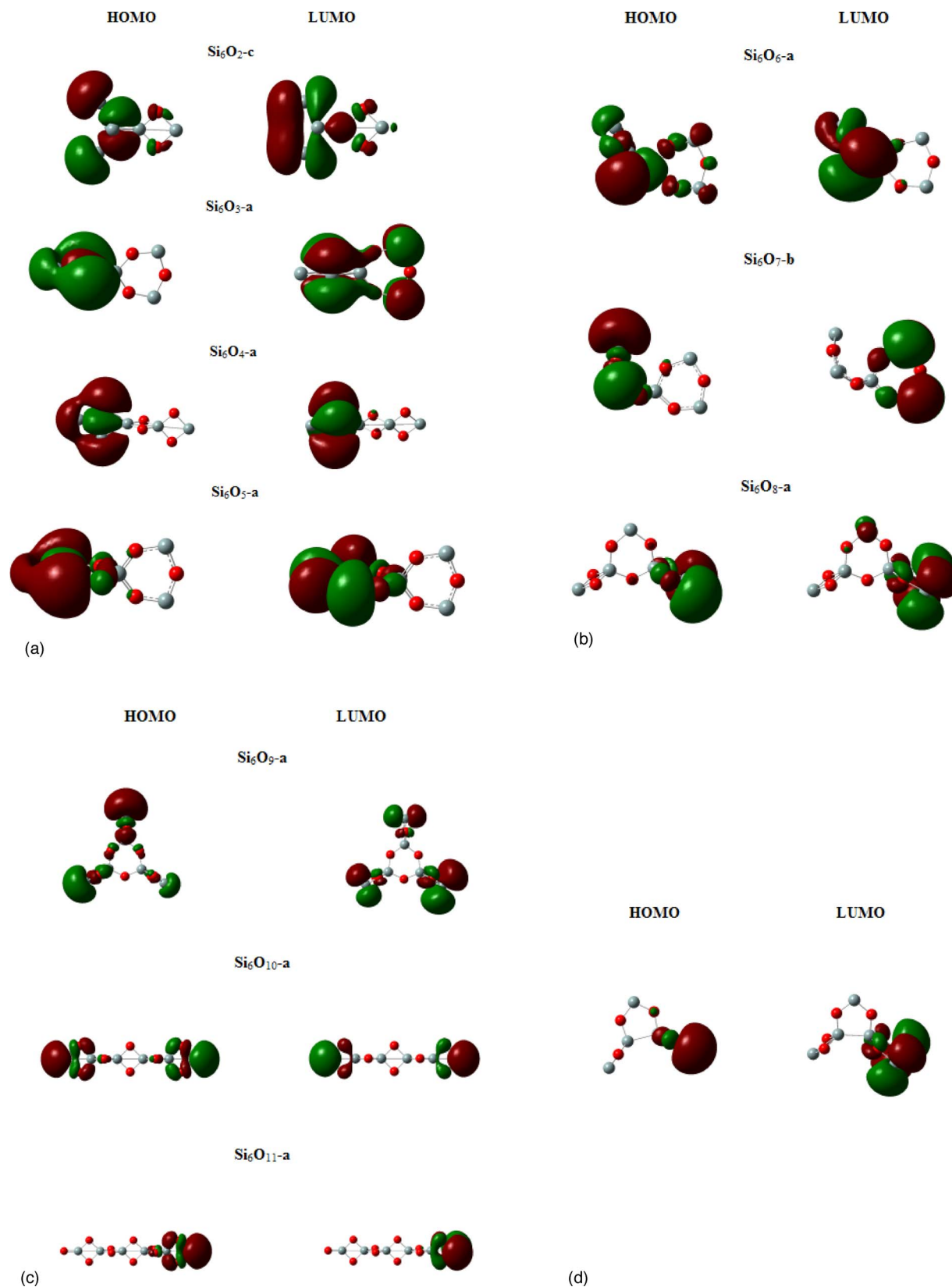


FIG. 12. (Color online) (a–c) HOMOs and LUMOs of the most stable  $\text{Si}_6\text{O}_m$  ( $m=2-11$ ) clusters which present 2MR, 3MR, or 4MR motifs. The iso-value of the contour map is 0.02. (d) HOMO and LUMO of the  $\text{Si}_6\text{O}_7\text{-c}$  cluster which present  $\text{Si}_3\text{O}_2$  motifs. The iso-value of the contour map is 0.02.

This behavior was mentioned by Wang *et al.* in the Ref. 25, the HOMOs and LUMOs are mostly located at pure silicon clusters.

Static dipole polarizability is one of the fundamental

properties of any atomic or molecular system. It has been extensively studied with theoretical as well as with experimental methods. In the field of atomic clusters, dipole polarizability has become one of the most important quantities to

TABLE IV. For each value of  $m$  (number of oxygen atoms) we report  $\alpha_{\text{best}}$ , the polarizability of the most stable isomer, and  $\sigma$  (vertical bars in Fig. 13), the dispersion of the polarizabilities of the other isomers around  $\alpha_{\text{best}}$ . Also, we report the polarizability anisotropy  $\Delta\alpha$  for silicon oxides. All data are expressed in a.u.

System	$\alpha_{\text{best}}$	$\sigma$	$\alpha_{\text{best}}/m$	$\Delta\alpha$	$\Delta\alpha/m$
Si <sub>6</sub> O	182	8	182	3183	3183
Si <sub>6</sub> O <sub>2</sub>	179	13	89	2835	1417
Si <sub>6</sub> O <sub>3</sub>	186	6	62	13 878	4626
Si <sub>6</sub> O <sub>4</sub>	174	12	44	19 729	4932
Si <sub>6</sub> O <sub>5</sub>	178	3	36	6234	1247
Si <sub>6</sub> O <sub>6</sub>	169	6	28	1492	249
Si <sub>6</sub> O <sub>7</sub>	151	7	22	348	50
Si <sub>6</sub> O <sub>8</sub>	152	1	19	1409	176
Si <sub>6</sub> O <sub>9</sub>	141	4	16	2641	293
Si <sub>6</sub> O <sub>10</sub>	136	1	14	10 892	1089
Si <sub>6</sub> O <sub>11</sub>	132	3	12	11 437	1040

guess the clusters geometry.<sup>46</sup> The dipole polarizability indirectly provides a measure of the extent of distortion in the electron density and hence the response of the system under the effect of an external static electric field. It is sensitive to the nature of the bonding and the geometrical structure of the system. Several authors studied the relation between cluster shape and polarizability per atom.<sup>47,48</sup>

In general, the static response properties of a molecule can be defined by expanding the field-dependent energy  $W(E)$  as a series in the components of a uniform electric field  $E$ ,

$$W(E) = W_o + \mu_i E_i + \frac{1}{2} \alpha_{ij} E_i E_j + \dots \quad (i, j = x, y, z), \quad (8)$$

where  $W_o$  is the energy of the system in absence of electric field,  $\mu_i$  is the dipole moment, and  $\alpha_{ij}$  is the  $ij$  component of the polarizability tensor (repeated Greek subscripts imply summation over Cartesian coordinates,  $x$ ,  $y$ , and  $z$ ).

$$\alpha_{ij} = \left( \frac{d^2 W}{dE_i dE_j} \right)_{E=0}. \quad (9)$$

In this work, we computed the static dipole polarizabilities for ground state structures. The calculations have been carried out in the framework of DFT implemented in the GAUSSIAN code,<sup>42</sup> using B3LYP exchange correlation functional and 6-311G(2d) basis set. The mean polarizability  $\alpha$  and the polarizability anisotropy  $\Delta\alpha$  for silicon oxides were calculated from equations

$$\alpha = \frac{1}{3} (\alpha_{xx} + \alpha_{yy} + \alpha_{zz}) \quad (10)$$

and

$$\Delta\alpha = \frac{1}{2} \sum_{i,j=x,y,z} (\alpha_{ii} - \alpha_{jj})^2. \quad (11)$$

The polarization anisotropy  $\Delta\alpha$  seems to be the convenient quantity to evaluate how much the cluster shape differs from one spherical molecule.

The results for  $\alpha$  and  $\Delta\alpha$  are reported in Table IV and depicted in Figs. 13 and 14. For each value of  $m$  (number of oxygen atoms) we report  $\alpha_{\text{best}}$ , the polarizability of the most

stable isomer, and  $\sigma$  (vertical bars in Fig. 13), the dispersion of the polarizabilities for the other Si<sub>6</sub>O <sub>$m$</sub>  isomers around  $\alpha_{\text{best}}$ , defined as

$$\sigma_m = \frac{\sum_{i=1}^{N_m} (\alpha_{im} - \alpha_{bestm})^2}{N_m}, \quad (12)$$

where  $N_m$  is the number of isomers for each Si<sub>6</sub>O <sub>$m$</sub>  series.

We can see that the  $\alpha/m$  values present an asymptotic tendency as a function of the number of oxygen atoms  $m$ , different from the behavior of the homonuclear silicon clusters.<sup>30,35</sup> for which the quantity  $\alpha/\text{atom}$  is approximately constant. This behavior is curious and very interesting. Clusters with large LUMO-HOMO gaps are indicative of chemical inertness<sup>44</sup> and the dipole polarizability indirectly provides a measure of the extent of distortion in the electron density, so clusters with large LUMO-HOMO gaps will have low polarizability.<sup>49,50</sup> Our hypothesis is that the oxygen electronegativity plays an important role in the tendency shown in Fig. 13. Oxygen pulls the electrons and successive oxidation might contribute to increase the “system electrone-

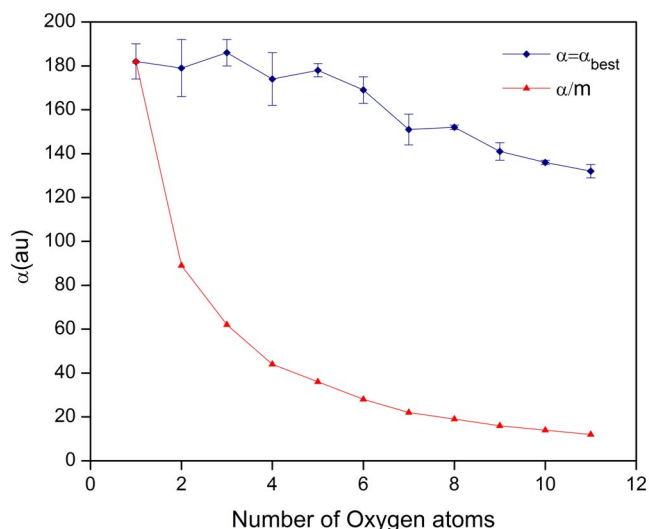


FIG. 13. (Color online) The polarizability of the most stable isomer,  $\alpha_{\text{best}}$  ( $\blacklozenge$ ) and  $\alpha/m$  ( $\blacktriangle$ ), as a function of  $m$ , the number of oxygen atoms. The vertical bars in  $\sigma$  indicate the dispersion of the other Si<sub>6</sub>O <sub>$m$</sub>  isomers around  $\alpha_{\text{best}}$ , as defined in Eq. (12).

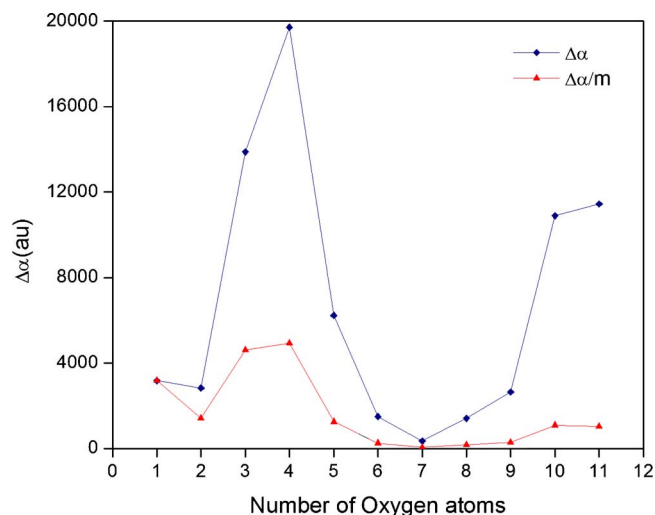


FIG. 14. (Color online) The polarization anisotropy,  $\Delta\alpha$  (◆) and  $\Delta\alpha/m$  (▲), as a function of  $m$ , the number of oxygen atoms.

gativity” causing the polarizabilities to decrease. To our knowledge, there are neither calculations nor experimental data of silicon oxides polarizabilities in the literature. Present evaluations show that this topic needs of further studies to provide additional explanations for the very interesting observed tendency. Note that  $\Delta\alpha$  in Fig. 14, which is positive by definition, is in total correspondence with the cluster shape, oblate, or prolate, illustrated by the inertia momentum depicted in Fig. 7.

#### IV. CONCLUSIONS

We have shown that our results agree well with previously reported structures and we found new ones. The system prefers to form 2MR and 3MR motifs. This characteristic is important in the geometric determination of Si<sub>6</sub>O<sub>9–11</sub> clusters, which present only combinations of 2MR and 3MR.

The behavior of BE per atom calculated in this work reproduces the features observed by other authors. For  $m < 6$ , the dissociation proceeds to SiO, while for  $6 \leq m \leq 12$  the preferred decay channel is  $(m-6)\text{SiO}_2$  plus  $(12-m)\text{SiO}$  molecules. For  $m \geq 12$ , finally, the favored channel is SiO<sub>2</sub> plus oxygen. The minimum energy fragmentation pathways contain Si<sub>2</sub>O<sub>2</sub>, Si<sub>3</sub>O<sub>3</sub>, and SiO clusters so these fragmentation products are considered to be relatively stable clusters.

We did not find experimental data or theoretical evaluations of Si<sub>6</sub>O<sub>m</sub> polarizabilities in the literature to compare with our calculations, but the behavior of the polarizability against oxidation is very interesting and needs further investigation. However, we note that  $\Delta\alpha$  is in total correspondence with the cluster shape compact, oblate, or prolate.

The behavior of the LUMO-HOMO gap shows that it is more difficult to ionize Si-rich clusters than O-rich clusters. Si<sub>6</sub>O<sub>3</sub> and Si<sub>6</sub>O<sub>4</sub> are the most stable isomers among Si-rich clusters and Si<sub>6</sub>O<sub>9–11</sub> is the most stable among O-rich clusters. Finally we observed that HOMOs and LUMOs are located preferably in those regions of the cluster where those motifs are not present, i.e., outside of the motifs.

#### ACKNOWLEDGMENTS

Financial support to this work from the University of Buenos Aires and from the Argentinean CONICET is gratefully acknowledged. Generous resources allocations at the Center for High Performance Computing (CHPC-Utah University) are also acknowledged.

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